



Universidade do Minho Escola de Engenharia

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Production of cellulose-based bioplastics from agroindustrial residues

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Production of cellulose-based bioplastics from agroindustrial residues

Tese de Doutoramento Gestão e Tratamento de Resíduos

Trabalho efetuado sob a orientação de Professora Doutora Ana Vera Alves Machado Professora Doutora Maria Cândida Lobo Guerra Vilarinho

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University of Minho, 31 of October of 2019

David Jefferson Cardoso Araújo

RESUMO

Production of cellulose-based bioplastics from agroindustrial residues

O desenvolvimento de plásticos biodegradáveis e de base biológica desempenham um papel fundamental na transição para uma economia circular, nomeadamente em aplicações de caso único, como embalagens. Esta tipologia de materiais basia-se no uso de matérias-primas renováveis, na possibilidade de reciclagem orgânica e na economia de recursos por reprocessamento. A biomassa lignocelulósica destaca-se como uma fonte promissora de biopolímeros naturais, sendo os resíduos agroindustriais uma fonte atraente devido à sua disponibilidade, baixo custo e alto teor de celulose. Paralelamente, é igualmente importante que os produtos emergentes de base biológica acompanhem as tendências do mercado, nomeadamente no uso de técnicas de processamento ecológicas e no desenvolvimento de materiais com propriedades ativas. No presente trabalho pretendeu-se desenvolver bioplásticos de base celulósica a partir de resíduos agroindustriais. Estes foram identificados com base num diagnóstico e na implementação de um modelo linear, tendo siso selecionado o caroço do milho como matéria prima a estudar. Um pretratamento combinado de dois estágios (tratamento hhidrtérmico seguido da reação com solução diluída de hidróxido de sódio) foi realizado para extrair a sua fração celulósica. A celulose extraída foi utilizada para produção de acetato de celulose atavés de uma técnica acetilação sem a presença de um meio solvente para dissolução. Na síntese do acetato de celulose obteve-se um grau de substituição de 2,68 e um rendimento de 60%. O acetato de celulose produzido apresentou ainda características adequadas para a preparação de filmes com características mecanicas significativas. Avaliação do ciclo de vida foi realizada para contabilizar os impactos ambientais associados à aplicação conjunta das ténicas de pré-tratamento e acetilação. A análise de ciclo de vida mostrou que as técnicas utilizadas repercutiram em impactos ambientais menos significativos que os gerados por abordagens convencionais. Um bionanocompósito foi produzido através da dissolução do acetato de celulose sinterizado em uma suspensão coloidal contendo nanopartículas de dióxido de titânio e óxido de magnesio, obtida a partir da ablação a laserdos metais percussores em diclorometano. A técnica utilizada possibilitou a produção de filmes bioplásticos sem o uso de qualquer outro reagente químico ou geração de subprodutos.

Keywords Resíduos agroindustriais; bioplásticos; celulose; pré-tratamentos verdes; acetato de celulose; bionanocompósitos; análise de ciclo de vida

ABSTRACT

Produção de plásticos baseados em celulose a partir de resíduos agroindustriais

Bio-based and biodegradable plastics play a key role towards the successful transition to a circular economy, namely regarding the application of single-use products, such as packaging. The benefits brought by this class of materials rest on the use of renewable resources of raw-materials, organic recycling and resource-saving by reprocessing. Lignocellulosic biomass stands out as a promising source of natural biopolymers, being the agroindustrial residues are especially attractive alternative due to their availability, low-cost and high cellulose content. On top of that, it is likewise important that the emerged bio-based products keep up with the latest trends on the market, which are focused on the use of eco-friendly processing techniques and development of active materials. In this regard, the present thesis is intended to develop cellulose-based bioplastics from agroindustrial residues. Potential agroindustrial residues were identified based on a diagnosis study and on the implementation of a linear model, from which corncob was selected as the raw material to be studied. A combined two-stage liquid hot water/dilute-NaOH pretreatment was carried out to extract its cellulose content. Furthermore, cellulose acetate (CA), a recognized biodegradable polymer, was synthesized from the extracted cellulose by a solvent-free method using iodine as catalyst. The cellulose acetate synthesis resulted in substitution degree of 2.68 and yield of 60%. The synthesized CA was suitable to produce films with remarkable mechanical properties. A life cycle assessment (LCA) was modeled in order to count the environmental impacts resulting from the joint application of pretreatment/acetylation techniques. The LCA showed that, in response to the less consumption of chemicals and high yield of acetylation, the proposed techniques have environmental impacts less significant than those of conventional approaches. A bionanocomposite was produced via the direct dissolution of the synthesized CA in a colloidal suspension containing titanium dioxide and magnesium oxide, the latter obtained through the laser ablation of bare metal targets in an appropriate solvent. The processing technique used allowed the production of bioplastic films without the use of any other chemical reagent or generation of byproducts.

Keywords: Agroindustrial residues; bioplastics; cellulose; green pretreatments; cellulose acetate; bionanocomposites; life cycle assessment

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Chapter 1. Introduction

This chapter aims to clarify the scope of this PhD project by enlightening the motivation, the objectives and the structure of the thesis.

1.1. Study background and motivation

Synthetic plastics have long since played an important role throughout many application domains, namely science, technology, industry and household, whether as an advanced or basic material. Their wide applicability, easy and low-cost processability, versatility and physical-chemical properties (e.g. tensile strength, durability, flexibility, low weight, resistance to the action of microorganisms and weathering) have contributed to their attractiveness and demand, which fostered the rapid development of polymer industry in the last decades. In 2017, over 348 million tonnes of synthetic plastics (mainly derived from fossil resources) were manufactured around the world [1]. From this total, the polyolefins accounted for more than 50%, which were mainly used for packaging production [2]. However, the same properties that make synthetic plastics a ubiquitous class of material, can also unleash negative environmental effects if they are mismanaged or littered. For instance, a great share of macroplastics-related findings in the ocean or in coastal areas corresponds well with the main sources of short-lived consumer goods ending up in municipal solid waste systems, which include mainly packaging [2].

Recently, the growing concerns over economic and environmental issues regarding synthetic plastics' value chain, namely oil market price fluctuations, finiteness of fossil raw materials and biotic/abiotic pollution of earth's compartments, have become nucleation supports for new sustainable thoughts on polymers' life cycle. At the same time, the narrowing of legal frameworks regarding the production, consumption and disposal of plastics has increasingly forced a change of position at the social and organizational level. The progress of good practices in this area tends to continue as governments and businesses are narrowing and driving their policies towards a circular economy model.

Ongoing practical applications of circular economy are dated since the late 1970s, but only recently it gained notoriety and momentum [3]. The idea of circular economy emphasizes that, by employing a regenerative approach through reuse, refurbishment, repair and recycling, the value and time of materials would be maximized, while the demand of resources and waste generation would decrease [4]. As a part of public policy, the change of a linear economy to a circular one could provide economy grow and opportunities in line with social development and environmental protection. A promisor route identified in countries trying to promote circular economy as national policy deals with the substitution of non-renewable resources by biomaterials. This aspect is in synergy with the principles of bioeconomy, which, conceptually, comprises any value chain that uses biomaterials as starting point, such as agriculture and forestry sources [5]. In the field of

polymers, the development of biopolymers has shown a dynamically growing market and a central theme of research in countless vehicles of scientific communication. Bioplastics can be bio-based, biodegradable or feature both properties. Bio-based polymers have been one of the most relevant drivers for the development of the European Union bioeconomy, with recognized potential to accelerate the transition to a circular economy. Global production capacities of bioplastics are predicted to grow from around 2.11 million tonnes in 2018 to approximately 2.62 million tonnes by 2023 [6].

To date, the bio-based plastics with highest growth rates on the market are bio-based polyethylene (PE), bio-based polyethylene terephthalate (PET), polylactic acid (PLA) and polyhydroxyalkanoates (PHA) [6,7]. Although they do have better environmental performance in comparison to the oil-based options, they have been still criticized for being not completely bio-based, nonbiodegradable or derived from first-generation products. In response to the aforementioned issues, there has been a growing interest in the development of biodegradable plastics from renewable polymer sources that have technological, economic and environmental viability, and that are also not eligible for feed or food production. In this context, lignocellulosic biomass has been pointed out as one of the most promising renewable alternatives since it is mainly composed by natural polymers (cellulose, hemicellulose and lignin) and available worldwide in the form of agroindustrial, forestry and municipal residues.

Annually, billions of tons of agroindustrial residues are generated by agri-food industries operations. These residues are largely available in the form of harvesting residues or byproducts and, despite some of them are used for added-value applications, the largest share is typically collected, burned, left on the ground to decompose or disposed of in municipal solid waste systems. Therefore, in addition to be a low-cost and neutral-carbon alternative, the agroindustrial residues are also greatly available as a biopolymer source, and their valorization could help to prevent environmental pollution and would foster the circular economy.

Among the main components of lignocellulosic biomass, the cellulose stands out as the most abundant renewable organic material in the biosphere [8]. Its availability and numerous properties (e.g. renewability, low cost, biodegradability, biocompatibility, chemical and thermal stability as well as its ability to form derivatives) makes it an especially attractive biopolymer [9, 10]. However, in the cell wall of plants, cellulose is incorporated (and tightly bound by covalent bonds) into a matrix of lignin and hemicellulose and, when it comes to the production of cellulose-based bioplastics, the coexistence of these other constituents may difficult cellulose accessibility and

entail undesirable properties to the final products. As a result, pretreatments are usually carried out to disrupt the compact structure of agroindustrial residues and extract the cellulose content.

There are different techniques available to pretreat lignocellulosic biomass (LCB), which may pose distinct effects on extraction efficiency of biomass components and on their physicochemical properties. In addition to influencing subsequent processing of LCB components, the pretreatment step is also considered one of the main bottlenecks within LCB valorization. For instance, in terms of environmental feasibility, the most conventional and commonly applied pretreatments make use of harsh, toxic and corrosive chemicals, and usually demand long processing times. As these issues may jeopardize the environmental sustainability of bio-plastic production, in recent decades there has been a growing interest in the use of "green" technologies for cellulose extraction, among which stand out liquid hot water (LHW), ionic liquids (IL), as well as combined pretreatments. The potential application of cellulose-enrich samples emerged from agroindustrial residues to produce bio-based packaging have already been presented in various studies [11–16]. Notwithstanding, conventional pretreatments (e.g. alkali-acid treatments) are still commonly applied to extracted cellulose-enrich pulp from biomass.

Another issue that challenges the processing of pure cellulose for bio-based packaging production arises from its non thermoplasticity and poor solubility in many organic solvents. Consequently, cellulose is usually derivatized in order to enhance its physical-chemical properties and widen its functionalities and applications. To date, cellulose acetate is one of the most important cellulose derivatives. For commercial purpose, the largest fraction of cellulose acetate is synthesized from wood and cotton pulp and manufactured through the acetic acid process route. Recently, some methodologies claimed as green (e.g. iodine-catalyzed solvent-free method and homogeneous acetylation in ionic liquid [17, 18]), were successfully applied to synthesize cellulose acetate from lignocellulosic biomass. However, most of them lack environmental assessment studies, and their eco-friendly aspect is simply assumed based on the substitution of catalysts.

Taking into account the commercial value and properties of cellulose acetate films (including thermoplasticity, chemical resistance, low water permeability and biodegradability), they come across as an interesting option for food packaging applications. Nevertheless, it is important to bear in mind that, recently, new classes of packages with additional active and smart properties are increasingly being demanded by market constituencies. Active packages with antimicrobial properties have been one of the most explored products. At scientific research level, the nanotechnologies have been greatly harnessed to this purpose. Recent studies have shown

excellent results at the level of metal/metal oxide nanoparticles (NPs) synthesis (such as silver, zinc, titanium dioxide and magnesium oxide) and their antimicrobial effects when embedded into potential polymer-based packages (synthetic or bio-based) [19–21].

However, although substantial advances have already been made in the field of active packaging, namely in antimicrobial packaging containing metal/metal oxide NPs, the development of such products making use of renewable polymer sources along with eco-friendly techniques throughout the production chain is still lacking. Far less attention has been given to the implementation of environmental assessment studies to support the eco-friendly aspect of the concerned products and the integrated manufacturing techniques used. An effective and standardized tool suitable for this purpose is the Life Cycle Assessment (LCA) methodology. Depending on the significance of the data and methods used in the LCA construction, the results obtained may present sufficient validity and provide reliable justifications to support the choice of the best alternatives studied.

1.2. Objectives and structure of the thesis

Based on the foregoing considerations, this work proposes the development of cellulose-based bioplastics from agroindustrial residues by using environmentally friendly experimental procedures, with potential to be applied as packaging component. To achieve this, key tasks have been performed with the aim of: (i) making a diagnosis of the Brazilian agroindustrial sector with the view of identifying residues suitable for the concerned application; (ii) performing appropriate pretreatments technologies to extract cellulose from the selected agroindustrial residue; (iii) modifying the extracted cellulose into cellulose acetate by an eco-friendly synthesis route, thereby widening commercial acceptability; (iv) investigating, by life cycle assessment, the environmental impact of the whole production process applied to synthesize the cellulose acetate (including pretreatment and acetylation steps); (v) and designing a novel eco-friendly processing technique to produce cellulose acetate/metal oxides bionanocomposites with potential to be applied as active packaging.

This thesis is organized into seven main chapters stemming from research papers published and submitted in international ISI-indexed books and journals. The current chapter (**Chapter 1**) is composed by the motivation, objectives and structure of the thesis, which seeks to provide context to the following discussed issues and to give an overview about the subject addressed in each section of the thesis.

Fundamental concepts and general aspects related with the main topics covered by the thesis are revised in **Chapter 2**. Particular focus is given to the main resources and techniques applied in this work, including structure and sources of lignocellulosic biomass, properties of cellulose, green pretreatments, cellulose acetate, cellulose-based packaging derived from agroindustrial residues, pulsed laser ablation in liquids and life cycle assessment. This chapter was partially derived from the following conference proceeding:

Araújo, D. J. C., Vilarinho, M. C. L. G., Machado, A. V. Agroindustrial residues as cellulose source for food packaging applications. 5th International Conference: Wastes: solutions, treatments and opportunities III. Vilarinho et al. (Eds). Taylor and Francis, pp. 217-223, 4-6 September, Lisbon, Portugal, 2020. ISBN: 978-0-367-257774.

Chapter 3 presents a diagnosis of the Brazilian agroindustrial sector and highlights the main agroindustrial residues produced in the country. In this research, besides the data collection and estimation of residues availability, a linear model was applied to identify the most suitable residues to the concerned application. This chapter was based on the following conference proceeding (oral presentation) and paper:

- Araújo, D. J. C., Vilarinho, M. C. L. G., Machado, A. V. Suitability of agroindustrial residues for cellulose-based materials production. 4th International Conference: Wastes: solutions, treatments and opportunities II. Vilarinho et al. (Eds). Taylor and Francis, pp. 417-423, 25-26 September, Porto, Portugal, 2017. ISBN: 9781138196698.
- Araújo, D.J.C., Machado, A.V. & Vilarinho, M.C.L.G. Availability and suitability of agroindustrial residues as feedstock for cellulose-based materials: Brazil case study. Waste Biomass Valorization, 10: 2863, (2019). https://doi.org/10.1007/s12649-018-0291-0

Chapter 4 is related to the pretreatment of a selected agroindustrial residue (corncob). In this chapter, combined green pretreatments were performed individually and in combination in order to extract cellulose, and their fractionation capacity were assessed via physicochemical characterization. The best pretreatment technique was selected based, primarily, on the cellulose content and on the physicochemical properties of the pretreated samples. A preliminary assessment regarding the suitability of the obtained enrich-cellulose sample to produce bioplastics was attested via the production of a regenerated cellulose film. This chapter was based on the following and paper:

 Araújo, D.J.C., Machado, A.V. & Vilarinho, M.C.L.G. Effect of combined dilute-alkaline and green pretreatments on corncob fractionation: Pretreated biomass characterization and regenerated cellulose film production. Industrial crops and products, 141, 111785, (2019). https://doi.org/10.1016/j.indcrop.2019.111785.

In **Chapter 5** the extracted cellulose was derivatized into cellulose acetate: a recognized biodegradable material with higher commercial significance and applicability. This was accomplished by applying a claimed eco-friendly acetylation synthesis, which preceded the pretreatment step proposed in chapter 4. The suitability of the synthesized cellulose acetate was also verified by the production of films and their assessment through dynamic mechanical analysis. On top of that, the environmental performance of the integrated pretreatment/acetylation techniques was evaluated via life cycle assessment and compared with a conventional approach. This chapter was based on the following submitted paper:

 David Araújo, M. Cidália R. Castro, Aline Figueiredo, Maria Vilarinho, Ana Machado. Green synthesis of cellulose acetate from corncob: physicochemical properties and assessment of environmental impacts. Journal of Cleaner Production, (2019).

Chapter 6 address the production of a bio-based cellulose acetate/TiO₂/MgO nanocomposite. In this chapter, a novel eco-friendly and versatile synthesis route is proposed. A colloidal suspension containing TiO₂ and MgO nanoparticles was prepared via pulsed laser ablation in a liquid media suitable to dissolve the previously synthesized cellulose acetate, and a bionanocomposite film was successfully produced by solvent casting method. The nanoparticle synthesis and nanocomposite film properties were assessed by physicochemical characterization and dynamic mechanical analysis. TiO₂ and MgO nanoparticles are know from literature to confer additional active properties (e.g. biodegradability and antimicrobial activity) when embedded into polymeric matrices, and therefore the proposed approach stands out as a promising route for developing active food packaging. This chapter was based on the following submitted paper:

 David Araújo, M. Cidália R. Castro, Maria Vilarinho, Ana Machado. A novel eco-friendly synthesis processing to produce cellulose acetate/TiO2/MgO bionanocomposite films.
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Finally, **Chapter 7** sets out the main findings of the developed work, followed by recommendations for future work.

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Chapter 2. State of the art

This chapter aims to contextualize the mains aspects discussed in the following chapters, as well as to help readers with different backgrounds to a common level of knowledge on several subjects that are fundamental to this work.

2.1. Lignocellulosic Biomass: structure and sources

Lignocellulosic biomass is the most abundant and renewable source of biomass on earth, naturally generated from available atmospheric CO₂, water and sunlight through photosynthesis [1]. It encompasses all terrestrial plants, wastes (such as agroindustrial, forestry and municipal) and energy crops. With a global production capacity around 140 billion tons/year, the agroindustrial residues stand out as the main source of available lignocellulosic materials [2–4].

The term agroindustrial residues spans over two main categories, depending on the generation source: (1) crop residues and (2) industry processing residues. The crop residues include residues left in the field after the harvesting phase and comprise straw, branches, stover, leaves, stalks, roots, trimmings and pruning. On the other hand, crops such as sugarcane, grape, orange, potato, coffee, cocoa and coconut are particularly interesting to obtain processed products. From food processing industry, residues in the form of bagasse, pomace, husk, hull and peels are commonly generated. Despite the potential use of these residues to produce bio-based products (e.g. bio-energy, compost, fibers, bioactive compounds and polymers) they are still commonly disposed in landfills or burned in the field, which might cause serious pollution-associated problems. For instance, highlighted as one of the largest agricultural producers in the world, Brazil also ranks among the nations with the highest rates of CO₂ equivalent emissions from agricultural activities. In 2017 about 25 million tons of residues associated to the main crops produced in the country were burned, implying the emission of more than two thousand Gg (CO₂ equivalent) of methane and nitrous oxide [5], contributing to the global warming and health problems.

The agricultural crops can be classified accordingly to their growing cycle into temporary (those which are both sown and harvested during the same agricultural year) and permanent (plants that last for many seasons). Besides, the crop production can be geographical specific as three main factors influence its yield, namely soil fertility, water availability and climate [6]. These issues can lead to a spatio-temporal variability of residues generation and, hence, their knowledge is essential to a proper estimation of residues availability and to develop management strategies. Despite the assessment of residues availability can be done by applying Geographic information System tools [7], a more common and precise approach is based on the annual crop production and residue-to-crop ratio. Moreover, an appropriate estimative requires data on competitive uses and sustainable removal rates to meet environmental and economic issues.

As a typical lignocellulosic biomass, the agroindustrial residues are primarily composed of three natural organic polymers, namely cellulose, hemicellulose e lignin, as well as minor amounts of

proteins, pectin and extractives [8, 9]. LCB on a dry basis generally contain 50% of cellulose, 10-30% of hemicellulose and 20-40% of lignin [10]. This estimate slightly differs from chemical composition of agroindustrial residues (cellulose, 30-60%; hemicellulose, 14-40% and; lignin, 7-20%) [11, 12] as LCB also includes wood biomass, that refers to hardwood and softwood, which are denser, structurally stronger and contains larger amount of lignin [4]. In the cell wall of plants, cellulose is incorporated into a matrix of hemicellulose and lignin, linked together by covalent crosslinking, which ultimately results in the recalcitrant property of lignocellulosic biomass (Figure 2.1) [8, 13]. Cellulose is a semi-crystalline homopolymer consisting of a linear chain of several linked D-glucose units. Conversely, the hemicellulose is an amorphous heteropolymer formed by repeated polymers of pentoses and hexoses, while lignin is composed largely by three aromatic alcohols, and acts as glue by filling the gap between and around the cellulose and hemicellulose.

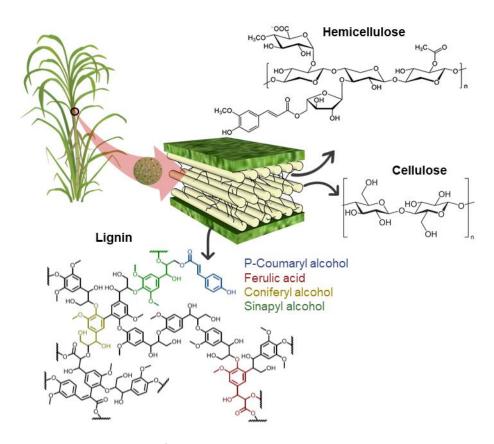


Figure 2-1. Structure of lignocellulosic biomass.

2.1.1. Hemicellulose

Hemicellulose is a branched heteropolymer with low polymerization degree, primarily formed by xylan and glucomannan chains intermixed with uronic acids and several kinds of monosaccharides, including glucose, mannose, galactose and pentose (comprising arabinose and xylose) [14, 15]. The variety and amount of each component depend on the lignocellulosic source. For instance, the agroindustrial residues are mainly formed by a backbone chain of β -1,4-linked xylose units, containing branched points formed by arabinose and glucose chains. In the cell wall of plants, the hemicellulose matrix act as a connection between the cellulose and lignin fractions, providing rigidity to the lignocellulose structure. It is attached to cellulose through an oxygen-hydrogen bond linkage and Van der Waals forces, and interacts with lignin through a hemicellulose–ferulic acid–ether bond–lignin structure (also called lignin-carbohydrate complex (LCC)) [16]. In response to its random and amorphous structure, the disruption and dissolution of hemicellulose are easier to occur in comparison to cellulose and lignin [17]. Hemicellulose can also be used for sustainable production of value-added products, such as xylitol, polymers and ethanol, but its conversion is still challenging. Notwithstanding, the economic viability of lignocellulosic biomass valorization into cellulose-based products is highly conditioned by the effective utilization of hemicellulose [18, 19].

2.1.2. Lignin

Lignin is a complex, amorphous and cross-linked polymer, located mainly in the secondary cell wall of plants where provides rigidity and strength, favors the transport of water and solutes in the vascular system and protects the cell against pathogens [20]. It is comprised mainly of three monolignols as the basic building blocks, including p-coumaryl, coniferyl and sinapyl alcohols. These components are polymerized into an irregular network via different aromatic or aliphatic ether bonds (β -O-4, 5-5', α -O-4, β -5, 4-O-5, β -1 and β - β), and are incorporated into lignin in the form of phenylpropanoids, namely p-hydroxyphenyl (H), guaiacyl (G), and syringyl (S) [14, 21]. Because of its no regular monomers sequences, lignin composition and structure are generally characterized by the relative abundance of H/G/S units, distribution of inter-unit linkages, as well as degree of condensation in the polymeric structure [22, 23]. The ratio of H:G:S constituents vary depending on the source of biomass and may play an important role during biomass pretreatment. For instance, sources of lignocellulose with high syringyl content, such as hardwood and some agroindustrial residues (i.e. rice straw, bagasse and corncob) are more easily hydrolysable via alkaline treatment than softwoods, which usually contain more guaiacyl [14, 24, 25]. This is because S-lignin features a higher level of labile β - \mathcal{O} -4' linkages which are readily cleavable during pretreatment [26]. Although many studies are emerging regarding the potential uses of lignin as a sustainable feedstock to produce value-added materials, such as polymers, fuel, aromatic compounds and carbon materials, this component is prominently complex and still challenging to study and valorize because of its variable composition and degree of cross-linking. Besides, the same beneficial recalcitrance property conferred by lignin to the plants is also largely responsible for the high cost and complexity of operations for lignocellulose conversions. Consequently, pretreatments are generally applied focused on removing the lignin, which is mainly relegated for heat and power in industrial plants [22, 23, 27, 28]. However, such as the hemicellulose, the use of lignin is a key issue for the techno-economic viability of lignocellulosic valorization.

2.1.3. Cellulose - structure and morphology

With an estimated annual production around 1.5 x 10¹² tonnes, cellulose is considered the most abundant biomass-derived organic compound in nature [29]. Its main sources include plants, sea animals, algae, fungi and bacteria, over which, in most cases, it acts as a reinforcing agent [30].

Cellulose is a high molecular weight homopolymer generated from repeating D-glucopyranose ring units linked by β -1,4-glycosidic bonds (cellobiose unit). Consequently, each unit of the cellulose chain is turned 180° with respect to the adjacent molecule [29, 31]. In literature, the individual glucose monomers within the cellulose chain are also referred to as anhydroglucose units (AGU). Each AGU unit contains three reactive hydroxyl groups, a primary group at C6 and two secondary groups at C2 and C3, positioned in the plane of the ring (Figure 2.2). Moreover, as typical of polymers formed by polycondensation, the chain ends of the cellulosic are chemically different, which gives it the characteristic of polarity, particularly important in shaping its different crystalline structures [32]. One end exhibits chemically reducing functionality, in equilibrium with the aldehyde structure (subject to oxidation reactions), while at the other end the anomeric carbon is involved in a glycosidic bond (non-reducing end) [29].

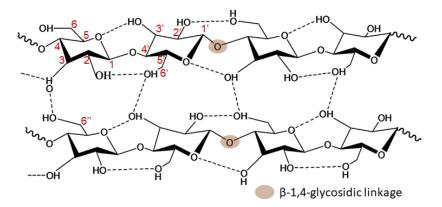


Figure 2-2. Chemical structure of cellulose chain. Source: modified from [33].

The hydroxyl groups of the AGU, along with the oxygen atoms of the D-glucopyranose ring and the glycosidic linkage, interact with each other by forming strong intermolecular and intramolecular hydrogen bonds [34]. Intramolecular bonds are formed by $O(3')H\rightarrow O(5)$ and $O(2)H\rightarrow O(6')$ interactions, which confer rigidity to the cellulose chain together with the β -1,4-glycosidic covalent linkage [33, 34] (Figure 2.2). This structural arrangement allows molecules to crystallize in a horizontal plane and parallel chains, forming microfiber bundles. In general, individual cellulose molecules are assembled into larger units, known as elementary fibers, which are concentrated into even larger units called microfibers, that are in turn grouped into cellulosic fibers [30, 31] (Figure 2.3).

When it comes to the interactions between cellulose chains, the intermolecular bonds are mainly produced by $O(6") \rightarrow O(3)$, and allow to form rather flat ribbons [35]. All these hydrogen bonding, along with other interactions, are responsible for the assembly of cellulose in layers and for conferring low solubility in common solvents and lack of thermoplasticity. Another important feature of cellulose deals with the wide chemical variability provided by the high reactivity of the hydroxyl groups.

The cellulose elementary fibers are comprised of crystalline regions (highly ordered) interspersed by amorphous (disordered) regions (Figure 2.3) [32]. The ordered regions are stabilized by an intra- and intermolecular hydrogen bonding network which, along with molecular orientations, can vary widely, resulting in cellulose polymorphs or allomorphs [31]. Four major crystalline cellulose structures (polymorphs), namely I, II III and IV, are known [29].

Cellulose I can crystallize into two different crystalline phases, I_{α} and I_{β} , coexisting in different proportions depending on the origin of the cellulose. Plant cellulose, including agroindustrial residues, mainly consists of I_{β} , while cellulose emerged from primitive organisms crystallize in the I_{α} phase. Both exhibit parallel chain configurations, but differ in hydrogen bonding patterns, implying differences in their crystal structures. While I_{α} presents triclinic cell unit, I_{β} exhibits monoclinic cell unit [29, 31, 34]. Cellulose II exists in a monoclinic phase and can be produced by treating cellulose I with sodium hydroxide solution (maceration) or by dissolving the cellulose followed by its regeneration [29, 36]. Unlike cellulose I, which has a parallel arrangement, cellulose II chains assume a more thermodynamically stable antiparallel arrangement [31, 37]. From different treatments (e.g. ammonia) of cellulose I or II, cellulose crystalline III, or III, can be obtained,

respectively. The latter, when treated at high temperatures with glycerol, result in the IV_1 and IV_2 polymorphs [31, 36].

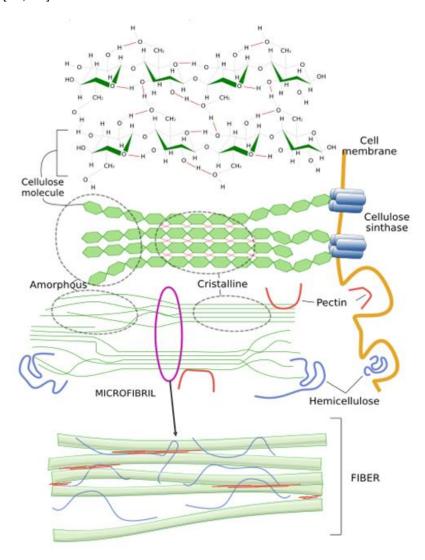


Figure 2-3. Morphological hierarchy of cellulose fibers. Source: [38]. (Accessed at September 2019)

Despite the wide availability and remarkable properties of cellulose, in the lignocellulosic residues this component is strongly linked to a matrix of lignin and hemicellulose [39]. The proportion between cellulosic and non-cellulosic constituents features a determinant role in the properties of fibers. The presence of substantial quantities of non-cellulosic materials can negatively influence their crystallinity, density, tensile strength, modulus of elasticity and moisture [40]. For instance, when not removed, these constituents can hamper chemical derivatization of fibers or endanger their application as reinforcing filler in polymeric matrices [40–42]. Therefore, for the successful use and valorization of cellulose from lignocellulosic biomass, pretreatments must be carried out to remove recalcitrant components.

2.2. Pretreatments of lignocellulosic biomass

Within the processes to convert LCB into value-added products, the pretreatment is an essential step, essentially carried out to disrupt the LCB compact structure and fractionate its main components [43] (Figure 2.4). Basically, the pretreatment methods can be classified into four categories, namely physical, chemical, physicochemical and biological. Each pretreatment method has specific effects on fractionation and physical-chemical properties of LCB components. In addition, several other factors may play a role in this step and directly influence downstream process, including the pretreatment settings, source of raw-material and its chemical composition, and even the climatic and geographical cultivation conditions [43, 44]. Currently, the pretreatment step is considered one of the main bottlenecks to the valorization of agroindustrial residues, and its cost may reach more than 40% of the total conversion process expanses [10, 44]. Therefore, the understanding and chosen of different techniques is imperative, and optimum selection must be led by the intended application of the pretreated sample.

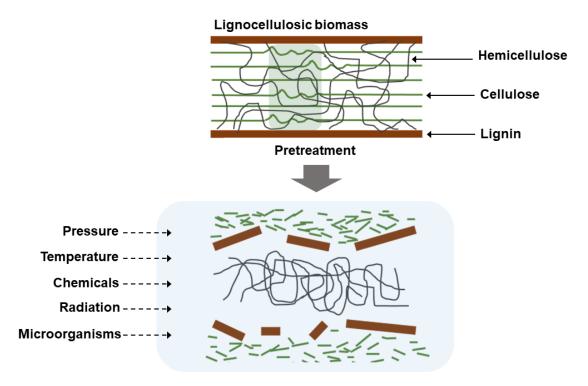


Figure 2-4. Pretreatment effect on lignocellulosic biomass. Modified from [45].

Furthermore, it is coherent that techniques with an environmental bias make up the precursor processing steps of eco-friendly products. However, the conventional techniques used for biomass pretreatments usually require harsh chemicals and extensive energy, in addition to generate excessive wastes. The most extensively used methods to pretreat LCB are acid and alkali

pretreatments. These are widely applied, followed by a bleach treatment, in bench scale experiments and pulp mills, even though atmospheric pollution and contaminants in wastewater effluent resulting from these chemical treatments are broadly documented [46, 47]. These drawbacks have induced the development of new approaches based on the "green chemistry" concept. Pretreatment techniques that fit into this concept include liquid hot water (LHW), stem explosion, wet oxidation, organosolvent, supercritical fluid, electron beam irradiation, ultrasonic energy, ionic liquids, deep eutetic solvent and biological treatment. Recently, great attention has also been focused on combining pretreatments [48]. The main advantages of the latter approach are that, by combining complementary treatments at different stages, the fractionation of specific components can occur easily, in a shorter processing time and by using less chemicals. For instance, a significant problem resulting from sodium hydroxide treatment, namely the black liquor generated as wastewater effluent, could be partially mitigated through this approach by using low concentrated solutions.

Therefore, the overall purpose of these new technologies deals with the use of non-environmentally hazardous chemicals, that can be recyclable or at least that produce less wastes [49]. Based on the foregoing, three pretreatments were selected to be applied in this work and will be discussed at some length in the next topics.

2.2.1. Liquid hot water

Liquid hot water is considered one of the most promising pretreatments due to its high efficiency and low cost, as it uses only water at elevated temperature (generally between 160-240 °C) and pressure (usually <5MPa) [50, 51]. In addition, it has several advantages over other conventional treatments since no chemical is needed, equipment Is little affected by erosion and, depending on the application, few inhibitory by-products are formed [52]. Depending on the LHW treatment parameters, the decomposition of main LCB components occurs at different rates. However, in general, it mainly leads to hemicellulose solubilization and hydrolysis. Conversely, lignin is only partially removed, and its insoluble fraction is almost completely retained or relocated in the solid residues along with the cellulose fraction [53].

During the LHW treatment the decomposition of hemicellulose is divided into three main steps, namely (1) surface reactions to produce primary products, (2) dissolution of primary products into the water and (3) decomposition of primary products [52]. Primary products, such as acetic acid, are released from hemicellulose as effect of water ionization, which in turn act as auto-catalysts in

the continued hemicellulose solubilization [54]. Because of the in situ generated acidic environment, the LHW pretreatment is categorized as a physicochemical method.

LHW treatment has been widely applied to pretreat agroindustrial residues [43, 44, 54–56]. Changes on microstructure and surface of LCB marked appear as cavities and cracks in the cell wall of pretreated samples in response of hemicellulose removal and lignin structural changes [53, 54]. Usually, the efficiency of LHW on biomass modification/fractionation is correlated with the severity factor (formulation that includes the combined effect of temperature and reaction time) [51]. Some expressions can also include the effect of pH (usually controlled between 4 and 7) since it may determine the formation of monomeric sugar and decomposition products. From literature, the optimal conditions for LHW treatment of agroindustrial residues are in the range of 180-220 °C and residence time varying from 5 to 30 min, at a severity factor limited to 3.64 -4.25 [54, 56]. Under these conditions, high levels of hemicellulose solubilization and low cellulose depolymerization are expected to occur, and the accessible surface area of fibers can be substantially increased, which ultimately has a positive impact in subsequent processing. In addition, the biomass recalcitrance is greatly reduced on account of β -0-4' linkages disruption and removal of acetyl groups of lignin [53]. By applying a severity factor of 4.13 in the treatment of corncob, Michelin et al. [56] obtained a hemicellulose extraction yield of 74.1%. Similar result (hemicellulose extraction yield of 77%) was obtained by Li et al. [53] for the LHW treatment of poplar at a severity factor of 4.2. Higher levels of hemicellulose hydrolyzation (in the range of 80-90%) have already been reported for other agroindustrial residues, including sugar-cane bagasse and rice straw [50, 55].

2.2.2. Pretreatment of lignocellulosic biomass with Ionic liquid

lonic liquids (IL) can be defined as salts containing an organic cation and an organic or inorganic anion present in the liquid phase at temperatures relatively close to room temperature (below 100 ° C) [57, 58]. In the last decades, numerous advances have been made towards understanding the synthesis, properties and mechanisms of action of ionic liquids, which has enabled its application in numerous manufacturing processes whether as components of technological devices, solvents or engineering fluids [59].

The following properties are generally reported in the literature as advantageous aspects of ionic liquids [59–61]: biodegradability, recyclability, low toxicity, low hydrophobicity, low viscosity, non-flammability, electrochemical and thermal stability, low volatility, ability to dissolve organic and

inorganic components, high polarity and wide electrochemical window. Nevertheless, through the combination of different anions and cations a wide range of ionic liquids can be synthesized, which makes it difficult to generalize their properties.

Regarding the dissolution of lignocellulosic biomass, ionic liquids have the advantage of being able to dissolve cellulose, lignin and hemicellulose under relatively moderate conditions without degrading their chemical structure [32]. Besides, they can be applied as an "environmentally friendly" technique as their low volatility minimizes the emission of pollutants.

For industrial scale use, the high cost associated with ionic liquids compared to conventional solvents represents one of the main limiting aspects of their application. However, due to their numerous properties, ionic liquids can be recycled and reused. In the literature, numerous attempts have been reported to reuse ionic liquids. Depending on the desired end product, different techniques may be employed, namely [62]: distillation, extraction of residual components, adsorption, induced phase separation, membranes and centrifugation. Good regeneration yields for few application cycles of recycled ionic liquids have already been reported [63–65]. For instance, by using a mixture of BmimCl and water for the treatment of vegetable straw, Wei et al. [63] observed a decrease in treatment efficiency only from the fourth application cycle.

The main ionic liquids currently used in cellulose dissolution are those that include the imidazole, pyridinium, ammonium and phosphonium cations [32, 66]. These may be linked to different radicals and associated with different anions, giving rise to a series of ionic liquids. When it comes to the pretreatment of lignocellulosic biomass, the main ionic liquids applied include 1-butyl-3-methylimidazolium chloride (BmimCl), 1-ethyl-3-methylimidazolium chloride (EmimCl), 1-allyl-3-methylimidazolium chloride (AmimCl) and 1-ethyl-3-methylimidazolium acetate (EmimMeCO2). Among these, it is worth mentioning BmimCl due to its power of dissolution [67–69].

Drawing from the high cellulose content available in lignocellulosic biomass, this latter has been extensively used as raw material in ionic liquid extraction processes. Two different approaches are currently used in the process of deconstructing lignocellulosic biomass with ionic liquids, namely dissolution process and ionosolv process. The dissolution process, which has the largest range of studies developed, consists in the solubilization of all components of lignocellulosic biomass. By contrast, the ionosolv process promotes the selective solubilization of LCB components. Currently, the ionosolv approach does not have fully optimized procedures, which reflects the scarcity of studies developed [66].

Within the dissolution process, ionic liquids capable of solubilizing all components of the lignocellulosic biomass are used, such as BmimCl and EmimMeCO₂. Once the biomass is solubilized, an antisolvent is used to dilute the solution and promote the precipitation of the cellulose. Such precipitated material may be removed from the solution by filtration and should be washed to remove residual ionic liquid. As a subsequent step there is the recovery of the ionic liquid, carried out by distilling the antisolvent and recovering the lignin and hemicellulose fractions in the solution.

It is currently accepted and proven that the dissolution property of ionic liquids is directly related to the ability of their anions (e.g. chloride and acetate) to form hydrogen bonds with the hydroxyl group of cellulosic material [32, 60]. It is worth mentioning that, due to the complex hierarchical structure of cellulosic fibers, their dissolution occurs distinctively. During IL pretreatment, a heterogeneous swelling and dissolution takes place firstly in the secondary wall of the fibers. As the cellulose swells, the primary wall rolls up in such a way as to form collars, rings, or spirals that restricts the uniform expansion of the fiber forming balloons (Singh et al., 2015). Notwithstanding, the effective dissolution of LCB depends on the multi-component structure and content of the biomass, as well as the solvation property of the IL [70].

One possible way of predicting the solvency power of a solvent is through empirical or semi-empirical polarity scale equations, such as the Kamlet-Taft model. The Kamlet-Taft equation is formulated as a function of the parameters β (hydrogen bond basicity), α (hydrogen bond acidity), π * (polarizability) and δ (dipolarity) [71]. More details about the Kamlet-Taft model and possible correlation of parameters can be found in Mäki-arvela et al. [71]. Especially in ionic liquids with non-functionalized cations, the parameter β is essentially affected by the anion. According to Brandt et al. [66], it is possible to verify that ionic liquids with a β value higher than 0.8 present higher cellulose dissolution capacity. Regarding the cations' role in the dissolution mechanism, this is still a subject of discussion among authors. However, increasing the alkali chain in the cation is likely to reduce cellulose solubility [66].

2.2.2.1. Regeneration process

Cellulose pulp obtained through the dissolution process is called regenerated cellulose. The regeneration is one of the most important steps in the treatment process as it will directly result in the product to be used. As previously mentioned, regenerated cellulose can be obtained by the

addition of an antisolvent to the solution, such as water, acetone, methanol, ethanol as well as mixtures of water and organic solvents [63, 64, 72].

When added to the IL-dissolved biomass solution, the antisolvent surrounds the anions of the ionic liquid and promotes the breakdown of their interactions with cellulose [61]. Firstly, intermolecular hydrogen bonds (glucopyranose rings stacking by hydrophobic interactions) are reconstructed forming sheet like structures. As coagulation proceeds, many sheet-like structures are stacked by intramolecular hydrogen bonds to form thin planar crystals. Some aggregates that are tightly stacked with each other and free of defects transform into the crystalline regions, while those that incorporate defects in their structure become less ordered domains or amorphous regions [70]. These randomly dispersed structural units make contact with other units and, by a diffusion cluster-cluster aggregation mechanism, form a regenerated material with a mixture of crystalline and amorphous regions [73].

By changing the regeneration parameters (e.g. coagulating solvent, time and temperature), regenerated cellulose with different shapes (such as powder, fibers, films, hydrogels, aerogels and spheres) and properties (e.g. mechanical, thermal and barrier) can be obtained [74, 75]. Regenerated cellulose in the form of films, emerged from ionic liquid-pretreated agroindustrial residues, with suitable properties to be applied as packaging, were produced by Reddy et al. [73], Vanitijinda et al. [76] and Li et al. [77].

2.2.2.2 Effect of IL treatment conditions

The first factor with direct influence on the efficiency of lignocellulosic biomass treatment with ionic liquids corresponds to the types of anions and cations. It is widely accepted that the main component responsible for cellulose dissolution is the anion. In general, the higher its basicity and the smaller its size, the greater is the dissolution capacity of the IL. Some examples of anions with superior cellulose dissolving capabilities are CI and methylimidazolium acetate (MeCO₂) [60].

Regarding the cation role, although several authors consider its contribution as a secondary factor, its influence on cellulose dissolution is mainly governed by the nature of existing functional groups, cation size and side chain unit [60]. Ionic liquids with heterocyclic aromatic functionalities, such as pyridinium and imidazole, exhibit better dissolution and fractionation of lignocellulosic components. Consequently, the largest share of studies on LCB treatment with IL make use of these cation categories [69, 73, 78–80]. The presence of allyl, methyl, ethyl and butyl radicals in the imidazole aromatic chain provides good dissolution performances. Moreover, according to the

results compiled by Badgujar and Bhanage, it is found that, by increasing the length of functional groups the dissolution capacity of ionic liquids is reduced [60]. As an example, the authors noted that the solubility property of four ionic liquids, with the same anion and cation categories, decreases in the following order: BmimCl > HexmimCl > OmimCl > DmimCl.

Due to the excellent dissolution capacity of BmimCl, as well as its lower cost relatively to ionic liquids with allyl and ethyl radicals, it has been widely applied in the treatment of lignocellulosic biomass. Abdulkhani et al. [81] used two imidazolium-cation ionic liquids, namely BmimCl and 1,3-methylimidazole dimethyl sulfate (DimimMeSO₄), to dissolve three different types of lignocellulosic biomass with different cellulose contents. The BmimCl presented dissolution performance much higher than DimimMeSO₄, and showed better efficiency for higher cellulose-content species. Similar result was obtained by Grävik et al. [69], in a study dedicated to evaluating four ionic liquids (BmimCl, EmimMeCO₂, BmimMeCO₂ and BmimHSO₄) for the treatment of different lignocellulosic species. The authors concluded that BmimCl has better treatment efficiency for simple cellulosic substrates. The main advantages of using BmimCl are its easy synthesis, stability and lower sensitivity to moisture compared to ionic liquids with acetate ions. However, its high viscosity and melting point may pose negative aspects.

In addition to the typology of cations and anions, several other factors, or treatment conditions, such as reaction time and temperature, particle size, biomass type and biomass / ionic ratio, as well as the presence of water in the ionic liquid or biomass may play an important role on IL treatment. Overall, the efficiency of ionic liquid treatment decreases with increasing recalcitrant biomass capacity. For instance, for the treatment of materials with high lignin content, higher reaction times and processing temperatures are required [61]. In addition, low concentrations of biomass appear to favor the dispersion of molecules in the solution and lead to a higher rate of dissolution and regeneration. Good solubility results are found in the literature with a biomass / ionic liquid mass ratio of less than 10% [82]. The use of particles smaller than 0.5 mm also appear to have better dissolution properties [66]. The temperature and reaction time required for the treatment of lignocellulosic biomass are defined primarily from the physical properties of the ionic liquid, such as its viscosity and melting point, as well as from the type of biomass. Increasing the temperature by 10 °C can significantly reduce the viscosity of ionic liquids [60]. In addition to favoring the dissolution of biomass, the increase in processing temperature is necessary because some ionic liquids are not liquid at room temperature.

2.2.3. Sodium hydroxide pretreatment

Sodium hydroxide (NaOH) is the most widely studied and effective alkaline reagent for LCB treatment. This chemical treatment has received great attention in the last decades as a potential pretreatment option because it is inexpensive, effective on a variety of feedstocks, and less energy intensive as compared to other pretreatment options [14]. Throughout the NaOH pretreatment of LCB, lignin is the main component affected. The main reaction mechanism for this effect is the cleavage of intermolecular ester bonds, by means disassociated hydroxide ions, between ferulic acid and the carbohydrate of the LCC [14]. The partial or total solubilization of lignin, as well as the effects over hemicellulose and cellulose fractions, depends on the pretreatment conditions (solution concentration, temperature, time and solid loading). Pretreatments performed with elevated NaOH concentrations seems to increase the solubilization of lignin and hemicellulose, but detrimental situations may also take place as hemicellulose degradation and cellulose losses can be set. High alkali concentrations can also trigger many structural and morphological changes in cellulose. Concentrations >6% can cause the swell of cellulose fibers, decrease the degree of polymerization and crystallinity index and lead to a transformation from cellulose crystalline I to II [14, 83]. At low alkaline concentrations, the lignin and hemicellulose fractions are partially solubilized, while cellulose remains relatively unaltered. Under this condition, the glycosidic linkages are alkali-stable and structural changes on cellulose are insignificant. In Haque et al. [84], the treatment of barley straw with low concentrated NaOH solution (0.5-2%) resulted in 84.8% removal of lignin and 79.5% removal of hemicellulose. In Talha et al. [85], a lignin removal around 86% was obtained by treating sugarcane bagasse with 1% NaOH solution at 100 °C for 3h.

The efficiency of NaOH treatment depends also on the lignin content and structure characteristic of the biomass, in a way that, sources of LCB containing a higher proportion of syringyl units are more easily delignified, such as rice straw, sugarcane bagasse and corncob [25]. On top of that, the alkali treatment works more effectively on low lignin content biomass, namely agroindustrial residues [44].

Some limitations regarding the sodium hydroxide pretreatment are the required long reaction times when operating in mild conditions and the production of liquors as a by-product. This liquid stream liquor contains inorganic chemicals and dissolved organic chemicals from biomass and usually features significant oxygen biological demand, which may pose risks to the environment if not treated [86]. In this regard, the use of low concentrated alkali solution, applied in combinations with other pretreatment technologies, can partially mitigate these issues.

2.2.4. Combined pretreatments

Studies have found that the combination of complementary treatments is more effective to fractionate the LCB than a single treatment. The combination of chemical, physicochemical and biological treatments, carried out as multi-stage or in simultaneous, have been extensively reported in the last few years [48]. For instance, it has been shown that the disruption of the cell wall layers induced by steam explosion and acidic hydrolysis, eases the dissolution of LCB in common solvents, such as ionic liquids [70]. High cellulose recovery was also obtained through the combination of an alkali solution (ammonia) and 1-ethyl-3-methylimidazolium acetate to pretreat rice straw [87]. In another approach, Wang et al. [88] combined biological pretreatment with LHW to pretreat *Populus tormentosa* and reported a hemicellulose removal rate of 92.33%, which could not be achieved by the individual treatments. Sun et al. [89] pretreated cercariae with two-step combined water and NaOH pretreatment and could increase cellulose extraction rate. By applying NaOH solution in combination with Fenton pretreatment, Zhang et al. [90] also obtained excellent recovery rates of straw cellulose. In Yoo et al. [91], a two-stage fractionation process using aqueous ammonia and hot-water efficiently fractionated the three main components of corn stover with relatively high purity. Ammonia treatment selectively removed lignin, while hot-water treatment separately hydrolyzed hemicellulose.

In addition to improve the fractionation of LCB, the combination of pretreatments can contribute to reducing costs, time of processing, as well as the use of chemicals and energy [48, 91]. However, understanding the structure of biomass, selecting a proper treatment method for target components and designing efficient process and optimal conditions are crucial prerequisites for the development of economically feasible combined process.

2.3. Cellulose modification and derivatives

Despite the advantageous properties associated with cellulose, another set of characteristics, intrinsic to its molecular structure (e.g. lack of antimicrobial properties, high hydrophilicity and low dimensional stability and thermoplasticity), limits its application as a polymeric component [92]. The existence of free hydroxyl groups in the cellulosic chain ends up making lignocellulosic fibers susceptible to oxidation, hydrolysis and swelling reactions, thus compromising their physical and chemical integrity [93]. In this sense, the enhancement of the physical-chemical properties of cellulose through modification is a fundamental strategy to maximize its application range.

Cellulose modification techniques can be classified into physical and chemical methods. Physical methods are mainly applied to change the surface energy of cellulose, which may result, for instance, in a better compatibility between the cellulosic fibers and polymeric matrices. Regarding the chemical modification, it is worth mentioning that this approach is the main modification technique used to improve the properties of natural polymers, including cellulosic fibers. Through chemical modification, new properties are provided to the cellulose by replacing or activating their hydroxyl groups without compromising its natural characteristics. Chemical modification can be subdivided into two main methods [94–96]: (i) direct chemical modification, with emphasis on silanization, alkalinization, esterification, etherification, maleated coupling, and permanganate and peroxide treatments; (ii) and the graft copolymerization method.

The most commercially important cellulose derivatives are synthesized through the direct chemical modification method and include cellulose esters (e.g. cellulose acetate, cellulose acetate butyrate, cellulose nitrate and cellulose acetate propionate) and ethers (e.g. methylcellulose, ethyl cellulose, hydroxypropyl methylcellulose and carboxymethyl cellulose). All of them feature a wide array of applications (ranging from coating to pharmaceutical components [42, 97–99]), and their common characteristic as a good film forming make them particularly suitable for packaging. With a projected market growth over the next years, cellulose acetate is by far the most important cellulose derivative (Malhotra, Keshwani, & Kharkwal, 2015; Zion Market Research, 2016), and this economic prospect can press forward even more its production from non-conventional sources, such as agroindustrial residues.

2.3.1. Cellulose acetate

Cellulose acetate is an amorphous, colorless, though and flexible natural polymer, widely used for industrial and consumer goods applications, such as filter, optical films, toys, eyeglass frames, home furnishings and osmotic drug delivery systems [102, 103]. It is also thermoplastic and easy to be processed into fibers and films [97]. As a film-like product, cellulose acetate also displays other particular properties, namely low haze, no odor, high moisture vapor transmission, low water permeability and excellent resistance to organic/inorganic weak acids and vegetable oils [97, 103], which are potentially desirable for semipermeable packaging.

Commercial cellulose acetate is typically produced from cotton linter or wood pulp-extracted cellulose via an esterification reaction, at which the natural cellulose is reacted with acetic anhydride and acetic acid (reaction medium) in the presence of sulfuric acid as a catalyst. In this

conventional method, known as acetic acid process, the mechanism of acetylation is mainly based on the formation of cellulose sulphate, resulting from the reaction of cellulose with sulphuric acid, which is transesterified from sulphate to acetate through a sulphoacetate intermediate stage [104, 105].

Depending on the degree of substitution (DS) of acetyl groups, the produced cellulose acetate can be classified in cellulose diacetate (DS of 2.4–2.6) or cellulose triacetate (DS>2.75) [106]. The DS corresponds to the average number of acetate groups in each AGU unit of cellulose, originally provided with three hydroxyl groups. Substitution occurs preferentially on the C-6 and C-2 carbon, and the acetyl content has an important influence on CA properties, namely on the solubilization, polarity and reactivity with other chemical species [107]. For instance, cellulose triacetate does not dissolve easily in common solvents, unlike cellulose diacetate. Notwithstanding their different DS, both cellulose acetates are widely used industrially in plastics (e.g. food packaging as a rigid wrapping film and membranes) and textiles [42, 106].

Recently, several other synthesis methods, claimed as green approaches, have been developed for cellulose acetylation, which have yielded great results even when applied to cellulose extracted from agroinsdustrial residues, namely sugarcane straw, rice husk, corn stover and oil palm empty fruit bunch [27, 108–110]. These methods include, for instance, the use of ionic liquids, one step heterogeneous acetylation or solvent-free methods [111, 112]. Recently, Das et al. [111] reported a great acetylation yield by applying a solvent-free method catalyzed with iodine to esterify rice husk. The iodine has the advantage of being a cheap, convenient, commercially available and environment friendly reagent. It is worth mentioning that the assumed eco-friendly aspect related to those previously mentioned methods comes mainly from the substitution of conventional catalysts and reaction medium. However, environmental assessment studies in this scope are still limited.

In addition to the existing eco-friendly processing technologies to acetylate cellulose, cellulose acetate also stands out because of its biodegradability property. Although some studies have shown that the biodegradation rate can be decreased by increasing DS, this does not imply inhibition of the biodegradation process. Currently, CA is recognized as a biodegradable polymer, and several studies have confirmed its biodegradation in natural environments, under aerobic or anaerobic conditions, independent of the acetyl content [102, 113, 114].

Therefore, as a potential plastic component for packaging food items, CA is able to meet one of the most pursued demand in the scope of packaging technology by ensuring a balance among food protection, environmental pollution and sustainable solid waste disposal options [97, 100].

2.4. Cellulose-based packaging emerged from agroindustrial residues

Polymer-based packaging play an essential role throughout the food distribution and storage chain. In addition to ensuring food safety and quality, packaging must protect against chemical and microbial contaminants, moisture, dust, light and oxygen, in order to increase the food shelf life. When it comes to cellulose-based packaging, the most innovative products released in the market, to date, are still based on high quality cellulose derived from wood pulp or cotton. However, the use of agroindustrial residues as a renewable source for that application may offers favorable environmental and economic advantages, and therefore has increasingly been encouraged. Recent developments and applications on this field include the use of cellulose fibers, regenerated cellulose and nanocellulose, which have been successfully applied to produce food packaging materials (and food packaging components) with remarkable physicochemical properties. The use of cellulosic fibers extracted from agroindustrial residues have been extensively reported, but because of their non-thermoplastic property, they are commonly applied as fillers to produce composites [115–117].

Regenerated cellulose refers to a class of materials prepared directly via the dissolution of cellulose into a solution, followed by its shaping and regeneration process [75]. Recently, regenerated cellulose materials have drawn considerable attention mainly after the novel developments in the field of green techniques and solvents, which opened up new possibilities to manufacture biodegradable, thermal and chemically stable products by using simple and sustainable processing methods [75]. Regenerated films with mechanical and barrier properties comparable with those of conventional films were prepared by simply dissolution and regeneration of pretreated sugarcane bagasse [74, 76, 118], borassus fruit fibers [73], and oil palm empty fruit bunch [119]. Recently, the development of regenerated cellulose-based hydrogels and aerogels from agroindustrial residues as components in food packaging have been an attractive subject of research, mainly due to their liquid absorption capacity [120–122].

In recent years, nanostructured cellulose has attracted considerable research attention due its potential for maximizing the mechanical and barriers properties of bio-based packaging [123]. From literature, the nanocellulose (CNC) is generally applied as a reinforcement filler for food

packaging applications. Nanocellulose from some agroindustrial residues, such as pea hull [124], mango seed [125], oil palm empty fruit bunch [126, 127], wheat straw, rice straw and barley straw [128] have already been applied in bio-based packaging with remarkable mechanical and barrier properties. Another relatively new application in food packaging is the use of nanocellulose-based aerogels and hydrogels. Absorbent aerogels applied for food packaging were produced from rice husk, oat husk and cotton stalk-extracted cellulose nanocrystals [122, 129].

Despite the recent advances and potential applications above mentioned, some technical and economic limitations still make the conversion of agroindustrial residues into nanocellulose and regenerated cellulose a difficult task. The main challenges deal with the low yield of CNC production and high costs related to cellulose dissolution. A promising alternative to overcome these drawbacks lies on the derivatization of cellulose, especially into cellulose acetate, as previously mentioned. In addition to meet the desired properties to replace the petroleum-based plastics, CA has the complementary advantage to be biodegradable (and disposed of by biological means) and can be processed on conventional injection molding machines or extruders adapted to their specific processing properties [130, 131].

The processing of agroindustrial residues extracted cellulose into derivatives is a very recent research topic, with substantial increase of published papers being verified only after 2000s. An inexpensive and biodegradable thermoplastic was developed from acetylation of rice straw, and through casting technology transparent thin films were formed [132]. Daud et al. [109], by using oil palm empty fruit bunch as cellulose source, obtained an acetone soluble cellulose acetate film with mechanical properties higher than those of commercial CA. A fully bio-based and transparent all-cellulose composite film was fabricated by simply aqueous blending of water-soluble CA synthetized from waste cotton fabrics (WCFs) and nanocelluloses [133]. The same Cao et al. authors [133] converted high yields of CA from WCFs through a quasi-homogeneous and environmental-friendly process, that required a small amount of acidic ILs as catalyst. Recently, cellulose acetate emerged from agroindustrial residues was also qualified as a suitable material to produce active packaging with enhanced and additional properties [131, 134, 135].

2.5. Active packaging and additives

In recent years, the ever-growing global food trade, along with the consumers' demand for food safety and environmentally friendly products, have forced the research and development of innovative packaging with enhanced and multi-properties. Advances in engineering and materials

science have boosted the development of technologies towards this purpose [136], and the materials that falls under this scope are designated as active packaging. By definition, active packages are those intended to extend the shelf-life or to maintain or improve the condition of packaged food. This is accomplished via the embedment of components that would release or absorb substances into or from the packaged food or the environment surrounding the food [137]. Numerous active packaging systems with a multitude of functions have already been reported, namely absorbing/scavenging properties, realizing emitting properties, removing properties and microbial control [136, 138]. The main active agents responsible for conferring such functionalities include inorganic and organic materials/nanomaterials (e.g. metal/metal oxides nanoparticles, nanoclays, plat extracts, essential oils, phenolic compounds, chitosan and lysozyme) [136, 139].

Within the current explored agents, those with antimicrobial properties are one of the most studies components, since the growth of pathogenic and/or spoilage microorganisms are by far the major cause of food spoilage [136]. For this purpose, metal/metal oxides in their nanoscale form (including silver, gold, selenium, titanium dioxide, zinc oxide, copper oxide and magnesium oxide) have found to display effective responses. When incorporated in polymers, these nanoadditives can also improve their original mechanical and barrier properties [139–141].

Just recently more attention has been given to the effects of embedding metal/metal oxides nanoparticles on cellulose and cellulose acetate-based polymers [142–148]. This came about as the development of active packaging is also being shifted towards the use of bio-based and ecofriendly materials. Nevertheless, very few studies are reported in literature regarding the use of cellulose/cellulose acetate derived from agroindustrial residues to produce active packaging. For instance, cellulose acetate-water soluble was synthesized from waste cotton fabrics by Cao et al. [149] and used to produce nanohybrid films containing silver nanoparticles. In addition to be effective in inhibiting the growth of bacteria, the produced film also exhibits catalytic activity. In Panis et al. [150], banana peel was used to produce CA-incorporated silver particles. The film produced exhibited enhanced tensile strength and antibacterial activity against Staphylococcus aureus e Escherichia coli. More recently, Harini and Sukumar [151], by using banana residuesextracted cellulose, developed a transparent cellulose acetate film with bioactive properties, but instead of metal/metal oxides nanoparticles they used thymol (an oregano compound) as biocidal agent. The bioactive films presented great retention of the natural compound, which infused CA films with increased antioxidant and antimicrobial properties. Several other studies also incorporated natural compound as antimicrobial agents into cellulose-based polymers (not necessarily emerged from agroindustrial residues), namely thymol [131], murta fruit extract [152], essential oil from pink pepper [153] and carvacrol [154]. However, some drawbacks of incorporating these natural active compounds into polymers are their low chemical and thermal stability, as well as high volatility, which pose challenges to their processing [155].

In this perspective, inorganic nanoparticles, such as TiO₂ (TiO₂-NPs) and MgO (MgO-NPs), have attracted considerable interest as they resist to harsh processing conditions and presents superior photocatalytic property with biocidal effects against foodborne pathogens [156]. Besides, their use as food additives is regulated as safe by the U.S. Food and Drug Administration and European Food Safety Authority

TiO₂-NPs are among the most studied metal oxide NP to produce active packaging [139, 157]. In addition to the aforementioned characteristics, TiO_2 is chemically stable, cheap, durable and abundantly available [139, 156]. TiO_2 exist in nature as one three main polymorphs, namely anatase, rutile and brookite, which differ by their crystalline structure. The photocatalytic activity of TiO_2 nanoparticles is recognized in anatase and rutile, but they have been reported to exhibit antimicrobial activity only when exposed to UV light (generally in a wavelength range of 320-380 nm) [148, 156]. Upon exposure to light, reactive oxygen species (ROS), such as hydrogen peroxide (H_2O_2) and hyperoxide anions (O_2), are generated through redox reactions. ROS leads to severe oxidative stress and damages to cells' macromolecules, causing lipid oxidation, inhibition of enzymes, RNA/DNA damages and cell lysis [158]. The generated ROS and oxidative radicals can also cause the photocatalytic oxidation of ethylene when TiO_2 -NP are incorporated in food packaging [159]. Photodegradation of CA films can be also enhanced by the embedment of TiO_2 NP [160].

Currently, few studies have been devoted to the application of MgO-NPs as an active agent in bio-based packaging. The key advantages of MgO-NPs rely on their high thermal stability, low cost, non-toxicity and potent antimicrobial activity [161]. Their biocidal action is also resultant from the production of reactive oxygen species and electrochemical interactions, as reported by Gold et al. [158]. In addition to antibacterial efficacy, Swaroop and Shukla [140] also observed that the incorporation of MgO-NPs in PLA improved oxygen barrier and tensile strength of the film produced.

Indeed, nanotechnology has been widely applied for developing in the field of packaging systems as it can fulfill almost all the basic functions currently demanded. However, despite all beneficial properties introduced by the nanostructure particles, the most recent researches are targeted in pursuit alternative green methods of synthesis in substitution of the chemical-conventional ones, that usually undergo multi-step process, are time-consuming and use toxic reducing agents. In this

regard, the pulsed laser ablation in liquid medium emerges as a promising alternative since it is a flexible and fast method to produce nanoparticles and does not require any chemical or stabilizer.

2.6 Pulsed laser ablation of metal/metal oxides in liquids

Pulsed laser ablation in liquid medium is a relatively new method for producing nanoparticles, and has gained increasing interest because of its versatility, low cost, flexibility and easy execution. It is particularly interesting to integrate sustainable processing techniques since it meets the 12 principles of green chemistry (in briefly, the PLAL does not necessarily require chemicals or produce wastes) [162].

Concisely, PLAL consist of irradiating a pulsed laser beam onto a solid target immersed in a liquid medium. The mechanisms within the nanoparticles synthesis via PLAL can be distinguished into six main steps based on the temporal evolution of physical-chemical phenomena, as depicted in Figure 2.5 [162]. The process starts with the passage of the laser through the liquid medium. After going through the liquid, the laser reaches the bulk target, which absorbs energy through multiphoton absorption and direct photoionization. The laser photons of the laser couple with the electrons and phonons of the target and initiate the detachment of material. Detachment proceeds sustained by the immediate rise of the electron temperature, which leads to thermal process such as vaporization and phase explosion (material superheating up to the thermodynamic critical temperature) [162–164]. Simultaneously to the ablation process, the recoil pressure of the ablated material give rise to a shockwave, which heats the liquid and the target and may also promote the detachment of material [162]. As the ablated material is composed mainly by highly ionized atoms and electrons, it is considered a plasma plum. The constant absorption of photons by the plasma cause its expansion and subsequent cooling, with releasing of its energy to the solvent and target [163, 165]. During this process, nucleation and growth of nanomaterials take place, and reactions with solvent species may occur. Melted drops, with higher dimensions, can also be ejected from the target [162, 166]. The energy released from the plasma plum induces the formation of a cavitation bubble, that expands and then later collapse emitting a shockwave with possible effects over the NPs, namely aggregation, spreading or phase transition [162, 167].

Although this is a brief explanation of the nanoparticle formation process, there are still many events to be detailed that require a deep understanding of the physical and chemical processes involved in PLAL method.

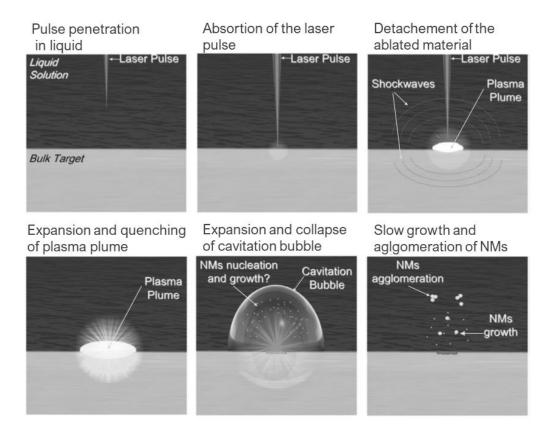


Figure 2-5. Schematic illustration of metal nanoparticles formation by PLAL. Modified from [162].

Two main processing conditions can affect the yield, composition, properties and characteristics of NPs produced by PLAL, whether material controlling and laser parameters. The bulk target and solvent composition are the main materials parameters. Several studies reported significative changes on NPs characteristic depending on the liquid medium. For instance, brucite-like Mg(OH)₂ particles were the primary crystalline product produced by laser ablation of Mg in deionized water, while MgO NPs were presented in higher percentage when acetone was used [168]. The same authors [168] also reported the generation of different morphologies and agglomeration patterns. Chaturvedi et al. [169] found that deionized water medium is favorable for the growth of anatase phase TiO₂ nanoparticles, while surfactant added water medium is favorable for rutile phase growth. Moura et al. [171] reported the production of smaller silver NPs when ablation was carried out in deionized water instead of ethanol. The multitude of solvents suitable to be applied in PLAL opens up a range of possible applications, whether as a colloidal solution or NP powder after centrifugation.

The main laser/setup parameters influencing the NPs production and properties during PLAL are laser fluence, repetition rate, wavelength, laser energy, pulse duration and liquid depth. In

general, ablation efficiency increases with wavelength, laser fluence and pulse energy [163, 170]. Moreover, smaller metal nanoparticles are obtained by using higher laser fluence values [171].

2.7 Life cycle assessment

Life cycle assessment (LCA) is an effective tool for understanding the environmental impacts associated with the development of products, processes or services. Its application can encompass all phases of a product's life, from resource extraction to final disposal, or only part of its manufacturing process.

The search for the establishment of a solid and standardized scientific base related to the application of LCA led the International Organization for Standardization (ISO) to prepare documents containing the fundamental guidelines, principles and framework for conducting this type of study, including [172, 173]: (i) definition of the goal and scope; (ii) life cycle inventory analysis (LCI); (iii) life cycle impact analysis (LCIA); and (iv) interpretation of results.

The definition of the goals and scope of the LCA study provides, among other issues, the reasons for executing the LCA, definition of functional unit (FU), the product system and system boundaries, as well as the environmental impacts categories and methodologies considered [174, 175]. The system boundaries define which unit process included in the product system will take part of the LCA study, and may extend to different stages of a product's life cycle. In the literature there are mainly projects that encompass activities from the product generation phase until the previous stages of use (cradle-to-gate), or until the end of its useful life (cradle-to-grave). Regarding the functional unit, it provides a reference to which the inputs and outputs of the product system are related. Its choice should take into consideration the type of study performed and its objectives [176]. The functional unit may be expressed, for example, in units of mass, volume, or in terms of units produced by time interval [177, 178].

The inventory analysis aims to identify and quantify the inputs (resources) and outputs (waste and emissions) of the system. Therefore, the limits of the system and subsystems (or processes) involved in the object of study must be carefully defined [174]. Once the inventory analysis is established, sufficient information is obtained to assess the contributions of resource extraction and waste emissions to potential environmental impact categories, which falls under the life cycle impact assessment [174]. As standard, the LCIA is divided into mandatory (classification and characterization) and optional (normalization, weighting, grouping and sensitivity analysis) steps (ISO 14040, 2006). In the mandatory steps, emissions from life cycle Inventory (LCI) results are

assigned to impact categories according to their potential impacts, and characterization factors are used to quantify their impact contributions. Regarding the optional steps, their application depends on the object of study and relevance of possible outcomes. In general, normalization and sensitivity analysis are usually performed, while weighting and grouping are rarely implemented since they require a subjective interpretation [179]. Through normalization, impact potentials are expressed relative to a reference situation and in a common scale, which enables comparisons across impact categories. Sensitivity analysis is highly dependent on the LCA model structure. By performing it, is possible to identify how variations on data, parameters and processing can influence the output results [179–182]. Sensitivity analysis can be performed as perturbation analysis (effect of a single parameter) or scenario analysis (effect of combination of parameters and process in a specific model layout).

There is no guideline regarding the choice of impact methodologies and categories. Most of the existing life cycle impact assessment methods (ReCiPe, Usetox, culmulative energy demand, CML, Eco-Indicator 99, TRACI 2.1 and Ecological scarcity method) can be applied for midpoint and endpoint level. Endpoint indicators are defined at the level of the areas of protection (e.g. human health and natural resources), and midpoint indicators indicate impacts somewhere between the emission and the endpoint. Naturally, on midpoint level a higher number of impact categories is differentiated and the results are more accurate and precise compared to the endpoint [183]. Consequently, most of reported studies on LCA are modeled at the midpoint level to limit uncertainties. There are many impact categories on the level of midpoint studies, and their selection must be primarily based on the purpose and type of application of the LCA [184]. The general recommendation regarding the choice of impact categories is that the obtained results must be robust enough to form a basis for further considerations or decisions. Also, the study must include impact categories for which international consensus has been reached (e.g. global warming potential, stratospheric ozone depletion, photochemical oxidant formation, acidification and resource consumption) [182]. Finally, the interpretation, significance and validity of the results are conducted. These comprise fundamental procedures as they make it possible to identify the major environmental burdens involved in the life cycle of products, processes or services.

Currently, a variety of LCA software-specific database and tools are available (e.g. GaBi, SimaPro, OpenLCA and EASETECH). These can be applied for single product evaluation or comparative studies as a means of determining environmentally preferable alternatives. Due mainly to the current expansion of the biopolymer market and the constant development of production

routes, several studies have discussed, through the application of LCA, the advantages and disadvantages of these biopolymers compared to conventional plastics. However, in addition to this application, LCA studies can be used as an important tool for identifying more sustainable production and disposal methodologies, as well as for optimizing production processes.

Recent review studies on LCA have shown that the prospects for advancement in the field of biopolymers depend mainly on the development of new materials, the application of less polluting production techniques, the use of renewable energy, and the adoption of sustainable disposal methods [179, 185–188]. Besides, the use of second-generation raw materials is mentioned in order to avoid conflicts with food production. However, the most commonly studied biopolymers via LCA are still PLA, PHA and thermoplastic starch (TPS), which are synthesized from first-generation inputs, such as corn, soybeans, sugarcane. and cassava [178, 185, 189, 190].

Overall, the most suitable disposal technologies for minimizing impacts caused by solid waste seems to be composting and recycling [185, 187, 191]. However, for the vast majority of countries, a management system that incorporates such disposal technologies still represents a utopian scenario. In many underdeveloped countries efforts still need to be made to eliminate open dumps. From this perspective, the development of biodegradable plastics seems to be the most promising alternative since, theoretically, their persistence in the natural environment is lower, and their impacts as well. Nevertheless, technically, this is not the right way to tackle this issue, as biodegradability must not be an excuse to litter. Indeed, within the circular economy approach, biodegradable plastics cannot be a sustainable solution without the infrastructure required to manufacture, collect and recycle them.

In addition to the challenges imposed by the country-specific waste management practices, several studies are focused on the development of products with applications not clearly defined, which makes decision on post-use management options a difficult task. In order to reduce model uncertainties, a coherent approach is based on the adoption of gate-to-gate perspective, with impact analysis focused on the product's processing steps [192, 193]. Besides, when comparing product systems with similar properties and manufacturing processes, common pats of the LCA may be disregarded if consistently justified.

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Chapter 3. Availability and suitability of agroindustrial residues as feedstock for cellulose-based materials: Brazil case study

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Abstract

Bio-based polymers have emerged as a feasible alternative to petrochemical polymers mainly due to their biodegradability and renewable feedstock. Brazil is considered one of the largest producers of agricultural commodities. Hence, the country is also distinguished by the large generation of this residue type, which can be potentially used as a source to obtain biopolymers, such as cellulose. Based on the Brazillian agriculture market, the study aims to analyze the suitability of agroindustrial residues as raw material for cellulose-based materials. A methodology for the selection of the most suitable residues is proposed, which takes into account the chemical composition of residues, namely the cellulose content and the cellulose-to-lignin ratio, as well as their availability. In order to meet conservation issues, the availability of residues is calculated as a function of sustainable removal rates and competitive uses. Taking as reference the main crops identified, the average amount of agroindustrial residues available in Brazil was estimated at 108 million tons/year. Among the most suitable residues to be used as cellulose feedstock are soybean straw, sugarcane top/leaves, maize husk and stover and sugarcane bagasse.

Keywords: Cellulose, lignocellulosic biomass, bio-based materials, availability of residues.

3.1. Introduction

In the last decades, the industrial segment of polymeric products has shaped the humankind life. Increasing demand for versatile products for several applications has been a major factor responsible for the growth of this segment. Nowadays, polymers are found throughout many sectors and industries including packaging, build and construction, automotive, electric and electronics, agriculture and consumer goods [1].

Currently, their improper disposal, persistence and fossil-based raw material represent factors with negative environmental and economic effects, namely greenhouse gas emission and land and marine pollution [2, 3]. In terms of disposal for example, only in Brazil, plastic represents around 13.5% of the total solid waste collectible [4]. Nonetheless, in 2015, from 72.5 million tons of solid waste collected, almost 30 million tons were sent to unsuitable destinations such as dumps [4]. Depending on the kind of plastic, it will take a very long time to degrade when released into the environment, enhancing its accumulation [5]. Even in the controlled environment of landfills plastics hardly degrade [6]. In the marine environment, common plastics used in packaging

degrade at a very slow rate [2, 7]. According to Eriksen et al. [8], a minimum of 5.25 trillion plastic particles weighing 268,940 tons is estimated to be floating in the world's oceans.

In this sense, the search for renewable raw materials suitable for the production of biodegradable polymeric products is imperative, especially considering the depletion of fossil resources, fluctuation of oil price and the negative environmental impacts. Indeed, bioplastics have been mentioned as a lead market by the European Commission, with global production capacity set to grow 350% by 2019 [1]. In recent years, the group of bioplastics derived from renewable and biodegradable resources has received great interest. They can be manufactured by different techniques using biopolymers as feedstock, such as cellulose, lignin and starch.

Among the available sources of renewable feedstock, lignocellulosic biomass stands out in the global scenario. Currently, this category of biomass assumes relevant importance for the generation of renewable energy, fuels and several other products (dispersants, flocculants, textile fibers and activated carbon), mainly due to its chemical constitution. Lignocellulosic biomass consists mainly of three natural organic polymers - cellulose, hemicellulose and lignin - and small amounts of proteins, pectin and extractives [9, 10]. Among the aforementioned components, cellulose is the most abundant organic polymer on earth, and owing to its mechanical and thermal properties, renewability, widely availability, non-toxic, low cost, biodegradability and derivatizability, cellulose is considered a promising feedstock for the development of sustainable materials [11–14].

Cellulose is a high molecular weight homopolymer assembled from the repetition of cellobiose units (β 1-4 glucosidic covalent bonds between D-glucopyranose subunits) [15, 16]. Each cellobiose unit has hydroxyl groups that promote strong interactions by intra and intermolecular hydrogen bonds. This spatial structure allows the molecules to crystallize in a horizontal plane and in parallel chains, forming microfiber packages [17]. In the plant cell walls, the cellulose is embedded in a matrix of hemicellulose and lignin, bound together by covalent cross-links, which results in the recalcitrant property of the lignocellulosic biomass. Hence, the performance of physical or chemical pretreatments aiming the fractionation of lignocellulosic biomass is essential to convert cellulose into polymeric materials [10, 18]. After extraction, cellulose can be processed in the form of nanocellulose, derivatized cellulose, composites fillers or matrix and regenerated materials, such as films and gels [13, 19].

In general, lignocellulosic biomass may result from agricultural waste, forest waste, energy crops and municipal and industrial wastes [10, 20]. The set of activities associated with the

vegetable agribusiness sector stands out as one of the main residues generators. Framed worldwide as one of the leading agricultural producers, Brazil also excels due to the large waste generation originated from this type of activity. It is estimated that the production of agricultural residues in Brazil exceeds 200 million tons per year [21, 22]. Despite these residues being already applied to a range of valorization processing, a large portion is still directed to sanitarium landfills and incineration plants, or burned in the harvest field.

The first step in the sense of analyze the feasibility of agroindustrial residues valorization, regarding its application to produce cellulose-based products, consists in the identification of residues types that besides to have high content of cellulose, are also generated and available in significant quantities. Therefore, in this study, the required estimates were carried out based on the agroindustrial market of Brazil. The availability of residues was calculated as a function of the average crop production and conservative assumptions, such as sustainable removal rates and competitive uses. Then, the suitability of each residue for cellulose-based materials production was evaluated taking into consideration their availability and chemical composition.

3.2. Methodology

3.2.1. Crop Residues Production and Chemical Composition

Spatiotemporal characterization of agricultural crops production was performed with statistical data of Brazilian Institute of Geography and Statistics (IBGE) [23], extending from 2004 to 2014. In order to complement this previous assessment and support the selection of main crops, information about residues types and their respective cellulose, hemicellulose and lignin contents were obtained from associated literature.

The crop residues production can be calculated in terms of the average annual production of crops (AAP) and the residue generation rate (RGR). For the crops turned to processing industries, AAP corresponds to the portion of production intended for this application. For those which do not present this characteristic, the AAP values resulted directly from IBGE database. RGR values were obtained directly as default values from Natural Resource Module (NRM), as well as from other published documents. The NRM consist of a data processing and handling digital module, applied to assess the availability of bioenergy feedstock originating from crop production, agricultural residues and forestry [24].

3.2.2. Residue Generation Rate

The RGR can be understood as the ratio, on a dry basis, of the weight of residue produced to the total weight of the main crop product (e.g. the ratio of straw to wheat grain). RGR values may vary temporally and spatially according to several aspects, depending on the cultivation techniques, soil fertility and weather conditions [22]. Considering that RGR data used come from other studies targeted to different geographic areas, including those belonging to Natural Resource Module database, it is expected that these values encompass a technical, environmental and cultural variability, so that their application meet only a rough estimate. The diversity and variability of residues generation rates can be seen in Table 3.1.

Table 3.1. Residues generation rates of agroindustrial residues.

| Crop | Residue | Residues Generation Rate | | | | |
|-----------|--------------|--------------------------|-----------|-----------|----------|-----------|
| Cotton | Hull | 0.26 [24] | 0 0E [00] | | | |
| | Stalk | 3.4 [24] | 2.95 [22] | 3.0 [25] | | |
| D: | Husk | 0.25 [24] | 1 40 [22] | | 1.0 [26] | 0.26 [27] |
| Rice | Straw | 1.33 [24] | 1.49 [22] | 1.0 [25] | | 1.55 [27] |
| Banana | Tree pruning | 0.8 [28] | | | | |
| Potato | Skin | 0.04 [29, 30] | | | | |
| Cocoa | Pod | 1.5 [24] | | | | |
| Coffee | Husk | 1.32 [24] | | | | |
| Cugaraana | Top/leaves | 0.20 [24] | 0.22 [22] | 0.1 [25] | | |
| Sugarcane | Bagasse | 0.26 [24] | | | | |
| Coconut | Husk | 0.49 [24] | | | | |
| Coconut | Shell | 0.39 [24] | | | | |
| Beam | Straw | 1.7 [25] | | | | |
| Apple | Pomace | 0.30 [31] | | | | |
| Cassava | Stalk | 0.13 [24] | 0.20 [22] | | | |
| | Cob | 0.30 [24] | 1.42 [22] | | 0.7 [26] | 1.96 [27] |
| Maize | Husk | 0.22 [24] | | | | |
| | Stover | 1.96 [24] | | 2.0 [25] | | |
| Coulbaana | Straw | 1.53 [24] | 2.05 [22] | 2.12 [27] | | |
| Soybeam | Pod | 1.09 [24] | | | | |
| Tomato | Pomace | 0.01 [32, 33] | | | | |
| Wheat | Straw | 1.28 [24] | 1.42 [22] | 1.1 [25] | 1.0 [26] | 1.5 [27] |
| Mango | Seed | 0.07 [34, 35] | | | | |
| Pineapple | Peel | 0.12 [34, 36] | | | | |
| | | | | | | |

3.2.3. Availability of Agricultural Residues

After identifying the main crops produced in Brazil, the availability of crop residues (AR_{cR}) (Equation1) was calculated as a function of the average annual production of crops (AAP), residues generation rate (RGR), sustainable removal rate (SSR) and other competitive uses, while the availability of industry-driven crops residues (AR_{loc}) (Equation 2) took into consideration the AAP, the portion of production turned to industrial processing (IP), RGR and other competitive uses (CU).

$$AR_{CR} = (AAP)(RGR)(SRR)(1 - CU)$$
(1)

$$AR_{IDC} = (AAP)(IP)(RGR)(1 - CU)$$
(2)

3.2.3.1 Sustainable Removal Rate

Sustainable removal rates (SRR) of crop residues are required to meet environmental and economic sustainability associated with the agricultural activity. The main objectives of sustainable crop residues removal consist in maintaining the natural soil fertility via incorporation of nutrients, promoting stability of upper soil layers and increase the organic matter content [37]. The percentage of crop residues that must be kept in the field depends on factors, such as the soil structure and type, planting techniques (e.g. fertilizers usage, crop rotation and till or no-till farming) and conservation practices.

In the literature is commonly suggested that a residue removal up to 30% does not imply damage to the soil [38–41]. However, estimates for sustainable residue removal may vary greatly from crop to crop, and it depends on another series of factors, such as geographical, climatic and technical/technological. For example, according to SoCo project team [42], regardless the cultivation method, it is indicated a maximum removal rate of straw around 70 percent. Karkee et al. [43] found that depending on soil characteristics, topography and farming practices (tillage, conventional and no-till), the percentage of available biomass to be removed may vary from 0% to 98%, without negative effects on the soil. In this study we adopted the sustainable removal rates of 40% for wheat straw, maize residues, rice straw and husk, bean straw and cotton stalk [44], and 30% for sugarcane top/leaves [38], cassava stalk and soybean straw and pods [44].

3.2.3.2 Competitive Use of Residues

Besides SRR, agroindustrial residues may be used in a range of applications, including energetic valorization, animal feed and bedding, application in the construction sector, as well as starting materials for activated carbon production [42, 45–47]. These applications represent competitive use with direct influence over the availability of residues and must be taken into account.

Generally, very little information is available on industrial processing residues used for recycling/reuse purposes. Notwithstanding, in order to proceed with conservative estimates, for those crops with no information on the percentage of residues turned to other valorization routes, it is assumed that only 50% is available. For residues that no other valorization route was identified, such as cassava stalk, rice straw, cotton stalk, soybean straw and sugarcane top/leaves, is estimated that 75% of the total generated is available (Table 3.2). It was also assumed that 35% of coconut [48], 10% of mango [49, 50], 40% of tomato [51], 10% of potato [52] and 15% of apple national production [53] are allocated to industries processing. Besides, it is verified that practically all the production of cocoa and coffee goes to processing industries [54, 55].

Table 3.2. Competitive uses of agroresidues.

| Residue | Availability (%) | Competitive uses | Reference |
|----------------------------|------------------|---|-----------|
| Sugarcane bagasse | 10 | Fired in stem boilers in the own mill to produce electricity. | [56, 57] |
| Sugarcane top/leaves | 75 | Not identified. | _ |
| Maize husk, cob and stover | 40 | Animal feed. | [56] |
| Rice husk | 25 | Drying, power generation at rice mills and chicken bedding production. | [56] |
| Rice straw | 75 | Usually burned in the harvest field. | [57] |
| Bean straw | 50 | Used to produce activated carbon and as a substrate for the production of other crops. | [58, 59] |
| Cassava stalk | 75 | Used as animal feed, but its toxic principle limits this kind of application. | [57, 60] |
| Orange residues | 0 | Aromatizing and animal feed. | [61] |
| Grape residues | 0 | Used in food, pharmaceutical and cosmetics industries. | [21, 62] |
| Coconut residues | 90 | Biomass, composites, agricultural fertilizer, activated carbon and filler for automotive banks. | [63–65] |
| Mango stone and husk | 50 | Animal feed. | [66, 67] |
| Tomato pomace | 50 | Fractionation of components, carotenoids extraction and biofuels. | [68–70] |
| Potato skin | 50 | Cattle feed. | [71] |
| Appla pomace | 50 | Organic fertilizer and animal feed. | [72] |
| Cocoa shell | 50 | Energetic valorization. | [73] |
| Coffee husk | 50 | Animal bedding. | [56] |
| Cotton stalk | 75 | Not identified. | |
| Soybean straw | 75 | Not identified. | |

3.2.4 Suitability of Agroindustrial Residues to Develop Cellulose-based Products

The assessment of residues suitability has been carried out based on methodology reported by Araújo et al. [74], where the suitability scale of residues is measured (equation 3) taking into account the normalized values of three main parameters (χ_{ρ}): Availability of residues (AR), cellulose content (CC) and cellulose to lignin ratio ($R_{c,\iota}$). The presence of lignin in lignocellulosic biomass can be considered as one of the major obstacles in biomass pretreatment processes [75, 76]. Therefore, the influence of lignin over the selection of residues was accounted by including the cellulose-to-lignin ratio. The greater the ratio, the greater is the cellulose percentage compared to the lignin content, and thus more efficient is the biomass treatment.

$$S'_{ij} = \sum_{i}^{n} w_i \times \chi'_{ij}; \quad \text{Lignin content} > 0$$
 (3)

Where,

$$\chi'_{ij} = a + \left[\frac{\left(\chi_j - \chi_{min}\right) (b - a)}{\left(\chi_{max} - \chi_{min}\right)} \right]; \tag{4}$$

The constants a and b In equation 4 correspond to arbitrary points, equivalent to 0.1 and 1, respectively, used to restrict the range of suitability values. Besides, w_i correspond to weights of parameters AR, CC and $R_{c/L}$, which are equivalent to 35, 45 and 20, respectively, and the subscript index j represents the residue under evaluation. In the same equations, the parameters designated by maximum and minimum subscripts are related to residues that present the highest and lowest values, respectively, of the parameter concerned. Thus, for each residue, the suitability index (S_{ij}) may vary from 10 (lowest suitability) to 100 (highest suitability), expressed as dimensionless (Equation 3).

3.3. Results and discussion

3.3.1. Agricultural Production and Harvested Area

The crop production can be divided into two categories: (i) permanent and (ii) temporary. The permanent category includes the long-term crops that do not require replanting after harvesting, while the temporary ones generally require replanting after harvesting. Within the period from 2004 to 2014, among all permanent crops, excel the high production of orange, banana, coffee and coconut (Figure 3.1). In fact, Brazil is among the world's leading producers of these crops. Only the Southeast region is responsible for more than 80% of national orange production and 55% of

the total permanent crops production. Northeast region accounts the largest production of coconut and banana and 23% of permanent crops production (Figure 3.1a and 3.1b).

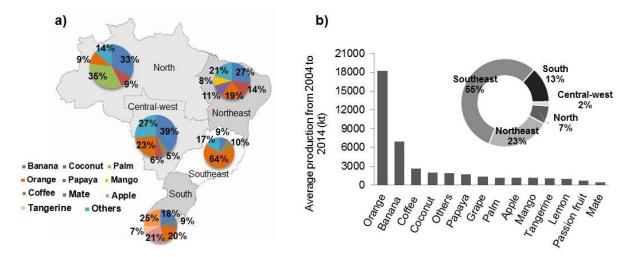


Figure 3-1. Average production of permanent crops in Brazil. a) Average percentage of the main crops produced by each macroregion. b) Average production of permanent crops from 2004 to 2014 and average percentage of permanent crop production by macroregions.

Southeast and Northeast regions accounted about 5,224,000 ha (43.43% and 39.68%, respectively), on average, of the total area used for the cultivation of permanent crops in Brazil. Around 80% of coffee plantation and 74% of orange plantation are located in the Southeast region. In the Northeast predominates plantation areas for banana, cocoa, cashew nut, coconut and sisal. In other regions, crops with lower percentage of area for cultivation prevails.

Regarding the temporary crops, over the last 11 years of data, it was possible to verify that the production almost doubled, reaching 972.2 megatons in 2014. Among all the temporary crops, sugarcane production is by far the most relevant. In 2014 were produced 737.2 Mt of sugarcane, 86.7 Mt of soybean, and 79.8 Mt of maize (Figure 3.2). These figures place Brazil among the world's largest producers of these crops. The other crops, which account for a total of 28 different types, contributed with 68.5 Mt (about 7% of the total production) (Figure 3.2). About 52% of the national temporary crops production takes place in the Southeast region. The Centre-west region presents the second higher percentage contribution, mainly due to its leadership as maize and soybean producer.

It could also be noticed that 43% of the total harvesting area were assigned to soybean plantation, 22.5% were related to maize and 14.8% to sugarcane. It is worth highlighting that, in

2014, two Brazilians regions accounted for about 45% of all harvested areas, namely Centre-west (with 24.3 million hectares) and South (with 21 million hectares).

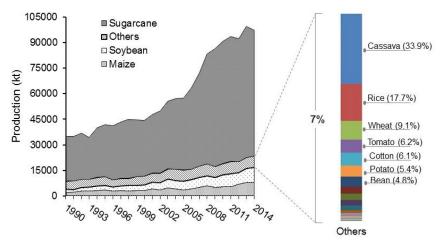


Figure 3-2. Annual evolution of temporary agricultural crops production from 1990 to 2014.

3.3.2. Chemical Composition of Agroindustrial Residues

Depending on the morphological structure of crops and the processing levels to which they are subject, different types of residues with specific cellulose, hemicellulose and lignin contents can be obtained (Tables 3.3 and 3.4). Cereals and oilseed crops may be mentioned as crops capable of generating wastes still in the harvesting phase. On the other hand, crops such as grape, orange, apple, potato, coconut, coffee, mango and cocoa, particularly interesting to obtain processed products, may result in the generation of wastes at different stages of industrial processing. Tables 3.3 and 3.4 present cellulose, hemicellulose and lignin values for a range of lignocellulosic residues derived from temporary and permanent agricultural crops, respectively. These tables highlight several agroindustrial residues with high cellulose content and potentially interesting to be used as sustainable resources to the production of cellulose-based materials.

Table 3.3. Chemical composition of lignocellulosic biomass derived from temporary crops

| Bio | omass | Cellulose (%) | Hemicellulose (%) | Lignin (%) | Reference |
|-------------|------------|---------------|-------------------|------------|-----------|
| Pineapple | Peel | 21.98 | 74.96 | 2.68 | [77] |
| Cotton | Hull | 44.35 | 11.2 | 16.15 | [78] |
| | Stalk | 40.4 | _ | 20.9 | [79] |
| Garlic | Skin | 41.7 ± 2.1 | 20.8 ± 1.6 | 34.5 ± 2.4 | [80] |
| Peanut | shell | 49.4 ± 1.4 | 8.1 ± 0.6 | 33.1 ± 1.5 | [81] |
| Rice | Straw | 39.5 - 41 | 24.4 - 31 | 15.9 - 24 | [82, 83] |
| THEC | Husk | 35 | 33 | 23 | [84] |
| Oat | Stalk | 31 - 48 | 27 - 38 | 16 - 19 | [85] |
| Potato | Skin | 10.5 | - | 4.0 | [86] |
| Cugaraana | Top/leaves | 39.8 | 28.6 | 22.5 | [87] |
| Sugarcane | Bagasse | 43.10 | 22.82 | 24.09 | [12] |
| Onion | Skin | 41.1 ± 1.1 | 16.2 ± 0.6 | 38.9 ± 1.3 | [88] |
| Rye (stalk) | Stalk | 33 - 50 | 27 - 30 | 16 - 19 | [85] |
| Barley | Straw | 33 - 40 | 20 - 35 | 8 - 17 | [10] |
| Pea | Hull | 62.3 | 8.2 | _ | [89] |
| Bean | Straw | 40.2 | 19.32 | 18.13 | [90] |
| Sunflower | Straw | 40.41 | 31.44 | 19.45 | [90] |
| Jute | Stem | 61 – 71.5 | 13.6 – 20.4 | 12 - 13 | [91] |
| Flax | Stem | 75.4 ± 0.2 | 13.4 ± 2.8 | 3.4 ± 0.9 | [92] |
| Cassava | Stalk | 38.8 | 7.2 | 11.8 | [93] |
| | Husk | 62.07 ± 0.86 | 17.93 ± 0.86 | 14.6 ± 0.6 | [94] |
| Maize | Stover | 40.8 | 34 | 22 | [83] |
| | Cob | 31.2 - 45 | 35 – 43.1 | 15 – 16.5 | [10, 95] |
| Ramie | Fiber | 69 - 91 | 5 - 17 | < 1 | [96, 97] |
| Soybean | Straw | 44 - 83 | 24.3 ± 3.0 | 5.0 - 14 | [98] |
| Sorghum | Stalk | 41.7 | 23 | 18.2 | [99] |
| Tomato | Plant | 39.1 | 28.8 | 12.1 | [100] |
| | Pomace | 29.1 | 13.5 | 57.4 | [101] |
| Wheat | Straw | 30 - 40.8 | 38.32 - 50 | 15 – 22.45 | [10, 90] |
| Triticale | Straw | 32.20 | | 15.02 | [102] |

 $\label{thm:composition} \mbox{Table 3.4. Chemical composition of lignocellulosic biomass derived from permanent crops.}$

| į. | Biomass | Cellulose (%) | Hemicellulose (%) | Lignin (%) | Reference |
|----------|-----------------------|---------------|-------------------|---------------|------------|
| Olive | Pomace | 24.1 – 14.45 | 11.0 - 6.63 | 14.1 - 8.54 | [103] |
| | Raquis | 48.7 | 16.1 | 12.2 | [104] |
| Banana | Pseudostem | 48.19 – 59.22 | 12.09 – 15.91 | 14.39 – 21.56 | [105] |
| | Peduncle | 48.31 – 60.41 | 10.20 – 13.99 | 17.56 – 20.66 | [105] |
| Cocoa | Shell | 35.4 | 14.0 | 37 | [73] |
| Coffee | Husk | 43.0 ± 8.0 | 7.0 ± 3.0 | 9.0 ± 1.6 | [106] |
| Coconut | Husk | 37.0 | - | 32.5 | [107] |
| Orange | Pell | 14.4 – 37.08 | 10.9 – 11.04 | 1.33 – 7.52 | [108] |
| Lemon | Pell | 23.06 ± 2.11 | 8.09 ± 0.81 | 7.56 ± 0.54 | [108] |
| Apple | Pomace | 7.2 | _ | 23.5 | [109] |
| Mango | Seed | 55.0 ± 1.0 | 20.6 ± 0.3 | 23.85 ± 0.21 | [110] |
| Nut | Shell | 53.9 | 15.4 | 30.7 | [101] |
| Oil palm | Empty fruit brunch | 40 - 50 | 20 - 30 | 20 - 30 | [111] |
| Pear | Pomace | 32.9 ± 0.3 | 22.1 ± 0.2 | 17.0 ± 0.4 | [112] |
| Sisal | Fiber | 64.9 - 78 | 10 - 25.4 | 8 - 11.7 | [113, 114] |
| Grape | Pomace | 27.9 | 9.1 | 63.0 | [101] |

3.3.3. Availability of Agroindustrial Residues

The availability of residues related to main crops identified in previous assessment took into consideration: (i) average annual production of crops; (ii) residue generation rate; (iii) sustainable removal rates; and (iv) other competitive uses. Table 3.6 shows the assumed values for variables encompassed in equations 1 and 2, as well as the annual residue production (RP) and the availability of residues. Exclusively for the industry-driven crops, the AAP is equivalent to the portion of production directed to industrial processing. Residues classified in this approach are highlighted in table 3.5.

Taking into account only the main residues identified, the total amount of agroindustrial residues produced in Brazil is estimated at over 589 million tons/year on average. However, in response to the conservation assumptions, the availability of residues is estimated at about 108 million tons/year (Table 3.5). The analysis of these results along with the data provided in Table 3.2 and 3.3, confirms the potential application of the accounted residues as cellulose feedstock.

Table 3.5. Average availability of agricultural residues. AAP – average annual production of crops; RGR – residue generation rate; RP – annual residue production; SRR – sustainable removal rate; AR – availability of residues; CU – competitive uses; * production intended to industry processing.

| Residues | AAP (kt/year) | RGR (t _{residuo} /t _{crop}) | RP (kt/year) | SRR (%) | CU (%) | AR (kt/year) |
|------------|---------------|--|--------------|---------|--------|--------------|
| Cocoa | 227.83 | (reside) | , , , , , | . , | ` ' | , , , , , |
| Shell | 227.03 | 1.5 | 314.75 | | 50 | 170.87 |
| Coconut | *694.86 | | | | | |
| Husk | 054.00 | 0.49 | 340.48 | _ | 10 | 306.43 |
| Shell | | 0.39 | 270.99 | _ | 10 | 243.90 |
| Coffee | 2,643.6 | | | | | |
| Husk | , | 1.32 | 3,489.60 | | 50 | 1,744.8 |
| Banana | 6,923.9 | | | | | |
| Pruning | | 0.8 | 5,539.12 | | 50 | 2,769.6 |
| Mango | *115.50 | | | | | |
| Seed | | 0.07 | 8.09 | | 50 | 4.05 |
| Apple | *173.5 | | | | | |
| Pomace | | 0.30 | 52.05 | _ | 50 | 26.03 |
| Maize | 56,688.7 | | | | | |
| Cob | 30,000.7 | 0.33 | 18,707.3 | 40 | 60 | 2,993.2 |
| Husk | | 0.22 | 12,471.5 | 40 | 60 | 1,995.4 |
| Stover | | 1.96 | 111,109.9 | 40 | 60 | 17,777.6 |
| Soybean | 64,194.4 | | | | | |
| Straw | | 1.53 | 98,217.4 | 30 | 25 | 22,098.9 |
| Sorghum | 1,861.00 | | | | | |
| Stalk | | 1.4 | 2,605.4 | 30 | 50 | 390.80 |
| Sugarcane | 625,452.7 | | | | | |
| Top/leaves | | 0.20 | 125,090.5 | 30 | 25 | 28,145.4 |
| Bagasse | | 0.26 | 162,617.7 | | 90 | 16,261.8 |
| Cassava | 24,743.0 | | | | | |
| Stalk | | 0.13 | 3,216.6 | 30 | 25 | 723.73 |
| Wheat | 5,130.8 | | | | | |
| Straw | | 1.28 | 6,567.5 | 40 | 50 | 1,313.5 |
| Rice | 12,181.0 | | | | | |
| Husk | | 0.25 | 3,045.2 | | 75 | 761.3 |
| Straw | | 1.33 | 16,200.7 | 40 | 25 | 4,860.2 |
| Bean | 3,194.5 | | | | | |
| Straw | | 1.7 | 5,430.7 | 40 | 50 | 1,086.1 |
| Cotton | 3,818.0 | | | | | |
| Hull | | 0.26 | 992.7 | | 50 | 496.3 |
| Stalk | | 3.4 | 12.981.4 | 40 | 25 | 3,894.4 |
| Pineapple | *648.9 | | | | | |
| Peel | | 0.12 | 77.9 | _ | 50 | 38.9 |
| Tomato | *1,265.4 | | | | | |
| Pomace | | 0.01 | 12.7 | | 50 | 6.3 |
| Potato | *349.5 | | | | | |
| Skin | | 0.04 | 13.98 | _ | 50 | 6.99 |

Figure 3.3 summarizes the information discussed so far about the availability of residues and the average percentage of cellulose. Evaluating only these two parameters, it can be observed the

greater relevance of sugarcane top/leaves, soybean straw, maize stover, sugarcane bagasse and rice straw, which accounted for about 82% of the total production of available residues. The remaining residues, despite having higher percentages of cellulose, present less significant AR values when compared to the others. It is worth highlight the relevance of sugarcane crop, which contributes alone with 41% of the total residues.

By comparison, the estimates obtained in this study resemble others identified in literature, especially when taken into consideration the total production of residues and not only the portion available [22, 44, 60]. For instance, Forster-Carneiro et al. [22] also highlight the sugarcane as the crop with the largest production of residues (estimated at 157 million tons), and stressed a possible increase in the generation of agroindustrial residues around 25.2% by 2020.

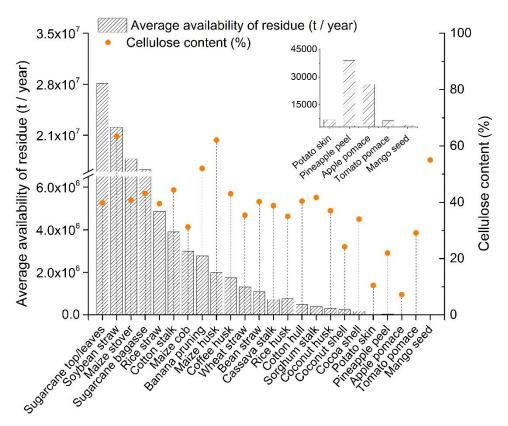


Figure 3-3. Availability and cellulose content of the major agroindustrial residues identified.

Assessing the annual variability of AR for the whole historical series (Figure 3.4) it is possible to notice a marked increase of residues generation throughout the years. For instance, from 2005 to 2014 the availability of residues increased 40%. It is estimated that 134 million tons of residues were available in Brazil in 2014. These estimates are consistent with current developments in the global agricultural sector. Only in the last three decades the Brazilian agricultural production more

than doubled in volume. The sector has contributed significantly to the country's trade balance, mainly as a foreign currency collector due to export activity. Driven by continuous productivity improvements, and the implementation of stimulus and investment policies in research and development as well, the outlook for 2024 is that the sector continues to grow [115].

In line with the expected increase of agricultural productivity is the large generation of residues (Figure 3.4), which still have marginal economic use [22]. Taking as reference the average availability of residues from 2004 to 2014, and the years in which have been verified the lowest and highest residues generation, positive and negative variations ranging from -25.7 and +24.04% were found. Moreover, it is possible to predict an increase of agroindustrial residues generation, considering that, since 2005, the availability of residues has shown positive growth rate, with average annual fluctuation of about + 5.43%.

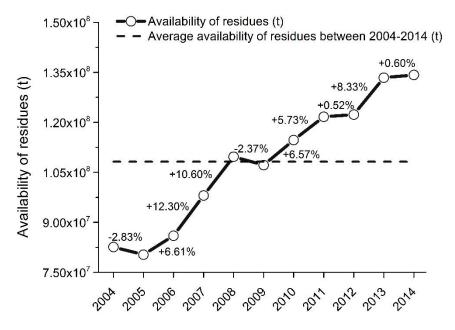


Figure 3-4. Annual evolution of availability of residues in Brazil.

3.3.4. Suitability of Agroindustrial Residues

The ranking of residues with greater potential to be used as raw material (Figure 3.5) was obtained from the substitution of calculated parameters AR, CC; e $R_{c/u}$ (Table 3.6) in equation 3. It could be noticed that, mainly the parameters AR e $R_{c/u}$ had a significant influence on the residues ranking. The low lignin content associated with pineapple peel resulted in a normalized cellulose-to-lignin ratio much higher than of other residues (Table 3.3 and 3.6), which contributed

significantly to the final value of suitability. The same is true for sugarcane top/leaves and soybean straw, in response to their great availability.

Table 3.6. Availability of residues, Cellulose content and cellulose-to-lignin ratio normalized values.

| Residue | Normalized available residue (AR;) | Normalized cellulose content (CC;) | Normalized cellulose-to-lignin ratio (R _{c/u} ') |
|----------------------|------------------------------------|------------------------------------|---|
| Cocoa shell | 0.105 | 0.551 | 0.353 |
| Coconut husk | 0.110 | 0.576 | 0.195 |
| Coffee husk | 0.156 | 0.672 | 0.610 |
| Maize cob | 0.196 | 0.484 | 0.302 |
| Maize husk | 0.164 | 0.977 | 0.550 |
| Maize stover | 0.668 | 0.637 | 0.277 |
| Sugarcane bagasse | 0.620 | 0.674 | 0.269 |
| Cassava stalk | 0.123 | 0.605 | 0.440 |
| Rice husk | 0.124 | 0.544 | 0.239 |
| Rice straw | 0.255 | 0.616 | 0.348 |
| Bean straw | 0.135 | 0.628 | 0.318 |
| Cotton hull | 0.116 | 0.694 | 0.378 |
| Cotton stalk | 0.224 | 0.631 | 0.285 |
| Banana pruning | 0.188 | 0.816 | 0.426 |
| Mango seed | 0.100 | 0.864 | 0.328 |
| Apple pomace | 0.101 | 0.100 | 0.100 |
| Soybean straw | 0.807 | 1.000 | 0.827 |
| Sorghum stalk | 0.112 | 0.652 | 0.326 |
| Sugarcane top/leaves | 1.000 | 0.621 | 0.267 |
| Wheat straw | 0.142 | 0.551 | 0.281 |
| Pineapple peel | 0.101 | 0.336 | 1.000 |
| Tomato pomace | 0.100 | 0.450 | 0.123 |
| Potato skin | 0.100 | 0.153 | 0.364 |

Among the major agroindustrial residues identified, those with greater potential to be used as raw material are soybean straw, sugarcane top/leaves, maize husk, maize stover and sugarcane bagasse. The lowest suitability values were assigned to apple pomace, potato skin and tomato pomace (Figure 3.5). A close relationship between the suitability of residues and the productivity of crops was identified, since the most suitable residues are originated from the most productivity crops. Additionally, in the production and manufacturing cycle of agricultural products, the largest parcel of residues is generated during the harvest phase. However, some residues coming from

processing industries excel as potential feedstock for cellulose-based materials manufacturing, namely sugarcane bagasse, mango seed, coffee husk and pineapple peel.

Residues with intermediate or low value of suitability may also be applied to the application concerned. However, in such situations, the feasibility of using these residues should be assessed, since the improvement and optimizing of treatments or pretreatments techniques must be carried out. For instance, besides to affect the efficiency of pretreatment steps, the presence of substantial amounts of non-cellulosic components may negatively influence their biodegradability, crystallinity, density, tensile strength, modulus and moisture of fibers and end products [11, 116, 117].

It is worth mentioning that the use of agricultural residues as polymer feedstock requires to encompass a number of issues besides the availability and chemical composition. The temporal variability of generation, technological alternatives available for valorization, perception and social impact on different farmers' category and logistical issues should also be taken into account. For instance, in Brazil, about 63% of sugarcane production takes place in the southeast region, hence, the largest share of sugarcane residues will be available in that region.

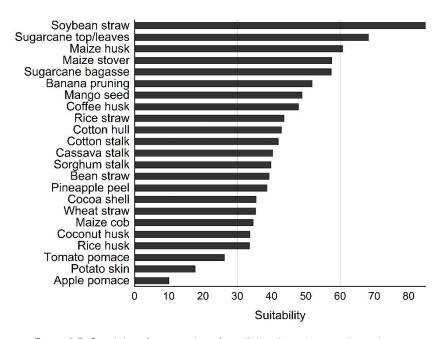


Figure 3-5. Suitability of agroresidues for cellulose-based materials production.

Several agroindustrial residues highlighted herein have already been reported as source for nanocellulose production, namely mango seed [110], soy hulls [118], cotton stalk [119], sugarcane bagasse [120], oil palm empty fruit bunch pulp [121], corncob [122] and wheat straw [123]. Ongoing studies in this field are focused mainly on the use of nanocrystalline cellulose (NCC)

and microfibrillated cellulose (MFC) as fillers to nanocomposites, aiming the improvement of packaging and films properties [124–126].

Other newly applications and trends on cellulose-based materials are focused on the functionalization of natural lignocellulosic fibers, to produce polymeric composites [127–129] and biosorbents [130, 131], and NCC and MFC for assembling well-defined nanomaterials with unique properties [14, 119, 126, 132]. Additionally, is worth mentioning the development of regenerated cellulose, which is manufactured through the dissolution of cellulose, followed by its shaping and subsequent regeneration [13], and the conversion of lignocellulosic biomass into chemical building blocks through the biorefinery concept approach [133, 134].

3.4. Conclusion

In this study, a methodology to select agroindustrial residues suitable for cellulose-based materials production has been conducted based on the agricultural sector of Brazil. The chemical composition and availability of residues were taken into account. The availability of residues was estimated at about 108 million tons/year, on average. Moreover, it is likely that residues generation keeps growing, once, since 2005 the availability of residues has shown positive growth rate. However, the spatial and temporal variability related to residues generation should be considered when planning valorizations routes.

A number of residues highlighted herein present potential to be used as feedstock to produce cellulose-based materials. Among the most suitable residues are the soybean straw, sugarcane top/leaves, maize husk and stover and sugarcane bagasse. Triggered by the search for sustainable process and products, over the next decades these residues tend to be increasingly seen and used as a potential resource. Notwithstanding, besides to present acceptable physical performance, the viability of using the new developed products will be defined by environmental and economic aspects associated with their production, use and post use destination chains, which must be assess by life and cost cycle analysis.

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Chapter 4. Effect of combined dilute-alkaline and green pretreatments on corncob fractionation: pretreated biomass characterization and regenerated cellulose film production

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Abstract

Pretreatment is an essential step prior to the effective valorization of lignocellulosic biomass. However, depending on the pretreatment method, it can be costly and originate potential environmental threats. In this study, green pretreatments, namely Liquid Hot Water and 1-butyl-3methylimidazolium chloride, as well as low concentrated sodium hydroxide solution, were applied individually and in combination to pretreat corncob, and their effects on physicochemical properties of fibers were analyzed. The pretreated samples were dissolved in ionic liquid and attempts to prepare regenerated cellulose films have been made. While NaOH was efficient to removal lignin, LHW presented higher solubilization hemicellulose. By combining these treatments, the biomass was fractionated in a complementary way, and a solid enrich-cellulose fraction with high thermal stability and crystallinity was obtained. The BmimCl did not significantly change the chemical composition of biomass and, independent of the treatment applied in combination, samples were regenerated as amorphous cellulose coexisting with lesser amount of cellulose crystalline I. This structural conversion was also confirmed through thermogravimetric analysis, from which a decrease in thermal stability was verified. This latter property was markedly affected by the presence of hemicellulose remained after some pretreatments. The obtained results indicate that each pretreatment performed can meet different application requirements. The LHW-NaOH pretreated sample was the only one suitable to produce regenerated films, which exhibited good mechanical properties to be used in value-added applications, such as packaging.

Keywords: lignocellulosic biomass; green pretreatments; regenerated film; cellulose; crystallinity; thermal stability.

4.1. Introduction

The depletion of fossil resources, as well as the social and environmental impacts associated with its extraction and production chain, have fostered, in recent years, the search for unconventional and renewable substitute raw materials. Among the available sources, lignocellulosic biomass has been extensively studied and has shown potential to be applied as a starting substrate to produce value-added materials, such as biofuels and biopolymers [1–4]. Its abundant availability and accessibility make the LCB a source of easy acquisition and low cost, but still underexploited [5]. For instance, Corncob is an agroindustrial residue widely generated and

little valued. To date, corncobs are used on a limited basis for bedding, animal feed, oil sorbent and polishing agent [5, 6]. In Brazil it is estimated an average generation of more than 18000 kilotons per year, of which 2990 kilotons would be available [5].

LCB is essentially composed of natural biopolymers (cellulose, hemicellulose and lignin) and its main sources include agroindustrial residues, forest residues and municipal and industrial waste [7]. The main drawback regarding the valorization of LCB deals with the presence of covalent crosslinks between cellulose and hemicellulose with lignin in the cell wall of plants, which leads to its recalcitrant property. Hence, in order to disrupt this complex structural arrangement, pretreatments are usually performed prior to its effective use and valorization.

The application of pretreatment techniques basically promotes the disruption and fractionation of LCB components, which opens the material structure and facilitates the performance of subsequent processes [8, 9]. When it turns to added value cellulose-based biopolymeric materials production, the application of pretreatments aiming the selective delignification of biomass to obtain a solid cellulose enriched fraction is of crucial importance. Otherwise, the presence of substantial amounts of non-cellulosic materials on fibers may negatively influence their structure and properties [10–12].

Basically, the pretreatments methods can be classified into four categories, namely physical, chemical, physicochemical and biological. Some of the existing and applicable methods make use of harsh chemicals and severe treatment conditions [13]. However, as the use of cellulose extracted from renewable raw materials is mainly motivated by its environmentally-friendly concept, it does not make sense to adopt non-sustainable processing techniques. Therefore, more recent studies within this scope have been aimed at optimizing the so-called green pretreatment technologies in order to develop less costly and environmentally harmful processes [13–16]. In addition, since the performance and efficiency of each pretreatment varies widely according to its reaction mechanisms and the chemical and structural characteristic of biomass, the optimization of a single pretreatment is not always the universal choice for a specific purpose [13, 17]. Thus, to overcome possible drawbacks associated with single operations, such as technological problems, high-energy consumption and long reaction times, combined processes have been extensively applied. Generally, their application can significantly improve the synergistic aspects involved in the treatment of biomass, which can lead, for example, to better performance of delignification [16, 17].

Liquid hot water and room temperatures ionic liquids in response to their fractionation efficiency and eco-friendly aspect, are among the main green pretreatment methods applied to fractionate biomass. Regarding the conventional techniques, NaOH pretreatment is the most widely chemical pretreatment used, mainly because of its simplified application and excellent delignification capacity. NaOH solutions essentially cause the cleavage of intermolecular α - and β -aryl ether linkages between hemicellulose and lignin, leading to a swelling of the biomass pores, disruption of the lignin structure and an increase of available surface area [17–20]. This alkaline treatment is one of the preferred pretreatment technologies since it is carried out under mild conditions, does not present physical hazard (explosive, combustible and oxidizing), does not have sufficient volatility to constitute an inhalation danger, besides being non-carcinogenic, non-persistent and nonbioaccumulative [21–23]. In addition, the direct contact with NaOH solutions should be prevented only for concentrations higher than 2% (in water), due to its corrosive effects [24]. The main downsides related to this pretreatment deals with the processing time and the black liquor generated. These problems can be partially mitigated by using the NaOH technique in combination with other treatments, which ultimately demands low concentrated solutions and reduced processing times [25]. Notwithstanding, it is worth to emphasize that the black liquor has yet to be treated even after dilute-alkaline treatment, and its economically and environmentally feasibility must be taken into account when planning large scale plants.

Liquid Hot Water is a physicochemical or hydrothermal method that uses only water at reasonably milder conditions compared to other pretreatments. During the process, water penetrates into the biomass structure and substantially hydrolyze hemicelluloses (up to 80%) and partially removes lignin [13, 18, 26–28]. The chemical effect in the LHW occurs as an autohydrolysis process since the organic acids released from polysaccharides auto-catalyze the hydrolysis of hemicellulose and lignin. The main advantages of this method comprise low environmental impact, low cost and the no need of chemicals or catalysts. Ionic Liquids are salts with low melting point (< 100 °C) composed by organic cations and organic or inorganic anions. The application of ionic liquids in chemical reactions is considered a green method due to its characteristics, namely thermal stability, inflammability, low volatility and recyclability [29–31]. Ionic liquids constituted by imidazolium cations, such as 1-allyl-3-methylimidazolium and 1-butyl-3-methylimidazolium chloride, are the most applied to biomass fractionation and cellulose dissolution [31]. In this process, firstly the ionic liquid anions are able to form hydrogen bonds with hydroxyl groups of biomass, leading to its structural disruption and dissolution. Subsequently, the

cellulose fraction can be recovered as a regenerated product by addition of organic solvents. The successful dissolution of lignocellulosic biomass in BmimCl was demonstrated by Fort et al. [32], Kilpeläinen et al. [33] and Pang et al. [34].

Within the scope of cellulose-based materials produced from LCB, regenerated cellulose films synthesized from green and non-derivatizing solvents (such as ionic liquids) have drawn increasing attention as they possess excellent physical and chemical properties, and also because the dissolution and regeneration techniques are simple, sustainable, versatile and easy for industrialization [35]. Regenerated films prepared from others LCB through dissolution in IL have shown remarkable mechanical properties, besides transparency and non-toxicity, which gives them great potential applications [36–39]. However, the production of regenerated cellulose films requires cellulose-enriched samples with a very small percentage of other constituents (such as lignin and hemicellulose). In order to achieve this, standard methods of pulping and bleaching as pretreatments are still commonly used, which goes against the environmental sustainability imputed by the resources and techniques (dissolution and regeneration) applied.

The aim of this work is to investigate the effect of combined pretreatments, including green and dilute-alkaline techniques, on physicochemical properties of corncob fibers and prepare regenerated cellulose films from pretreated samples using BmimCl as a solvent. Combined pretreatments including chemical and physicochemical methods, namely dilute sodium hydroxide, Liquid Hot Water and Ionic Liquid, were selected based on their complementary characteristics, and to the best of our knowledge, some combinations set in this paper have not yet been addressed for the treatment of corncob, namely LHW-dilute NaOH and LHW-BmimCl. In literature, studies performed in the scope of corncob fractionation by applying combined pretreatments can be found, whilst not exhaustive [40–45]. It should be emphasized that the combined treatments proposed are designed as two and three-stage pretreatments. Besides being scalable and non-conflicting, this arrangement may avoid drawbacks such as the generation of sugar degradation by-products in certain combinations, as verified by Imman et al. [45] and Yang et al. [47] when NaOH was used as a catalyst in LHW treatment. The characterization of the physicochemical changes on pretreated samples and regenerated films was accomplished by compositional, morphological, X-Ray diffraction and Fourier transform infrared spectroscopy analysis, as well as by thermal characterization. Dynamic mechanical analysis of the regenerated cellulose films was also performed.

4.2. Materials and methods

4.2.1. Materials

Corncob was obtained from local agricultural cooperatives from the North of Portugal. The corncob residue was firstly ground and sieved to particles sizes from 0.25 to 0.5 mm and stored in sealed plastic bags at room temperature. Cellulose sheets (cellulose pulp), provided by a local factory, were torn into small fragments and dried at 60 °C for 24 h. Ionic liquid 1-butyl-3-methylimidazolium chloride was purchased from Sigma-Aldrich (98%), Germany, and used as received. Sodium hydroxide (pellets) was acquired from Fisher Chemicals.

4.2.2. Liquid Hot Water treatment

Pretreatment was performed in a stainless steel cylindrical reactor (10 cm internal diameter and 24 cm internal height) with an approximate volume of 2 L. Feedstock loading rate of 10% (residue weight / distilled water volume) was pretreated at 190 °C, 0.79 MPa, for 30 min (after the heat-up time) under constant mechanical stirring. After the reactor was cooled by means of a coupled cooling system. The insoluble solid was collected, thoroughly washed with deionized water and dried at 60 °C for further treatments and analysis (this sample was named as LWH-CC). The choice of the experimental conditions was based on the calculation of the severity factor (log(R₀)), which can be used as an indicator of the effects of destruction, disaggregation and depolymerization of biomass. The severity factor was calculated according to Equation (1)[27], that associate the residence time (t, min), the holding temperature (T, °C) and an empirical parameter (14.75) related to activation energy and temperature.

$$\log(R_0) = \log\left[\exp\left(\frac{T-100}{14.75}\right)t\right] \tag{1}$$

4.2.3. Alkali treatment of corncob

Dried corncob and liquid hot water treated corncob (LHW - CC) were soaked in NaOH with 2 wt.% concentration at 90 °C for 1.5 h at a 1/30 solid/liquid ratio (w/w) under magnetic stirring. The alkali treated corncob (NaOH-CC) and LHW-CC (NaOH-LHW-CC) were then filtered and washed several times with deionized water until colorless. The resulting material was dried at 60 °C for further treatments and characterization.

4.2.4. Ionic liquid treatment and regeneration of treated biomass

Prior to biomass dissolution, BmimCl was poured into a 100 mL beaker and thoroughly stirred at 100 °C for 20 min in order to evaporate any residual moiety. After that, raw corncob, LHW-CC, NaOH-CC, NaOH-LHW-CC and cellulose pulp (CP) samples were dispersed, individually, in the ionic liquid at a 4% solid/liquid (w/w) ratio and treated at about 130 °C under magnetic stirring for 5 h, 2 h, 1.5 h,1.5 h and 0.5 h, respectively. The previously pretreated samples were kept under treatment in the ionic liquid until complete dissolution and a homogeneous solution was obtained. Particularly to the raw corncob residue, the treatment was undertaken under different dissolution times, ranging from 1 to 5 hours, in order to verify the optimal treatment condition. However, even after 5 hours of treatment the raw corncob did not dissolve completely.

After the treatment, the temperature was set to about 60 °C and a mixture of acetone and ethanol, composed as a molar ratio X (acetone): Y (ethanol): Z (ionic liquid) with X = 3.5, Y = 1 and Z = 1, was added into the solution containing IL and biomass and stirred vigorously to precipitate the regenerated cellulosic biomass. The solid material precipitated was separated by vacuum filtration, washed 3 - 4 times with distilled water to remove residual ionic liquid and dried in an oven at 60 °C for further treatments and analysis. The resulting BmimCl treated samples are referred to as IL-CC, IL-LWH-CC, IL-NaOH-CC and IL-NaOH-LHW-CC. A schematic arrangement of pretreatments performed is presented in Figure 4.1.

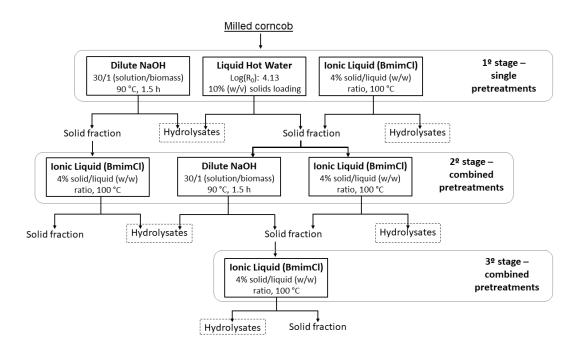


Figure 4-1. Schematic representation of pretreatments performed.

4.2.5. Characterization of untreated and treated corncob

4.2.5.1 Thermogravimetric analysis and Differential scanning calorimetry (DSC)

Thermogravimetry analyses and DSC of untreated and treated corncob were performed simultaneously on TA Instruments SDT 2960 model. The sample weight was in the range of 3-5 mg and heated from 30 to 700 °C at a rate of 10°C/min, under purified argon to prevent thermo-oxidative degradation.

4.2.5.2 Chemical composition of untreated and treated biomass

The chemical composition of all samples was determined by deconvolution of the derivative curves obtained from the thermogravimetry analysis [48–51]. The decomposition processes of the three pseudo-components presented in the lignocellulosic biomass (cellulose, hemicellulose and lignin) were modeled by a Gaussian distribution. For this, a multi-peaks method was used to determine the Gaussian distribution model using the Levenberg–Marquardt algorithm (the modeled equation, as well as graphs and parameters are available in the Appendix A). The percentage of each component was assumed to correspond to the integrated area above each single deconvoluted reaction curve.

4.2.5.3. Fourier transform infrared spectroscopy (FTIR)

FTIR spectra of cellulose, as well as treated and untreated biomass samples, were recorded on a Jasco FT/IR-4100 spectrometer. The ground and dried samples were pelletized with KBr and the spectra were obtained in the wavelength range 400 – 4000 cm⁻¹ at a resolution of 4 cm⁻¹.

4.2.5.4. X-Ray Diffraction (XRD) analysis

The X-ray diffractograms patterns of all samples were obtained at room temperature using a diffractometer Bruker D8 Discover, operated with Cu-K α radiation (wavelength of 0.154 nm) at a power of 40 mA and accelerating voltage of 40 kV. Diffraction intensities were scanned in the range of $2\theta = 5 - 50^{\circ}$ with a scanning rate of 0.05°/second. The crystallinity index (CrI) of samples was determined by the Segal method [52] according to the following Equation (2).

$$CrI = \left(\frac{I_{002} - I_{AM}}{I_{002}}\right) \times 100 \tag{2}$$

CrI express the relative degree of crystallinity, I_{002} is the maximum intensity of the 002 lattice diffraction, found between the scattering angles of $2\theta = 22^{\circ}$ and 23° , and I_{AM} is the minimum intensity, positioned between the 002 and 101 diffraction peaks (typically between $2\theta = 18^{\circ}$ and 19°). It is assumed that the I_{AM} have little crystalline contribution, comprising only the amorphous contribution, while the I_{002} represents both crystalline and amorphous regions.

4.2.5.5. Morphological analysis

The surface morphologies of untreated and treated samples were examined by scanning electron microscopy (SEM) using a FEI Nova 200. Samples were previously sputter coated with gold.

4.2.6. Preparation of regenerated cellulose film

To produce the cellulose films, a 4% (wt.) solution was prepared by dissolving 0.2 g of the extracted cellulose in 5 g of BmimCl under magnetic stirring at 120 °C for 1.5 h. After completely dissolved, the regenerated film was made by solution casting on a glass plate using an alloy spreader, followed by soaking in 70 ml of deionized water. After 1h the deionized water was replaced and left for another hour. Then, the regenerated cellulose film was dried at room temperature for 24 h in vacuum desiccator.

4.2.7. Dynamic mechanical analysis

Dynamic mechanical analysis was performed on a PerkinElmer DMA 800 using a dual-cantilever bending mode in order to obtain the storage modulus (E'), the loss modulus (E") and the damping capacity (tan δ), the latter calculated as the ratio of the loss to the storage modulus. Regenerated films with 20/4/0.015 mm of dimensions, approximately, were analyzed under temperature ranging from 40 – 250°C, heating rate of 5 °C/min, preload 1N, 10 μ m of displacement amplitude and 1 Hz of frequency. Dynamic Young's modulus (E*) was also determined by using equation 3 [53].

$$E^* = \sqrt{(E')^2 + (E'')^2}$$
 (3)

4.3. Results and discussion

4.3.1. Compositional analysis

The chemical composition of untreated and treated corncob is presented in Table 4.1, and images of all samples are presented in Figure 4.2. Depending on the treatment type, considerable physical changes, namely color and texture, are noticeable. These changes are mainly due to the removal or not of color-conferring constituents (such as lignin), chemical transformations or changes in crystalline structure.

The corncob showed a content of cellulose, hemicellulose and lignin similar to previous studies [6, 54]. It is important to remark that corncob presents also other constituents, such as ash (generally present in percentages around $2.88\% \pm 0.11$) and extractives (generally present in percentages around 2.1%) [55–57], which are not accounted in Table 4.1 because the compositional analysis method applied does not have enough sensitivity to estimate them.

The dissolution (and subsequent regeneration) of corncob in IL did not change significantly its final composition. As it can be seen, the treatment with ionic liquid was only able to remove a small fraction of hemicellulose. However, the physical appearance of the corncob after this treatment changed and seemed to be smoothed, probably due to a structural transformation of cellulose. As expected, the treatment with LHW effectively solubilized hemicellulose (around 90% of the original content), leading to a final product rich in cellulose and lignin. Other authors have also reported 80-100% of hemicellulose hydrolysis after LHW treatment of lignocellulosic biomass [27, 28, 58– 60]. This result also indicates that LHW treatment is not effective in lignin removal, which is in agreement with other published works [27, 61]. It's worth mentioning that, under the treatment conditions chosen, the severity (severity factor, $\log R_0$) of the reaction ($\log R_0 = 4.13$) results in high levels of hemicellulose solubilization, as well as low cellulose depolymerization [27, 28, 62, 63]. Under this treatment condition, the generation of hydronium ions by ionization of water is known to increase, leading to an effective dissolution of primary products related to hemicellulose constitution (e.g. xylan and acetyl groups) and their subsequent decomposition into oligomers and organic acids, such as acetic acid. This latter auto-catalyze the polysaccharides breakdown into soluble fractions and results in higher hydrolysis efficiency [27, 63]. As a result of the LHW treatment, substantial morphological changes have been taken on biomass (as will be also shown in section 4.3.4 Scanning electron microscopy), which are usually associated with an increase in available surface area and porosity [61, 63, 64]. In contrast, the alkali treatment removed mainly the lignin fraction. A lignin content reduction by 50% resulted in a mass loss at around half of the

original weight of the untreated corncob on a dry basis. Moreover, by removing lignin, the alkali treatment partially disrupts the linkages between the lignin and hemicellulose, and consequently solubilizes hemicellulose (its reduction can be noticed in Table 4.1).

Regarding the combined pretreatments, even after the prior changes and weakening of corncob structure reached through the LHW treatment, the subsequent dissolution in BmimCl only removed a small amount of the remaining lignin. Conversely, the regenerated product presents a smooth and homogenized morphology, as well as a film-like aspect (Figure 4.2), which proves that, by removing hemicellulose, the LHW treatment increased the accessibility of cellulose and favored its dissolution in BmimCl. Minor effects on chemical composition have also been noticed by combining NaOH and IL treatments (Table 4.1). In line with other studies, these results demonstrate the low capacity of the BmimCl in fractionating the biomass [65, 66]. On the opposite, remarkable changes were verified by combining LHW with alkali treatment. This can be explained by the solubilization of most hemicellulose during the LHW treatment, the disruption of the lignin-hemicellulose complexes and a decrease in β -0-4 linkages of lignin (the most abundant substructure in this component) also took place, which makes the lignin more accessible [61, 64]. Consequently, the cleavage of ester, ether and carbon-to-carbon bonds between lignin-cellulose, lignin components (e.g. syringyl and guaiacyl monolignols) and the remaining lignin-hemicellulose complexes was easier during the alkaline treatment, leading to the breaking of lignin into small molecules and their dissolution. The broken of lignin structure and its decoupling with carbohydrates are confirmed and easily identified by FTIR analysis (Figure 4.3). Their complementary and selective effects of LHW and NaOH treatments on fractionating the biomass contributed positively to obtain a solid celluloseenriched final product (84.73%), with reasonably low amounts of hemicellulose (reduced by 90%) and lignin (reduced by 34.5%). This result is comparable to the ones archived by established laboratory methods for delignification in which, besides alkaline, acidic and oxidative reagents are used. Little improvements have been made to the fractionation process by combining LHW, NaOH and IL treatments. However, the dried regenerated material was semitransparent and had a plastic film-like appearance (Figure 4.2g), completely different from the starting material (Figure 4.2f). This physical characteristic suggests a decrease in the Crl of the sample in response to an amorphous conversion, as it will be, indeed, confirmed and explored in detail in the following analysis.

| Sample | | - M (0/) | | |
|----------------|-----------|---------------|--------|-------------------|
| | Cellulose | Hemicellulose | Lignin | Mass recovery (%) |
| Corncob | 36.35 | 47.32 | 16.32 | - |
| IL-CC | 39.28 | 43.70 | 17.02 | 90 |
| LHW-CC | 78.34 | 3.95 | 17.70 | 59.5 |
| NaOH-CC | 64.12 | 27.88 | 8.0 | 46 |
| LHW-IL-CC | 80.96 | 4.02 | 15.01 | 56.52 |
| NaOH-IL-CC | 67.19 | 25 | 7.81 | 43.2 |
| LHW-NaOH-CC | 84.73 | 4.58 | 10.68 | 37.68 |
| LHW-NaOH-IL-CC | 86.10 | 3.53 | 10.35 | 36.7 |

Table 4.1.Chemical composition of untreated and pretreated corncob.



Figure 4-2. Photographs of (a) corncob, (b) ground corncob, (c) ionic liquid treated corncob, (d) LHW treated corncob, (e) NaOH treated corncob, (f) LHW-NaOH treated corncob, (g) LHW-NaOH-IL treated corncob, (h) NaOH-IL treated corncob.

4.3.2. Fourier-transformed Infrared spectroscopy

FTIR spectra of untreated and treated corncob are shown in Figure 4.3. The broad band observed at 3000-3700 cm⁻¹ is assigned to different OH stretching of cellulose and hemicellulose [67, 68]. The peak absorption at 1640 cm⁻¹, typical of absorbed water in biomass, was also verified in all samples analyzed [69]. From Figure 4.3, well defined common bands typical of functional groups in cellulose were identified at 3000-3700 cm⁻¹ (O-H stretching), 2919 cm⁻¹ (C-H stretching vibration), 1420 cm⁻¹ (CH₂ banding vibration), 1374 cm⁻¹ (C-H asymmetric deformation), 1318 cm⁻¹ (CH₂ wagging), 1164 cm⁻¹ (asymmetric C-O-C stretching), 1114 cm⁻¹ (C-OH skeletal vibration), 1059 cm⁻¹ (C-O-C pyranose ring skeletal vibration) and 897 cm⁻¹ (C-H deformation vibration, characteristic of the β-glycosidic linkages between glucose monomers) [37, 67, 69–71]. As it can be seen in Figure 4.3a and 4.3b, all typical bands of cellulose can be easily identified in the spectrum after NaOH and LHW-NaOH treatments. Besides, notable bands at around 1730 (acetyl group in hemicellulose), 1250 cm-1 (C-O stretching in the aryl-ether group of lignin and acetyl groups in

hemicellulose), 1598 and 1513 (in response to aromatic ring stretch ascribed to lignin), are notice mainly in the raw corncob spectrum (Figure 4.3a), which were attributed to the presence of hemicellulose and lignin [37, 67]. The intensity of these bands has become significantly reduced after LHW-NaOH and NaOH treated corncob, which evidences their better capacity to partially fractionate and remove non-cellulosic components of corncob.

Less noticeable chemical changes were observed on the spectra after other treatments, such as LHW and IL (Figure 4.3a), mainly due to the high content of lignin remaining in the biomass after treated. However, the appearance of peaks at 1420 and 1114 cm-1 (in the LHW treated sample) and the more pronounced peak at 1059 cm⁻¹ in both treatments, compared to those in raw corncob, suggest a higher cellulose content. Table 4.2 shows the main absorption bands and the respective group of cellulose, as well as treated and untreated biomass samples.

Table 4.2. FTIR absorbance bands and the respective functional groups in cellulose pulp, untreated and treated corncob.

| Wavenumber (cm ⁻¹) | Chemical group | assignment | Reference |
|--------------------------------|-----------------|---|-----------|
| 3000-3700 | О-Н | O-H stretching of cellulose and hemicellulose | [67, 68] |
| 2919 | C-H | C-H stretching in cellulose-rich material | [70] |
| 1730 | C=0 | C=O stretching of acetyl group in Hemicellulose and aldehyde in lignin | [37, 67] |
| 1640 | 0-H | O-H bending vibration of absorbed water molecules | [37, 72] |
| 1598 and 1513 | | Aromatic ring (Ar) stretch ascribed to lignin | [70, 73] |
| 1420 | CH ₂ | CH _z banding vibration in cellulose | [37] |
| 1374 | C-H | C-H asymmetric deformation in cellulose | [37] |
| 1318 | CH₂ | CH ₂ Wagging in cellulose | [37] |
| 1250 | C-O | C-O stretching in the aryl group of lignin and acetyl groups of hemicellulose | [74] |
| 1164 | C-O-C | Asymmetric C-O-C stretching ascribed to cellulose | [69, 71] |
| 1114 | C-OH | C-OH skeletal vibration in cellulose | [71] |
| 1059 | C-O-C | C-O-C pyranose ring skeletal vibration ascribed to cellulose | [71, 72] |
| 897 | C-H | C-H deformation of cellulose | [69–71] |

FTIR spectra can also be applied to investigate changes in cellulose crystalline structure. In the cell wall of plants, cellulose mainly contains cellulose crystalline I, but, depending on the type of processing and treatment to which it is submitted, this primary structure can be transformed into other allomorphic forms, such as amorphous cellulose and cellulose crystalline II [37, 75–77]. Variations in the absorption bands around 1425, 1370, 1157, 1108 and 898 cm¹, typical of cellulose I, can be used as an indication of crystalline structural transformations [37, 68, 69]. From the cellulose pulp spectrum in Figure 4.3, it is possible to assume that characteristic bands at 1420, 1114 and 897 cm⁻¹ correspond to cellulose crystalline I. Among the treatments performed, those involving the dissolution of cellulosic biomass with ionic liquid significantly decreased the intensity of bands at 1420 and 1114 cm¹, while the absorption intensity band at 897 cm¹ increased after LHW-IL treatment and kept unchanged after NaOH-IL and LHW-NaOH-IL treatments. Moreover, from ionic liquid combined treatments spectrum, it is visible that the peaks assigned to O-H stretching vibrations (3000-3700 cm⁻¹) became sharper, probably due to the scission of intraand intermolecular hydrogen bonds [78]. The FTIR spectra of BmimCl treated samples seems to be typical of amorphous cellulose, although they also resemble cellulose crystalline II [37, 78]. These crystalline structural changes will be clearly elucidated with the XRD analysis, which will confirm that the combined IL treatments result mainly in amorphous cellulose.

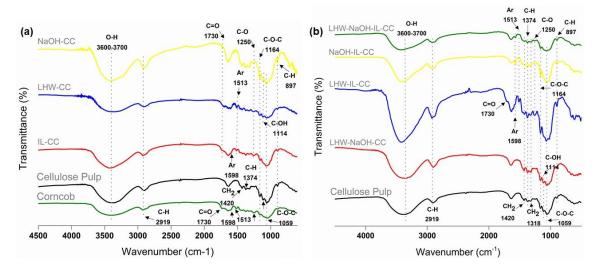


Figure 4-3. FTIR spectra of (a) cellulose pulp, corncob and single pretreatments and (b) combined pretreatments.

Abbreviation Ar in graphs means aromatic ring.

4.4.3. X-ray Diffraction Analysis

X-ray diffractograms of cellulose pulp, corncob and all treated samples are shown in Figure 4.4. Diffractograms of cellulose pulp and samples treated with LHW, NaOH and LHW-NaOH present three major peaks, typical of cellulose crystalline I. The main peak position varies with sample analyzed from $2\theta = 22.2^{\circ}$ to $2\theta = 22.6^{\circ}$. This peak is indicative of the distance between hydrogenbonded sheets in cellulose crystalline I. The second peak, viewed as a wide peak at lower 2θ values, is known to be correlated to allomorphs of cellulose crystalline I, and varies from 15.5° to 16°. The third is depicted as a small peak at 34.7° and arises from ordering along the fiber direction [78-82]. The Crl of samples was determined by the Segal method [52] as specified in section 4.2.5.4 of the materials and methods. This method provides a faster and useful measurement for comparing the relative differences of Crl among samples. In the literature several different techniques are reported to measure Crl, and depending on the chosen method the Crl can vary significantly [80]. The obtained Crl values of cellulose pulp (Crl = 70), corncob (Crl = 30) and samples treated with LHW (Crl = 50), NaOH (Crl = 63) and LHW-NaOH (Crl = 67) are shown in Figure 4.4a and 4.4b. The high Crl of samples treated with LHW, NaOH and LHW-NaOH, in comparison to the corncob, can be related to the removal of non-cellulosic components. Despite the high cellulose content in LHW-CC sample, it still presents a reasonable percentage of lignin (Table 4.1), which is closely associated with cellulose microfibrils and directly affects the crystallinity. It can also be observed that LHW-NaOH treated sample presented CrI value very close to the cellulose pulp, and from the high-intensity peak at 34.7° is possible to deduce that this sample has a greater ordering of fibers.

Samples subjected to combined treatments with ionic liquid (Figure 4.4b) gave diffractograms that clearly support the previous results obtained by FTIR. In general, a strong reduction of peaks around $2\theta = 22^{\circ}$ and 23° and $2\theta = 15^{\circ}$ is observed, as well as the appearance of a single broad peak extending from 15° to 22.5° , which are characteristic of amorphous cellulose [78, 80]. In fact, throughout the dissolution of cellulosic biomass in BmimCI, initially the cations and anions of the IL come into contact with the hydrogen bonds C2-0 and C6-0 of cellulose, forming new H-bonds and making the surface chains swelling, followed by the gradually peeled of its chains and the breaking of C3-0 hydrogen bonds, which result in the partial or complete dissolution of cellulose [83]. As a result of the hydrogen bond network disruption, the cellulose is not allowed to recrystallize during the fast regeneration process and settles as an amorphous form [82, 84, 85].

Similar results were reported in several studies [37, 65, 86]. Amorphous structures were also obtained by [85] independent time dissolution in BmimCl and the regeneration medium.

Especially for the NaOH-IL-CC regenerated sample, three distinct diffracted peaks appeared at $2\theta=12.2^\circ$, 19.9° and 22° (Figure 4.4b). The emerging of these peaks, along with the spectra shape, might suggest a possible conversion from cellulose crystalline I to II [37, 87], but actually these features result from the conversion of cellulose crystalline I to amorphous cellulose and the high amount of hemicellulose embedded in this sample. As a consequence of this amorphous conversion after IL treatment, the crystalline peaks identified in the NaOH pretreated samples were reduced and the diffracted peaks related to hemicellulose, typically located around $2\theta=12.4^\circ$ and $2\theta=19.1^\circ$ [88], were emphasized, given rise to the concerned diffractogram shown in Figure 4.4b. In addition, the high content of hemicellulose in this sample seems to have a negative effect on cellulose dissolution, since the peak at $2\theta=34.7^\circ$ associated with the cellulose crystalline I can be notice in the NaOH-IL-CC spectra. Therefore, it is possible to assume that, besides the high hemicellulose content, the NaOH-IL combined treatment results in a cellulosic biomass sample composed mainly by amorphous cellulose and lesser amounts of cellulose crystalline I. This result is also supported by the low temperature of maximum rate of weight loss related to this sample (Figure 4.6b) verified through thermogravimetric analysis.

Regarding the raw corncob, there were no significant changes between the patterns of untreated and only IL-treated, except for a decrease in the intensity of curve related to the IL-CC (Figure 4.4a). Therefore, from the obtained results it is possible verifying that, independent of the pretreatment applied prior to dissolution in BmimCl, amorphous cellulose is the dominant structure in the regenerated samples.

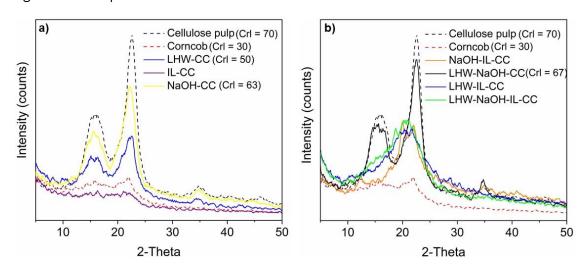


Figure 4-4. X-ray diffractograms of (a) cellulose pulp, corncob and single pretreatments and (b) combined pretreatments.

4.3.4. Scanning electron microscopy

SEM study of samples (Figure 4.5) shows that untreated corncob has a rigid, porous and irregular morphology assembled into compact layers with different surface textures. Corncob is made up of different components, namely the beeswing, coarse chaff and woody ring, which confers stiffness, and the pith, the innermost part of the cob, with a consistency similar to foam plastic [89]. Each component has different morphology, constitution (regarding the content of cellulose, hemicellulose, lignin and waxes), and consequently different physicochemical features, which may play an important role in the characteristics of the material obtained after the pretreatments. From Figure 4.5e it is clear that the morphology of BmimCl treated corncob remained relatively unchanged (no defibrillation process can be seen), only smoothing and homogenization of the surface were observed. After the treatment with LHW considerable morphological changes became evident. The defibrillation process, resulting mainly from the removal of hemicellulose, partly collapsed the cell wall of corncob, leading to an increase in available surface area and porosity. Imman et al. [64] showed that, under the same severity factor, the LHW treatment could duplicate the available surface area of corncob. However, due to the large amount of residual lignin remaining in biomass, the cellulose fibers have kept still arranged together in large bundles. Contrarily, the treatment with NaOH could dissolve bulk and inter-fibrillar lignin, leading to the disruption and disaggregation of biomass fibers.

As a result of the combined LHW-NaOH treatment, more uniform fiber distribution and a remarkable disaggregation are noticed (Figure 4.5d). By promoting the initial defibrillation of the biomass, the previous treatment conducted with LHW increased the accessibility to the complex lignin-cellulose matrix and enhanced the performance of the alkaline treatment. In general, cellulosic fibers with different shapes, sizes and rough surface topography were obtained at the end of the treatment (probably due to the mix of components that make up the corncob). Except for the average diameter of the longer fibers, the characteristics of LHW-NaOH-CC extracted fibers hardly resemble the cellulose pulp fibers (Figure 4.5d and 4.5i). This can be simply explained by the different morphology inherent to the corresponding plant sources and the effect of different background processing/treatments applied to both samples.

Regarding the combined treatments with BmimCl (Figure 4.5f, g and h), the obtained regenerated material had no fibrous aspect and a smooth and uniform surface. Similar regenerated products were also observed by Geng and Henderson [65], Abdulkhani et al. [10] and Reddy et al.

[37], who performed the dissolution of other cellulosic biomass with ionic liquids and obtained regenerated materials with a flat surface.

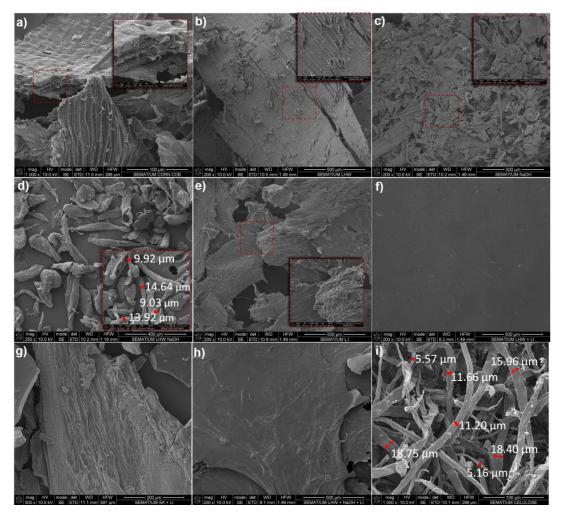


Figure 4-5. SEM micrographs of (a) untreated corncob, (b) LHW-CC, (c) NaOH-CC, (d) LHW-NaOH-CC, (e) IL-CC, (f) LHW-IL-CC, (g) NaOH-IL-CC, (h) LHW-NaOH-IL-CC and (i) Cellulose Pulp.

4.3.5. Thermogravimetric properties

The thermal properties of all samples were investigated by TG and DSC, as illustrated in Figure 4.6. The temperature at which the degradation processes start (T_{onset} , °C), the temperature of maximum rate of weight loss (T_{max} , °C), the weight loss (%) /WL) and the enthalpy of melting (ΔH_{f} , J/g) for all samples are presented in Table 3. Herein, T_{onset} will be considered as the temperature at which the process of decomposition starts, excluding the initial mass loss that was due to water evaporation. When it turns to lignocellulosic biomass, T_{onset} corresponds to the temperature at which low molar mass substances start to degrade, such as hemicellulose and possibly some lignin. Along with the T_{max} , T_{onset} will be used to evaluate the thermal stability conferred to the lignocellulosic

fibers of corncob after pretreatments. Besides, data regarding the thermal process in Table 4.3 are discretized into two phases, that arise, in some samples, as an effect of the chemical composition or physicochemical changes after the pretreatments.

Table 4.3. Thermogravimetric results for cellulose pulp, untreated and treated corncob samples.

| Sample | T _{onset} (°C) | Phase I | | Phase II | | Enthalpy |
|----------------|-------------------------|-----------|--------|-----------|--------|----------|
| | | Tmax (°C) | WL (%) | Tmax (°C) | WL (%) | (J/g) |
| Cellulose pulp | 275 | 357 | 72.8 | | | 101.6 |
| Corncob | 240 | 301 | 38 | 342 | 30.65 | 2.5 |
| IL | 247 | 282 | 37.5 | 356 | 34 | 2.1 |
| LHW | 294 | 351 | 75.17 | | | 233.3 |
| NaOH | 275 | 351 | 72.2 | | | 43 |
| LHW+IL | 241 | 313 | 60.6 | 348 | 8.6 | n.d. |
| LHW+NaOH | 294 | 363 | 75.45 | | | 104.5 |
| LHW+NaOH+IL | 252 | 300 | 70.47 | 348 | 2.9 | 56.6 |
| NaOH+IL | 238 | 277 | 64.7 | | | 38.8 |

From the thermograms exhibited in Figure 4.6, it is noticeable that the pretreatments performed had a remarkable effect over the thermal decomposition process. A small weight loss at lower temperatures ($< 150 \, ^{\circ}$ C) due to water evaporation or low molecular weight compounds is observed in all curves. Two other distinct peaks characteristic of hemicellulose and cellulose decomposition are observed at temperatures between 200 and 400 $^{\circ}$ C. Due to the wide temperature degradation range of lignin (200-600 $^{\circ}$ C), it overlaps partially with cellulose and hemicellulose thermal decomposition curves and, consequently, does not present maximum weight loss rate [19, 77]. In general, with exception of combined treatments with ionic liquid, all the pretreatments increased the thermal stability of fibers compared to the corncob raw fibers.

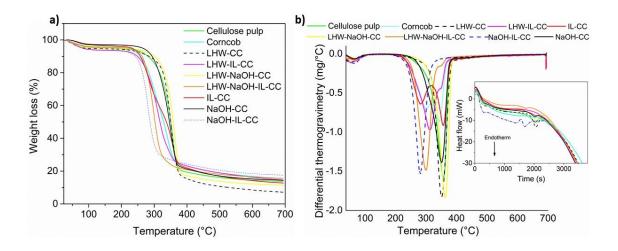


Figure 4-6. (a) TGA, (b) DTG and DSC (inset plot in the graph 5b) curves of the cellulose pulp, untreated and treated corncob.

The thermal profiles related to LHW-CC and NaOH-CC pretreated samples present single step with a narrow temperature range. In comparison to the cellulose pulp curve, the LHW-CC profile has a slightly lower T_{max} derived mainly from its rather high content of lignin. However, its T_{onset} increased significantly in comparison to the corncob profile as a result of the effective hemicellulose solubilization during the LHW treatment. Similar T_{max} value is also displayed by the NaOH treated sample, but due to the high content of hemicellulose embedded in this sample, it presents a broad curve base.

Degradation reactions observed in the curve associated to LHW-NaOH pretreated sample occurred in a single step and in a narrow temperature range, similar to the cellulose pulp curve, which decomposes in a narrow range between 275 and 400 °C (Figure 4.6a). The results evidence that this pretreatment had the best performance in terms of hemicellulose and lignin removal, resulting in a product composed mainly of cellulose, as evidenced by the other analysis performed. Compared to the single pretreatments, the combination between LHW and dilute NaOH solution resulted in a narrower and stable degradation curve, with higher T_{max}. By removing the amorphous hemicellulose and the aromatic rings of lignin, the thermal stability of fibers was increased in response to the high crystallinity of the remaining cellulose chains. The final product obtained after the coupled treatment of LWH-NaOH (T_{onset} of 294°C and T_{max} of 363 °C) presented higher thermal stability than the cellulose pulp (T_{onset} of 275 °C and T_{max} of 357 °C), which might be probably related to the different biological sources of cellulose.

In contrast, despite the increased thermal stability of the crude substrate after ionic liquid treatment, this study suggests that the BmimCl presents reduced performance on the fractionation

of lignocellulosic biomass, which is in agreement with other studies [66], [90], [91] and [65]. As shown in Figure 6, the IL-CC degradation profile can be discretized into two phases and presents two weight loss regions between 200 and 400 ° C (Figure 4.6 and Table 4.3). The first peak, with corresponding T_{max} of 282 °C, is related to hemicellulose degradation, while the second assigns to the cellulose degradation, with corresponding T_{max} of 356 °C. Few changes are verified when compared to the raw corncob profile. Moreover, when applied in combination with other treatments, the ionic liquid decreased the thermal stability of fibers and displaced T_{max} to lower values. Such effects can be attributed to the conversion of the native cellulose (cellulose crystalline I) to an amorphous structure, which is in agreement with the results obtained in the other analysis. The IL treatment gave rise to a second phase identified by a shoulder-like around 350 °C in the differential thermogravimetry profiles. This temperature is typical of cellulose decomposition and thus confirms the coexistence of cellulose crystalline I. The application of previous treatments on corncob decreased its recalcitrant property and provided the direct interaction of the ionic liquid with the cellulose, which, in turn, causes the rupture of more intermolecular and intramolecular hydrogen bonds of original cellulose during the dissolution process. The same was not observed in the direct treatment of the corncob with ionic liquid, since the cellulose dissolution was hindered by the dense packaging of hemicellulose and lignin and the high viscosity of the ionic liquid, limiting its performance to components more easily hydrolyzable. Particularly for the NaOH-IL sample, the lowest T_{max} value verified is greatly attributed to the high amount of hemicellulose remained after the treatments, which presents maximum rate of weight loss around 300 °C.

From DSC results in Figure 4.6 (inset plot in graph 4.6b), all samples presented endothermic peaks consistent with T_{max} from DTG, which are attributed to the depolymerization of cellulosic and non-cellulosic components. DSC results indicated that the melting enthalpy of the treated samples with IL was drastically reduced compared to the native cellulose and other treatments (Table 4.3), which can be associated to both a decrease of hydrogen bonds and crystallinity of cellulose [92, 93]. Thus, all the results demonstrate that the treatment with IL converted the native cellulose mostly to an amorphous and less thermally stable structure, which corroborates with XRD analysis.

4.3.6. Dynamic mechanical analysis and chemical characterization of regenerated film

Through dissolution with IL and regeneration in deionized water, attempts to produce regenerated films from the previously pretreated samples were made. However, the only sample

suitable to produce films resulted from the LHW-NaOH combined treatment. Due to the remaining reasonably amounts of lignin and hemicellulose in other samples the obtained films were very brittle films and impossible to be analyzed via DMA. The suitability of samples to produce regenerated films supports the previous chemical characterization discussions, from which the superior fractionation capacity of the LHW-NaOH combined treatment was revealed.

Briefly, the regenerated cellulose film presents a uniform and dense smooth surface (Figure 4.7c and d), as well as chemical characteristics similar to that of LHW-NaOH-IL pretreated sample (regenerated as a gel), as can be seen by comparing figures 4.3, 4.4 and 4.7b. Both samples also presented similar thermal properties (not shown). Therefore, it is clear that the different regeneration techniques and anti-solvents applied did not change the chemical structure or influence the formation of crystalline phases, but only resulted in different regenerated shapes. Thus, the regenerated film prepared from the LHW-NaOH-IL combined treatment sample is not included in the following discussions.

Figure 4.7a shows the viscoelastic properties of the produced regenerated film, E' and E'' drop as the temperature increases. The transparency of the film (Figure 4.7d) is also indicative of its amorphous state, since this property is related to the unpackaging of cellulose chains. The operating range of the regenerated film was limited by the glass transition temperature ($T_{\rm s}$), which was found as 120 °C, approximately, and determined through the peak of tan δ curve. Similar $T_{\rm s}$ values of other regenerated cellulose films were reported by Yeng et al. [94] and loelovich [95]. From this temperature, the physical properties of the film may change greatly, which end up compromising its performance depending on the intended application. Moreover, from approximately 75 °C the loss modulus drops markedly, and the mechanical deformation applied begins to be converted into internal friction and nonelastic deformation.

As a new material for potential applications it is important to determine the storage modulus and the dynamic Young's modulus, as they quantity the material's stiffness. Taking as reference the values at 40 °C, E' and E* are 3.04 GPa and 3.06 GPa, respectively. These values are higher than regenerated films produced from cotton linter pulp – 2.54-5 GPa [96, 97], Polylactic acid (PLA) – 2 GPa [98], PLA/bamboo fabric laminate – 1. 75 GPa [99], microcrystalline cellulose – 2.57 GPa [100] and bleached hardwood kraft pulp – 1.5 GPa [101]. The results presented suggest that the regenerated cellulose, produced herein via a sustainable processing from corncob residues, could be potentially used in value-added applications, such as packaging.

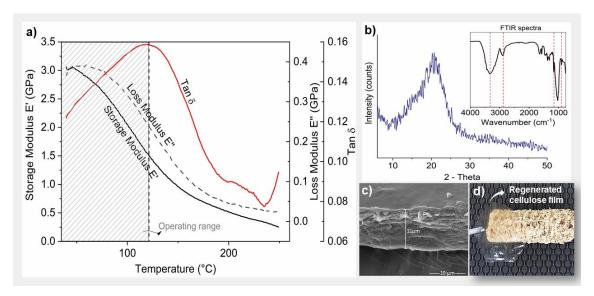


Figure 4-7. (a) Dynamic mechanical analysis, (b) X-ray diffraction and FTIR spectra (inset graph in figure 7b) and (c) SEM cross section of (d) regenerated cellulose film produced from LHW-NaOH pretreated corncob.

4.4. Conclusions

In this study, corncob was pretreated by performing individual and combined green pretreatments, namely liquid hot water and ionic liquid (BmimCl), as well as dilute NaOH solution, and attempts to produce regenerated cellulose films were carried out through dissolution of pretreated samples in BmimCl. The results confirmed the low fractionation capacity of BmimCl and the complementary effects of LHW (which efficiently removed hemicellulose) and NaOH (efficient in lignin removal). The disruption of hemicellulose and the subsequent weakening of the biomass structure resulting from the LHW treatment increased the accessibility of lignin. Therefore, the combined LHW-NaOH pretreatment was very efficient at removing the non-cellulosic components from corncob and leads to a solid enrich-cellulose fraction with high crystallinity and thermal stability. The results also revealed that, regardless of the pretreatment applied prior to dissolution in BmimCl, the regenerated samples were constituted mainly by amorphous structure.

This work shows the efficient application of some combined green and conventional treatments, with results comparable with those obtained by established laboratory methods for delignification. The results also revealed that, suitability of biomass to different types of valorization can be optimized by pretreatments combination. Moreover, the high stable and crystalline cellulose obtained after LHW-NaOH combine treatment gave rise to a regenerated film with outstanding viscoelastic properties, suitable for packaging applications. The sequential process design involving combined green pretreatments and ionic liquid dissolution to produce regenerated cellulose films can be seen as a sustainable approach, but it is still challenging in terms of costs and scalability.

Nevertheless, the development of life cycle and costs assessment can give more insights on the feasibility, design and optimization of this production method.

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Chapter 5. Green synthesis of cellulose acetate from corncob: physicochemical properties and assessment of environmental impacts

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Abstract

The widespread availability and chemical constitution of agroindustrial residues have stood them out as a promising source of natural chemicals and polymers, such as cellulose. However, to date, green alternative synthesis routes developed to turn the agroindustrial residues into value-added products lack environmental performance regarding the quantification of impacts. In this work a promising green pretreatment and acetylation techniques to obtaining cellulose acetate from corncob were applied. Physicochemical characterization techniques were carried out to investigate the effectivity of pretreatment, as well as the properties of synthesized CA. The results revealed successful cellulose extraction and acetylation. CA synthesized by the green and standard acetylation methods presented degree of substitutions of 2.68 and 2.89, and yield of 60% and 40%, respectively. The dynamic Young's modulus and the storage modulus of the produced film were found to be 1.88 GPa and 1.89 GPa, respectively, and its operating range was limited to 140 °C. The environmental performance of the green approach was modeled via life cycle assessment and compared with a conventional approach. The life cycle assessment shows that the green approach is more advantageous than the conventional approach. The less use of chemicals throughout the pretreatment and acetylation steps of the green approach was critical to achieving environmental benefits. Results also indicate that the acetylation processes were the main contributors to environmental impacts, and the primary consumption of dichloromethane in the purification step significantly influences this negative outlook. Environmental savings for human toxicity and stratospheric ozone depletion could reach 27% and 90%, respectively, if DCM was recycled and used in subsequent cycles of acetylation. In addition, sensitivity analysis shows that the high yield obtained by the green acetylation was crucial to lowering environmental impacts.

Keywords: lignocellulosic biomass; agroindustrial residues; pretreatment; cellulose acetylation; life cycle assessment.

5.1. Introduction

The use of lignocellulosic residues as feedstock for bio-based applications has been increasing as a promising alternative for the functioning of modern industrial societies [1-3]. Their attractiveness as a capable sustainable source of biomaterials comes mainly from their low-cost, availability and chemical composition. An unending stream of lignocellulosic residue is derived from agricultural activities. For instance, in countries where prevails an agriculture-based economy, such as Brazil, United States, China and Africa, the annual availability of lignocellulosic residues achieves remarkable values. It is estimated that in Brazil an average of 108 million tonnes/year of agricultural residues are generated [4]. Although some industrial valorization techniques have already been implemented in this country, such as the production of second-generation ethanol and composites, the use of these residues is still very low, and they are usually inserted in the market as waste without added value, which eventually integrate the currents paths of the linear waste management systems. In a recent publication, Araújo et al. [4] provided the availability of various crop and industry processing residues in Brazil, and highlighted, among others, the high generation of corncob, soybean straw, sugarcane top/leaves and rice straw. For instance, the generation of corncob is estimated to be approximately 18.700 kt/year, which is still used on a limited basis.

The agroindustrial residues are mainly composed by cellulose (50%), hemicellulose (20-40%) and lignin (10-40%) [5]. Cellulose is a basic structural component of plants and the most abundant renewable resource in nature. It is a high molecular weight homopolymer generated from repeating D-glucopyranose ring units linked by β -1,4-glycosidic bonds (cellobiose unit). Each glucose monomers within the cellulose chain present three hydroxyl groups positioned in the plane of the ring, which have a great influence on properties [6]. Due to the intra and intermolecular interactions of the hydroxyl groups presented in the cellobiose unit, the cellulose molecules crystallize in a horizontal plane and in parallel chains, forming microfiber packages. The elementary fibers that make up the microfiber packaging are basically organized into high order crystalline parts interspersed by amorphous regions. As a consequence of its structural organization, cellulose presents some properties that limit its application (e.g. lack of antimicrobial properties, high hydrophilicity and low dimensional stability high melting temperature). To overcome these drawbacks, cellulose is generally modified into derivatives. Due to its broad range of applications and easy processability, the modification of cellulosic biomass into cellulose acetate has been the most commonly explored approach. In 2015 the global cellulose acetate production reached the

astonishing value of 4.82 billion tones and the trend is upward [7]. While at large scale (industrial) cellulose acetate is synthesized primary from wood and cotton pulp via the acetic acid route (standard route), at experimental level (bench tests) the use of agroindustrial residues and environmentally friendly synthesis routes have been increasingly reported [8–13]. These alternative synthesis routes are mainly focused on replacing the sulfuric and acetic acids, used respectively as catalyst and reaction media in the standard acetylation, by eco-friendly reagents, such as iodine as catalyst in the presence of acetic anhydride [10, 12], phosphotungstic acid as catalyst [14], room temperature ionic liquid [13, 15] and solid superacid SO₂4/ZrO₂ [16]. However, despite the environmental benefits resulting from the application of these substituent reagents, the aforementioned studies almost unanimously made use of harsh chemicals and processing, such as the conventional alkali and bleaching pretreatment, to isolate the cellulose from the biomass. Since cellulosic fibers do not occur as an isolated component in the agroindustrial residues, instead they are bounded with other non-cellulosic materials which can negatively affect their physicochemical properties, as well as the modification and dissolution processes [8, 17, 18]. Thus, it is essential to apply fractionation techniques to extract the cellulose content prior to acetylation.

The fractionation techniques, also named as pretreatment methods, can be classified into four categories, namely physical, chemical, physicochemical and biological [5]. Currently, the pretreatment step is considered one of the main bottlenecks to the cost-effective valorization of agroindustrial residues. Its application may demand, besides high energy and material consumption, the use of harsh chemicals, which implies the production of excessive wastes. These led to the development of new approaches based on the "Green Chemistry" concept, among which it is worth to mention liquid hot water, stem explosion, organosolv and fine-tuned green solvents (deep eutectic solvent, ionic liquid and supercritical fluids) [19–22]. Recently, great attention has also been focused on combining pretreatments, as reported by Araújo et al. [23] and Sun et al. [24].

Although some studies have reported the application of less impacting pretreatments and acetylation techniques to synthesized cellulose acetate from lignocellulosic biomass [9, 14], for the best of our knowledge, there are no reports addressing the environmental performance modeling of these techniques. Therefore, life cycle assessment has been progressively used to analyze the environmental viability of bio-based products. Moreover, this analysis tool has proven to be reliable for the evaluation of environmental sustainability of products and methodologies within the

framework of lignocellulosic biomass valorization [25-27]. Hence, the purpose of this work is to use potential environmentally-friendly pretreatment and acetylation techniques to synthesize cellulose acetate from corncob and calculate the environmental impacts related to the applied process. The pretreatment method applied is described in a recent publication of Araújo et al. [23], who used a combined LHW-dilute NaOH solution treatment to pretreat biomass and obtained cellulose-enrich final samples with great yield. The acetylation step was based on the solvent-free method as suggested by Das et al. [10], since remarkable yields were obtained from the acetylation of rice husk. The environmental impact associated with this synthesis route was determined through life cycle assessment and compared with those associated with the conventional route, composed of alkali-bleaching pretreatment and acetic acid process acetylation. The specific objectives of this work are: (1) synthesize cellulose acetate by means of green and conventional approaches; (2) investigate the physical-chemical properties of both synthesized samples in order to assess their quality and suitability to be compared via LCA; (3) develop thin films from the cellulose acetate synthesized by the green approach to proves its potential application as a valueadded product; (4) evaluate the environmental impacts resulting from the green approach in comparison to the current conventional approach to synthesize cellulose acetate.

5.2. Materials and methods

5.2.1. Materials

Corncob was obtained from local agricultural cooperatives from the North of Portugal. The corncob residue was firstly ground and sieved to particles sizes from 0.25 to 0.5 mm and stored in sealed plastic bags at room temperature. Cellulose sheets (cellulose pulp), provided by a local factory, were torn into small fragments and fibers and dried at 60 °C for 24 h. Cellulose acetate (Mn₂50.000 by GPC) and Sodium hydroxide (NaOH) pellets were acquired from Sigma-Aldrich. Acetic anhydride (>99%) and sodium thiosulfate (Na2S2O3) (98.50%) were purchased from Acros Organics. Iodine (99.80%), acetic acid glacial (99.88%), sulfuric acid (H2SO4) (>95%), ethanol (99.80%), methanol (99.99%), dichloromethane (DCM) (99.88%) and dimethyl sulfoxide (99.99%) were purchase from Fisher Chemical. All the reactants were used without further purification.

5.2.2. Isolation of cellulose from corncob

Cellulose extraction from dried milled corncob was carried out by applying a combined green pretreatment previously reported [23]. Briefly, a two-step process was performed by combining a

first stage-liquid hot water, followed by dilute NaOH solution treatment. The LHW was performed in a stainless steel cylindrical reactor with an approximate volume of 2 L and feedstock loading rate of 10% (residue weight / distilled water volume). The pretreatment was carried out at 190 °C, 0.79 MPa, for 30 min under constant mechanical stirring. After the insoluble solid was collected, thoroughly washed with deionized water and dried at 60 °C. Then, the dried liquid hot water-treated corncob (LHW - CC) was soaked in 2 wt% NaOH at 90 °C for 1.5 h at a 1/30 solid/liquid ratio (w/w) under magnetic stirring. The alkali treated LHW-CC (NaOH-LHW-CC) was then filtered and washed several times with deionized water until colorless. The resulting material was dried at 60 °C and stored in a sealed plastic bag.

5.2.3 Synthesis of cellulose acetate

Two different acetylation methodologies were carried out, one proposed herein, referred to as a green acetylation, which is based on the method described by Das et al. [10] with some modifications, and another one mentioned as standard acetylation (acetic acid process), which is a commonly used technique and presents similarities (at the level of chemical components used) with the methods applied at industrial level. Regarding the green acetylation, basically, 0.4 g of cellulose extracted from corncob was taken into a 100 mL round bottom flask, and 20 mL of acetic anhydride and 0.6 g of iodine were added. The mixture was heated to 80 °C and left reacting for 5 hours with stirring. After that, the solution was allowed to cool at room temperature and treated with 10 mL saturated solution of sodium thiosulfate with stirring, in order to reduce the iodine to iodide. Then, 60 mL of ethanol was added to the mixture and stirred for 60 min. The product was filtered and thoroughly washed with 75% (v/v) ethanol and distilled water to remove the unreacted acetic acid and byproducts. The solid material obtained was dried at 60 °C in an oven under vacuum and further dissolved in dichloromethane and filtered. Cellulose acetate was formed as a film inside the flask after evaporating the filtrate, which was easily removed without any additional treatment and dried at 60 °C in a vacuum oven overnight.

The standard acetylation applied is based on methods described by Jo et al. [28], Coletti et al. [29], Bello et al. [30] and Cao et al. [9], and was carried out, initially, by reacting the extracted cellulose at different cellulose:reagents ratio (w/w) with acetic acid (1:31.5), acetic anhydride (1:14) and sulfuric acid (1:0.08) in a 100 mL round bottom flask. The mixture was heated to 40 °C and magnetically stirred for 3 h. After that, the reaction was allowed to cool to room temperature, filtered and the filtrate was collected in a large beaker. Then, methanol was added,

and the mixture was stirred overnight at room temperature. Further, the solution was filtrated, and the cellulose acetate was collected in a paper filter and placed in Petri dishes evaporating dishes to evaporate and dry at 60 °C in a vacuum oven overnight. The obtained dried cellulose acetate was dissolved in dichloromethane, filtered and collected as a film after evaporating the filtrate. Then, it was dried at 60 °C in a vacuum oven overnight.

5.2.4. Chemical characterization

FTIR spectra of pretreated corncob, cellulose pulp and cellulose acetate samples were recorded on a Jasco FT/IR-4100 spectrometer, in the wavelength range of 400 – 4000 cm⁻¹ and at a resolution of 4 cm⁻¹, using the attenuated total reflection technique. Thermogravimetry analyses of untreated corncob extracted cellulose, cellulose pulp and CA samples were performed on TA Instruments SDT 2960 model. The sample weight was in the range of 5-10 mg and heated from 30 to 700 °C at a rate of 10 °C/min, under purified argon to prevent thermo-oxidative degradation. The ⁻¹H NMR spectroscopy using a Bruker Avance III, 400 MHz, and deutered dimethyl sulfoxide as solvent. The spectra were internally referenced to the residual proton tetramethylsilane (TMS, δ = 0.00 ppm) and the chemical shifts (δ) are reported as part per million (ppm). Based on Biswas et al. [31], the average number of hydroxyl groups replaced by acetyl groups was determined from HNMR results and used for calculating the degree of substitution (DS) of synthesized CA samples. The DS was calculated by dividing 1/3 of the three methyl proton absorbance peak area of acetyl group, in the range of 1.5-2.2, by 1/7 of the seven anhydrocellulose absorbance peak area, in the range of 3.5-5.2 ppm (Equation 1).

DS=
$$\frac{7 \times \int_{1.5}^{2.2} f(x) dx}{3 \times \int_{3.5}^{5.2} f(x) dx}$$
 (1)

5.2.5. Chemical composition of untreated and treated biomass

The chemical composition of the untreated and treated corncob was determined by deconvolution of the derivative curves obtained from the thermogravimetry analysis [23, 32–35]. Briefly, the decomposition processes of the three pseudo-components presented in the lignocellulosic biomass (cellulose, hemicellulose and lignin) were modeled by a Gaussian distribution and the percentage of each component was assumed to correspond to the integrated area above each single deconvoluted reaction curve (the modeled equation, as well as graphs and parameters are available in the Appendix A).

5.2.6. Cellulose acetate film preparation

Cellulose acetate films were prepared by the casting method. Initially, 50 mg of CA synthesized by the green approach was mixed with 6 g of DCM and stirred for 5 h at 35 °C. After that, the mixture was sonicated for 10 min in order to remove air bubbles. Then, the solution was poured on a Teflon dish with 4 cm of diameter and placed in the vacuum oven at 40 °C to allow the solvent to evaporate.

5.2.7. Morphological analysis

The surface morphologies of untreated and treated samples were observed by scanning electron microscopy (SEM) using a FEI Nova 200. Samples were previously sputter coated with gold.

5.2.8. Dynamic mechanical analysis

A Perkin Elmer DMA 800 Instrument was used at a dual-cantilever bending mode and at a constant frequency in order to obtain the storage modulus (E'), the loss modulus (E") and the damping capacity (tan δ), the later calculated as the ratio of the loss to the storage modulus [36]. CA Films with the size of 20/5.7/0.04 mm, approximately, were analyzed under temperature ranging from 40 – 160°C, heating rate of 5 °C/min, preload 1N, 10 μ m of displacement amplitude and 1 Hz of frequency. Dynamic Young's modulus (E*) was also determined by using Equation 2 [36].

$$E^* = \sqrt{(E')^2 + (E'')^2}$$
 (2)

5.2.9. Life cycle assessment

5.2.9.1. Goal and scope

LCA performed in this study aims to compare the environmental impact associated with two methods used to synthesize CA from corncob. The concerned methods are distinguished by their processing design; while one takes into account strategies under the green chemistry, the other employs the use of conventional techniques. Both product systems contemplate the production of materials and energy, as well as pollutants emission, accounted at the level of laboratory scale when recorded as primary data. Corncob production and process such as transport and wastewater treatment are not considered in this study. Transportation of residues and chemicals are also not considered since processing plants, in this scope of this manufacturing purpose, are still at the

laboratory stage, and consequently their locations have not yet been defined. Notwithstanding, in a scenario projected to facilitate and make feasible logistic issues, these industrial plants would be installed in proximity to companies that generate such wastes, and thus the transportation of residues at least could be disregarded. The functional unit adopted was the synthesis of 10 g of cellulose acetate. The comparison between product systems' LCA results in terms of the same FU was validated through the physical-chemical characterization performed previously.

5.2.9.2. Modelling approach

LCA modeling was performed using the Open Life Cycle Assessment software (openLCA), developed by the GreenDelta and suitable to conducting sustainability assessment of products, process and services [37]. This study followed the guidelines specified in ISO14044 [38], and based on elucidations about LCI framework modelling presented in the International Reference Life Cycle Data System (ILCD) Handbook [39], it is considered an attributional LCA, with similarities, in scope, with studies presented by Arvidsson et al. [26] and Walser et al. [40]. As mentioned before, the cultivation, harvesting and processing of maize are not included within the system boundary as corncob is a by-product of maize with no market value, often disposed of or burned in the harvest field. Moreover, despite the potential application of the synthesized CA as a polymeric film (further presented in the subtopic 5.2.5), their processing and applicability have not been technically and economically defined yet, and therefore, the end-of-life of the material, as a commercial product, is also not included in the boundary system. Consequently, a gate-to-gate perspective is assigned to this study. As emphasized by Arvidsson et al. [26], this kind of perspective is particularly important for materials that might have many subsequent applications, of which some have not yet been developed.

5.2.9.3. Scenarios description and life cycle inventory data

Two scenarios were modeled and assessed as shown in Figure 5.1. The one Indicated as "Green Approach" (left branch in Figure 5.1) includes corncob milling, the combined LHW-Dilute NaOH pretreatment, green acetylation based on Das et al. [10], washing, drying and the respective inputs (water, electricity and chemicals) and outputs. The other mentioned as "Conventional Approach" (right branch in Figure 5.1), was modeled as a referential scenario since it encompasses the corncob milling, pretreatment and acetylation techniques conventionally applied. The detailed description of the combined LHW-Dilute NaOH pretreatment, the green acetylation and the

standard acetylation have already been presented in previous topics of materials and methods. When it comes to the conventional pretreatment (alkali and bleaching), it was not experimentally performed, but rather modeled according to Silvério et al. [41], Boonsombuti et al. [42] and Rosa et al. [43]. Succinctly, the milled dried corncob was treated with a sodium hydroxide aqueous solution of 2% (w/w) for 2h at 100 °C under magnetic stirring, and then filtered and washed with distilled water. After washing, the alkaline treatment was repeated, and the resulting material was dried overnight at 60 °C in an oven. After this treatment, the fibers were bleached with a solution made up of equal parts (v:v) of acetate buffer (27g NaOH and 75 ml glacial acetic acid, dilute to 1 L of distilled water) and aqueous chlorite (1.7 wt% NaCLO₂ in water). This treatment was performed at 80 °C for 6h. The bleached fibers were filtered, washed with distilled water and dried at 60 °C overnight. After the alkali-bleaching treatment, it was estimated a yield of 40%.

The inventory data related to all process here modeled was sourced from previously publications, life cycle assessment studies and Ecoinvent (version 3.3), as well as calculated experimentally. Inputs or process highlighted with an asterisk in the product system flow chart do not exist in Ecoinvent database, therefore they had to be modeled as individual process in Open LCA, based on background data from previously published studies. All assumptions made, as well as input and output associated with processes in both product systems, are described in detail in Appendix B.

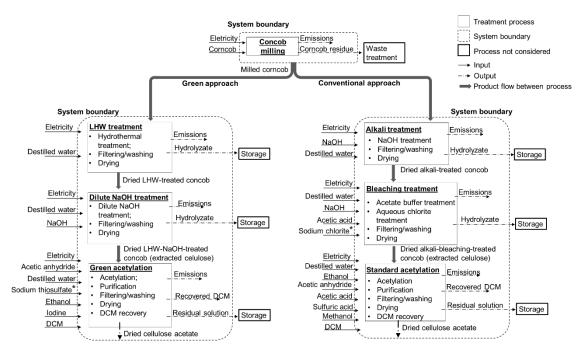


Figure 5-1. Flow chart of product systems: Left branch – modeled green approach scenario; right branch – modeled conventional approach scenario. Inputs highlighted with an asterisk do not exist in Ecoinvent database and had to be modeled in OpenLCA.

5.2.9.4 Impact categories

LCA impact categories, recommended under the International Reference Life Cycle Data System [39], relevant to the work, as already reported in similar studies and in accordance with the ISO 14044 [38], were used. The chosen categories, shown in Table 5.1, are segmented into non-toxicity, toxicity and resource related impact categories. In the absence of normalization references for Brazil, it was used global normalization references provided by Sala et al. [44], for the reference year of 2010. This limitation does not compromise the results (normalized impacts) since, at laboratory activity level, many chemical reagents are produced and supplied by foreigner companies. Moreover, in the absence of regional normalized factors it is preferable to use global factors, as some impact methods (such as CML 2002 and Usetox) present a global scope for the application of their impact categories. By using global values, it is also avoided an imbalance in normalization resulting from comparison with activities of different populations [45].

Table 5.1. Environmental impact categories and normalization references used. CFC: Chlorofluorocarbon, kBq: Kilobecquerel, NMVO: Non-Methane Volatile Organic Compounds; CTU: Comparative Toxicity Unit, e: Ecotoxicity, h: Human.

| Impact category | Abbreviation | Method | Normalization reference | Unit |
|---|--------------|-------------------------------|-------------------------|-------------------------------|
| Non-toxicity related impact categories | | | | |
| Global Warming Potential | GWP | IPCC 2013 | 8400 | kg CO ₂ -eq/person |
| Stratospheric Ozone Depletion | ODP | CML 2001 [46] | 2.34x10 ⁻² | kg CFC11-eq/person |
| Ionizing Radiation – Human Health | IR | ILCD 2016 [47] | 4.220 | kBq-U-235-eq/person |
| Photochemical Ozone Formation | POF | ReCiPe (H) [48] | 40.6 | kg-NMVO-eq/person |
| Freshwater Eutrophication | FE | ReCiPe (H) [48] | 0.734 | Kg P-eq/person |
| Marine Eutrophication | ME | ReCiPe (H) [48] | 28.3 | Kg N-eq/person |
| Terrestrial Acidification | TA | ILCD 2016 [47] | 55.5 | Mol H+-eq/person |
| Terrestrial Eutrophication | TE | ILCD 2016 [47] | 177 | Mol N-eq/person |
| Particulate Matter | PM | Selected LCI results [49] | 2.76 | Kg PM _{2.5} /person |
| Toxicity related impact categories | | | | |
| Human toxicity, Cancer Effect | HTc | USEtox [50] | 3.75x10 ⁻⁵ | CTUh/person |
| Human Toxicity, non-Cancer Effect | HTnc | USEtox [50] | 4.75x10⁴ | CTUh/person |
| Freshwater Ecotoxicity | ET | USEtox [50] | 1.18x10 ⁴ | CTUe/person |
| Resource impact categories | | | | |
| Depletion of Abiotic Resources - Fossil | Rf | Cumulative Exergy Demand [49] | 6.54x10 ⁴ | MJ-eq/person |

5.2.9.5. Sensitivity analysis

The sensitivity of the assessments' approaches was performed via scenario analysis. It was first modeled a scenario named as "intermediate approach" in order to assess the effect of changing the alkali-bleaching pretreatment at the conventional approach by the combined green

pretreatment. On impact categories, this change allows comparing the Influence of the two main processes (pretreatment and acetylation) on both product systems.

Moreover, taking into account the different acetylation yields reported in literature as a result of the application of the standard acetylation [10, 11, 30], improvements over the concerned technique, represented as a percentual increase in weight gain, were assumed. For this, all the process and parameters regarding the green pretreatment were kept unchanged and percentual increments of 10%, 20% and 30% were added upon the yield obtained experimentally for the standard acetylation (in the scope of the conventional approach). These changes gave rise to three scenarios with results presented as impact increment (in percentage) taking as baseline reference the green approach scenario. Through this, it is possible to assess the environmental viability of performing the green approach in face of possible variations in the standard acetylation yield.

5.3. Results

5.3.1. Effect of combined pretreatment on corncob fractionation – compositional analysis

The determined chemical composition of untreated corncob presented hemicellulose (47.32%), lignin (16.32%) and cellulose (36.35%) fractions similar to others in literature [41, 51]. Remarkable changes on chemical composition of corncob were verified after the combined pretreatment. The LHW caused the hydrolyze of hemicellulose and effectively solubilized around 90% of its original content, leading to a final product rich in cellulose and lignin. High levels of hemicellulose solubilization (over 90%) were also reported by Sbanci and Buyukkileci [52] and Michelin and Teixeira [53] in response to LHW treatment of lignocellulosic biomass. In the wake of this treatment, the dilute NaOH solution caused mainly the disruption of the lignin structure through the cleavage of intermolecular α - and β -aryl ether linkages. The complementary effects of these pretreatments on fractionating the biomass contributed positively to obtain a solid cellulose-enriched final product (84.73%), with reasonably low amounts of hemicellulose (reduced by 90%) and lignin (reduced by 34.5%). This combined process converted the milled corncob into a fine, light gray and fluffy powder.

5.3.2. Cellulose acetate syntheses - degree of substitution and yield

The acetylation methods performed in this study differ from each other since the chemical reagents used and the reactions mechanisms are dissimilar. Briefly, the green acetylation runs as

heterogeneous process in which, firstly, the iodine activates the carbonyl carbon of the acetic anhydride in the presence of an alcoholic group. Then, the acetic anhydride reacts with hydroxyl groups of cellulose and the esterification takes place, leading to the generation of acetic acid as a byproduct [10, 31]. Conversely, the standard acetylation occurs as a homogeneous process, set in through sulfate reaction in response to the catalytic effect of H₂SO₄, which helps to dissolve the cellulose pulp in the reaction media (acetic acid). Following, subsequent acetylation via displacement of hydroxyl with acetyl groups by acetic anhydride is triggered [54].

The applied synthesis methods gave different acetylation yields (calculated as dry weight percent gain), calculated as 60% for the green method and 40% for the standard. These results are comparable with others yields derived from the acetylation of cellulose-rich biomasses, such as rice husk (66%) [10], cotton stalk (48.5%) [30], cotton linters (54%) [55] and landscaping waste (45.6%) [9]. These indicate that the optimization of the synthesis using iodine as catalyst in the presence of acetic anhydride, proposed by Das et al. [10] for the acetylation of rice husk, also works out for the acetylation of corncob.

Figure 5.2 shows the ¹H NMR spectra of CA synthesized from green (Figure 5.2a) and standard (Figure 5.2b) acetylation methods. Both spectra display typical resonances of CA chemical structures, with peaks in the region of 3.5-5.3 ppm related to the protons bonded to the seven ring protons of anydroglucose (AGU) (indexes H₁ to H₆ in both graphs). The successful synthesis of CA by both methods was confirmed by the presence of peaks in the region 2.2-1.5 ppm, assigned to the methyl (CH₃) of acetyl groups. The peak around 2.49 and 3.3 are related to the protons of DMSO and residual water, respectively. The DS of CA samples were calculated based on methodology previously described [31]. Both methodologies resulted in synthesized CA with high DS, namely 2.68 and 2.89 obtained by the green and by the standard, respectively. Cellulose acetate with DS varying from 2.5-3, synthesized from standard acetylation, or through few variations of this technique, was also obtained by Jo et al. [28], Coletti et al. [29], Candido et al. [8] and Cao et al. [9]. By varying parameters of the green acetylation method, Das et al. [10] synthesized cellulose acetate with DS ranging from 2.12 to 2.91. These results demonstrate that both methodologies originate CA with similar characteristics and in agreement with other studies.

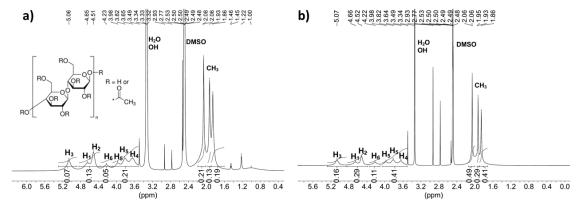


Figure 5-2. H NMR spectra of cellulose acetate synthesized by (a) green and (b) standard acetylation methods.

5.3.3. Fourier Transformed Infrared analysis

The FTIR spectra of cellulose pulp, corncob extracted cellulose (through combined green pretreatment), CA commercial, as well as the CA synthetized from green and standard acetylation methods are presented in Figure 5.3. The absorbed peaks in corncob extracted cellulose around 2890 cm⁻¹ (C-H stretching in cellulose-rich material), 1431 cm⁻¹ (CH₂ banding vibration in cellulose), 1368 cm⁻¹ (C-H asymmetric deformation in cellulose), 1315 cm⁻¹ (CH₂ wagging in cellulose), 1161 (asymmetric C-O-C stretching), 1102 cm⁻¹ (C-OH skeletal vibration), 1028 cm⁻¹ (C-O-C pyranose ring skeletal vibration) and 897 cm⁻¹ (C-H deformation), typical of cellulose-rich materials [17, 56–59], confirm the effectivity of the combined green pretreatment performed in fractionating the corncob and extracting cellulose. Besides, the absence of peaks around 1598, 1513 (aromatic ring stretch ascribed to lignin) and 1250 (C-O stretching in the aryl group of lignin and acetyl group in hemicellulose), related to non-cellulosic polysaccharides [17, 57], highlight the efficiency of the pretreatment in removing lignin and hemicellulose.

The presence of absorbed peaks related to the cellulose chemical structure, along with the rising of new absorption peaks in acetylated samples, namely 1734 cm⁻¹, 1368 cm⁻¹ and 1217 cm⁻¹ related to C=O stretching of ester group, C-H in O-(C=O)-CH₋₃ and C-O stretching of acetyl groups, respectively, confirm the successful acetylation of corncob extracted cellulose. Moreover, the reduced intensity of peaks assigned to -OH stretching also provides evidence of acetylation, since it results from the substitution of -OH groups in cellulose by acetyl groups. Lately, the definitive proof of the successful synthesis of cellulose acetate by both methods is taken from the comparative analysis (peak-by-peak) between the synthesized CA samples and the CA commercial. The absence of absorption peaks around 1700 cm⁻¹ and 1760-1840 cm⁻¹ indicates that the obtained synthesized CA are free of acetic acid and unreacted acetic anhydride [10, 11, 33].

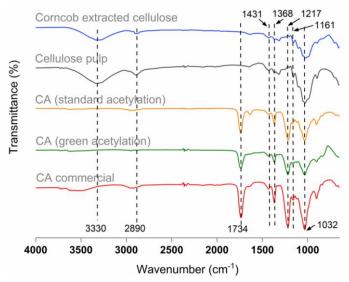


Figure 5-3. FTIR spectra of cellulose pulp, commercial CA, extracted cellulose (LHW-dilute NaOH pretreatment) and CA synthesized by green and standard acetylation.

5.3.4. Thermogravimetric analysis

The TG and DTG curves of all samples are presented in Figure 5.4a and 5.4b, respectively. In general, degradation reactions observed in all curves occurred in a single step and in a narrow temperature range, allowing to infer that all samples are composed mainly by a unique component. The onset (Torona) and end (Torona) temperatures of both synthesized cellulose acetate samples are identical (Torona) and end (Torona) temperatures of both synthesized cellulose acetate samples are identical (Torona) slightly differ, as shown in Figure 5.4 and Table 5.2. Taking into to account the error associated with measurement, these results clearly show that, notwithstanding the different acetylation techniques applied, the thermal stability of synthesized CA samples is very similar. Furthermore, Torona and Torona of synthesized samples are slightly higher than that of the commercial CA, probably due to their higher DS or to the origins of precursor materials. As evidenced by Cao et al. [60] and Morgado and Frolline [61], the thermal stability of CA samples increases as the DS value increases. This can be explained by the fact that, as DS increases more acetyl groups are introduced onto the C2 and C3 carbons of the anhydroglucose unit (AGU), which require more energy, when compared to the least sterically hindered group C6COOCH₃, and increase the apparent activation energy for thermal decomposition, leading to an increase in thermal stability [61].

The most remarkable changes on thermal profiles are verified when comparing the extracted cellulose profiles with the synthesized CA curves. The onset and end temperatures of the extracted cellulose ($T_{onset} = 294$ °C, $T_{end} = 361$ °C) are higher than those of CA obtained by both acetylation methods (Figure 5.4 and Table 5.2). Likewise, the temperature of maximum weight loss rate

related to the extracted cellulose (363 °C) is higher than that of synthesized samples (CA green acetylation – 350 °C; CA standard acetylation – 347 °C). These indicate that the thermal stability of the extracted cellulose decreased due to acetylation, which is in agreement with the literature [60, 62].

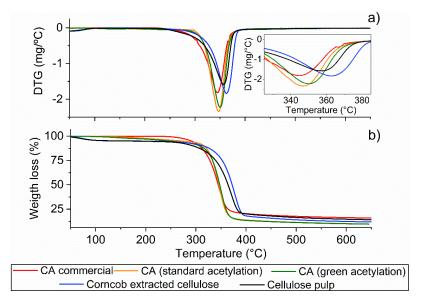


Figure 5-4. (a) TG and (b) DTG (as well as inset plot in the graph 4b of DTG ranging from 330 °C to 390 °C) curves of cellulose pulp, commercial CA, extracted cellulose (LHW-dilute NaOH pretreatment) and CA synthesized by green and standard acetylation.

Table 5.2. Thermogravimetric results for cellulose pulp, commercial CA, extracted cellulose (LHW-dilute NaOH pretreatment) and CA synthesized by green and standard acetylation. Tonset temperature, Tonset temperature of maximum weight loss rate, Tonset temperature.

| Sample | DS | T _{onset} (°C) | T _{max} (°C) | T _{end} (°C) |
|-----------------------------|------|-------------------------|-----------------------|-----------------------|
| CA commercial | 2.4 | 310 | 345 | 361 |
| Cellulose pulp | 0 | 327 | 357 | 383 |
| Corncob extracted cellulose | 0 | 242 | 363 | 392 |
| CA (green acetylation) | 2.68 | 320 | 350 | 366 |
| CA (standard acetylation) | 2.86 | 320 | 347 | 366 |

The thermogravimetry analysis, along with the previous chemical characterization, were performed first and foremost to investigate the properties of cellulose acetate produced from both synthesis route and assess their suitability to be compared via LCA, which was successfully confirmed. Upon reaching this target of the work, the next following two topics will be mostly dedicated to the elucidations about the potential application of the CA produced by the green approach as a film forming.

5.3.5. Scanning Electron Microscopy Analysis

A thin film of cellulose acetate synthesized by from the green approach was produced in order to demonstrate the potential of this material for industrial applications. SEM images of the raw corncob, LHW-NaOH pretreated corncob, as well as the cellulose acetate film surface and cross-section are presented in Figure 5.5.

As a result of the combined green pretreatment, the rigid and layered assembled structure of the corncob (Figure 5.5a) was unpacked and a remarkable disaggregation of fibers is noticed (Figure 5.5b). After the acetylation of extracted cellulose and subsequent dissolution, the quite smooth, flat and homogeneous surface of the produced film (Figure 5.5c and 5.5d) hardly assemble the disparate fibrous shapes of the extracted cellulose (Figure 5.5b). This indicates that the acetylation reaction successfully occurred, and reagents used could penetrate into the cellulose fibers. The few porous and rough surface in the cross section and film surface may be attributed to the solvent evaporation process.

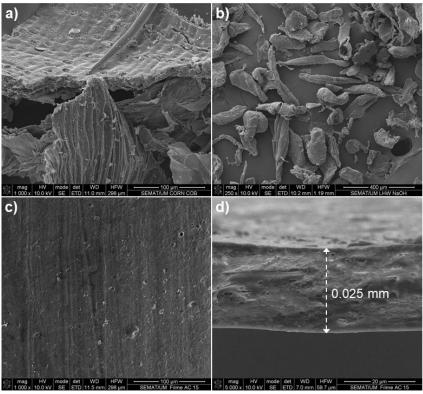


Figure 5-5. SEM micrographs of (a) untreated corncob, (b) LHW-dilute NaOH pretreated corncob, as well as the (c) surface and (d) cross-section of the cellulose acetate film.

5.3.6. Dynamic mechanical analysis

The viscoelastic behavior of the produced film was Investigated by dynamic mechanical analysis (Figure 5.6a). The acetylation reaction, as well as the cellulose acetate film prepared, are depicted in Figure 5.6b. The viscoelastic properties of the film were characterized by the storage modulus (a measure of the material's ability to recover from deformation), the loss modulus (a measure of the material's ability to loses energy as heat) and the Tan δ or damping (an indicator of how efficiently the material loses energy to molecular rearrangement and internal friction). The behavior of E' and E'' in Figure 5.6a is typical of amorphous thermoplastics, as no defined transition zones are identified, but rather a continuous decrease along the temperature range. This can be associated to the introduction of acetyl groups onto the AGU unity, which prevent the packaging of cellulose chains and the resultant polymer becomes more amorphous and less opaque [63]. At lower temperatures, despite the presence of a secondary relaxation, the molecules are reasonable immobile and remain in energy elastic state, which was characterized by a drop in E' of about 0.8 GPa over the temperature range 40-138 °C. This is followed by an intermediate state, the glass transition, characterized by a sharp drop in E' of about 0.8 GPa in a short temperature range 138-152 °C, and further by an entropy elastic state at elevated temperatures. For this class of material, the operating range is usually defined by the glass transition temperature, which was found to be 140 °C, approximately, using the peak of tan δ curve method. Similar T_{ϵ} values of other cellulose acetate films were reported by de Freitas et al. [63], Bao et al. [64], Abdel-Naby et al. [65] and Számel et al. [66]. However, the T_g found may differ from others in literature since this property is intrinsically related to the DS, and higher DS leads to a lower T_g.

The E'' reaches the maximum dissipated energy at 136 °C, which means that the mechanical deformation applied is converted into the maximum internal friction and nonelastic deformation. From this point until 160 °C the storage modulus drops roughly one decade. A broad secondary peak in E'' is also verified at about 64 °C. In response there is a decrease in E' from 40 °C to 100 °C, probably originated from the natural mobility of chains in response to temperature rising and relaxation transitions triggered by local mobility of chains plasticized with water [67].

The storage modulus of the produced film, at 50 °C, is 1.88 GPa, which is higher than the E' value of a commercial one (1.58 GPa) with DS of 2.48, as reported by de Freitas et al. [63]. Furthermore, at a temperature close to 160 °C the cellulose acetate film produced presents a rather significant E', with value around 178 MPa, demonstrating excellent processing conditions and high resistance. Despite the dynamic Young's modulus calculate from DMA is not exactly the

same as Young's modulus (elastic modulus) resultant from the classic stress-strain curve, they are ideally equivalent so that their comparison does not commit the interpretation of results. From literature, commercial cellulose acetate is known to have elastic modulus varying from 600-3000 MPa, depending on the DS and the average molecular weight [65]. By applying Equation 2, at 40 °C the E* of the produced film is equivalent to 1.89 GPa, which is consistent with the previously mentioned range of values. The similar values of E' and E* come from the prevailing elastic property of the film. Notwithstanding, at 40 °C the material runs through an increase of the atomic vibration yet, which, in turn, may decrease the stress needed to produce a given strain. Thus, these results suggest that the cellulose acetate film produced via a green approach, using corncob residue as a material source, could be potentially used in value-added applications, such as packaging.

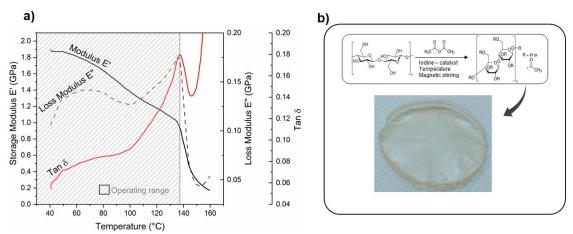


Figure 5-6. (a) Dynamic mechanical analysis of cellulose acetate film produced and (b) cellulose acetate film.

5.3.7. Life Cycle assessment

The overall environmental impact categories of conventional and green approaches, considering the functional unit of 10 g of synthesized cellulose acetate, are compared in Figure 5.7a. The LCA results show that, except for ET, the proposed green approach exhibited lower environmental impacts for all categories. This is the result of the lowest energy consumption and chemical throughout the green pretreatment and acetylation steps, which, in percentage terms, resonated in a reduction of impacts varying from 4% (fossil resources depletion) to 19% (ionizing radiation). The higher ET impact associated with the green approach were caused mainly by copper emissions involved in the production of acetic anhydride. In the green acetylation method this reagent is used in amount over four times greater than the amount deployed in the standard acetylation.

The acetylation processes were the main contributors to environmental impacts in both approaches, mainly due to their energy-demanding characteristic and chemicals used. The

following aspects/process were identified as having major influences on environmental impacts of conventional approach: electricity consumption (related to the standard acetylation process) > alkali and bleaching treatment (chemicals used and electricity consumption) > dichloromethane consumption (purification step) > ethanol consumption. The proportional contribution of individual processes associated with the green approach can be found in Figure 5.7b. The overall impact analysis for this approach also shows that the electricity consumption related to the acetylation process is the main contributor to environmental burdens, followed by the chemicals used throughout the acetylation (being the ethanol and acetic anhydride consumption the most prominent contributors), dichloromethane consumption in the purification step and, lastly, the combined green pretreatment. The proportional impacts of corncob milling were also accounted, but its contribution to the global impacts was too low to be evidenced in Figure 5.7b. As the precedence of aspects/process for both approaches were drawn up on the basis of an overall LCA analysis, they might slightly vary in some impact categories, though. By comparing both approaches within the order of precedence dictated by the process/aspects' contribution impacts, it is possible to verify that, not surprisingly, significant environmental savings could be achieved through the reduction on electricity and chemicals in the pretreatment stage. The effective savings provided by the green pretreatment are discussed in more detail further in the sensitivity analysis. Furthermore, it is interesting to note that the combined green pretreatment contributed more than 20% only for three impact categories, namely IR, FE and PM, (Figure 5.7b). These categories are particularly affected by impacts related to energy generation, which coherently agree with the energy-demanding characteristic of the LHW treatment. Comparatively to the combined green pretreatment, the proportional contribution of the alkali-bleaching pretreatment on each impact categories is substantially greater. For the same impact categories referred previously (IR, FE an PM) the conventional pretreatment accounted roughly for 33%, 29% and 29% respectively of their total impacts.

Besides, in previous studies Boonterm et al. [25] and Dedpakdee et al. [68] have shown that, in comparison to hydrothermal treatments, concentrate NaOH solution can lead to higher potential impacts on toxicity (HTc, HTnc and ET). Consistently with these findings, the dilute NaOH solution here applied reduced significantly impacts in the context of toxicity, and showed lower impacts on HTc, HTnc and ET than that of the LHW treatment by 43%, 42% and 43%, respectively. Moreover, within the framework of the conventional approach, the alkali-bleaching pretreatment had relevant

influence over these ecotoxicity categories and affected the impacts on HTc and ET by 20% and 26%, respectively.

It is worth mentioning that, within the acetylation process are also computed the impacts related to the recovery of dichloromethane. At the laboratory scale level its recovery was possible and can represent an overriding strategy in view of environmental gains. For instance, the primary consumption of dichloromethane leads to a significant impact share for ODP, HTc, GWP, ME and TE. Particularly for HTc and ODP, classified as some of the largest impact categories when expressed in person-equivalents (Figure 5.8), the primary dichloromethane consumption contributed roughly with 33 and 98%, respectively, in both approaches. Assuming a recovery rate of dichloromethane at about 80% (estimated experimentally in laboratory) when performing the green approach processing, the savings for HTc could reach 27%. In addition, by using recovered dichloromethane ultimately reduces atmospheric emissions of carbon dioxide and methane originated from its production, which also substantially imparts environmental negative impacts on the GWP and POF. Therefore, the recovery and recyclability of dichloromethane is an effective way to reduce the environmental footprint of subsequent acetylation cycles.

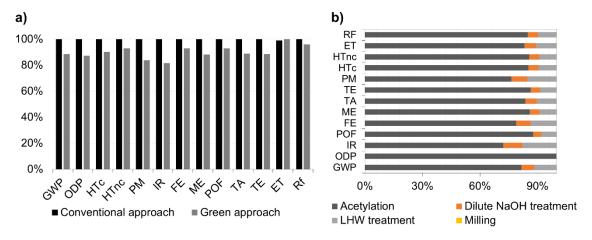


Figure 5-7. (a) Potential environmental impacts of cellulose acetate synthesized by green and conventional approaches and (b) proportional contributions of green approach processing steps on each environmental impact category.

By dividing the LCA results of each impact categories by their correspondent normalization reference, their characterized values are adjusted to a notionally common scale – person-equivalents (PE) – that means the fraction of the contribution to the concerned impact deriving from the average person in the reference geographical area. Thus, these normalized values can give information on impacts' magnitude and enable comparison of different impact categories. From Figure 5.8, it is clear that the magnitude of HTc is much higher than that of the other

categories. The main aspect enclosed to potential impacts over human toxicity (with carcinogenic effects) deals with the emission at long-term of chromium VI to groundwater, resulting from the landfilling of residual materials originated in dichloromethane and electricity production. In the conventional approach this impact is compounded by the effects associated with the pretreatment used. The large magnitude of impact also seen on freshwater eutrophication is mainly affected by the emission at long-term of phosphate resulting from the landfilling of residual material originated from fossil fuel energy production and chemicals production. Especially for the conventional approach, it is also resultant from the pretreatment step and chemicals used in the acetylation (namely acetic acid and anhydride acetic). Taking into account also the marked magnitude of HTnc, it is reasonable to infer that both processing approaches contributed the most to toxicity related impact categories. More precise information on this regard could be obtained by performing weighting, but this step is not in the scope of the study. By contrast, the lower magnitude of categories IR, ME and TE, in both approaches, are partially raised from the processing design and requirements in its core, as well as from the system boundary assumed, which ultimately limited the inputs and consequential outputs linked to the occurrence of environmental aspects with repercussion on such impacts.

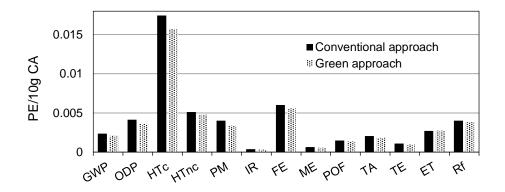


Figure 5-8. Potential impact contribution in PE/10g of cellulose acetate of the green and conventional approach.

5.3.7.1. Sensitivity Analysis

A sensitivity analysis based on comparison of scenarios was conducted to investigate the specific contribution of acetylation and pretreatment process over the impact categories. For this, an additional scenario, referred to as intermediate approach, was modeled by setting up the combined green pretreatment preceding the standard acetylation. The following discussion will be carried out taking as reference the green approach impacts, since it sets a lower limit for all impact categories, except for ET. From Figure 5.9 it is noticeable that the enlargement of impacts over

ODP, TA, TE and Rf are independent of the pretreatment applied, indicating that they come almost entirely from the standard acetylation process. Conversely, the alkali-bleaching pretreatment step was the main contributor to increase the HTnc and FE impacts. For all other impact categories, the extent of impacts is almost equally provided by both process (conventional pretreatment and standard acetylation), other than for a slightly superior contribution assigned to the standard acetylation. The findings drawn from this analysis support the previous discussion and makes clear that the application of green pretreatments is an obvious way to reduce negative environmental impacts, although its gains may not extend to all impact categories. Additional savings can be achieved by reducing the use of chemicals, as afforded by the green acetylation, or through recovery of solvents. Last, but certainly not least, reducing energy consumption might be the most effective way to reduce the environmental footprint of the concerned process. However, this has been reported to be a challenge when it comes to biomass valorization [25, 26, 69], and the most obvious way to manage this bottleneck seems to be the use of integrated renewable energy sources, as suggested by Yates et al. [70].

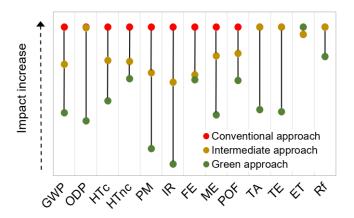


Figure 5-9. Influence of pretreatments and acetylation techniques on environmental impacts. Comparison of reference scenarios with Modeled intermediate one constituted by LHW-dilute NaOH pretreatment and acetic acid acetylation process.

The sensitivity of the LCA results regarding the overall advantages of the green approach in comparison to the conventional approach was also tested via a different perspective. As stated previously, the yield of green acetylation is 12.5% greater than that of standard acetylation. Considering that both approaches are assessed under the same functional unit, this yield gap has a direct influence on the amount of raw material (corncob) to be processed and consequently on the number of chemicals and electricity consumed, which means that the lower the yield, the greater the need for inputs. However, the yield of the standard acetylation is susceptible to optimization by slightly changing the reaction parameters, as reported by Bello et al. [30]. Thus,

potential perspective scenarios for the conventional approach were modeled by varying its acetylation weight gain from 40% to 70%, and named as WG-40%, WG-50% and so on, as presented in Figure 5.10. By conducting this assessment is possible to ascertain to what extent the green approach (baseline scenario) remains environmentally viable in face of a possible optimization of the standard acetylation processing.

The LCA sensitivity analysis for each impact category associated with all modeled scenarios is reported in Figure 5.10. Taking as baseline scenario the green approach, by comparison, a positive impact increment indicates a negative environmental effect, while a negative impact increment indicates a positive effect. Results indicate that, although both processes have a quite similar environmental footprint in the assumption of same yield (WG-60%), the performance of the conventional approach only exhibits overall positive environmental effects for weight gains $\geq 70\%$. Along with the results previously discussed, this analysis clearly indicates that, in comparison to the standard acetylation synthesis, the key factors to lower the environmental impacts resulting from the green acetylation are the high yield obtained and not use acetic acid as reaction media. The attempts made in previous works to substitute the traditional catalyst H_2SO_4 by eco-friendly ones, could not be deeply investigated here since, based on the FU adopted, this chemical was used in little quantity which could have masked its pollution effects.

Nevertheless, even with a weight gain of 70% the conventional approach still results in superior environmental burdens for IR and PM, and notwithstanding the prospective analysis, the green acetylation may also be subject of optimization, as shown by Das et al. [10], who successfully varied the acetylation yield as a function of the acetylation time and iodine amount. It is also important to bear in mind that environmental impacts related to the green pretreatment would probably be reduced when scaling-up this processing step, mainly due to the higher cellulosic fiber yielding produced per electricity consumed [25].

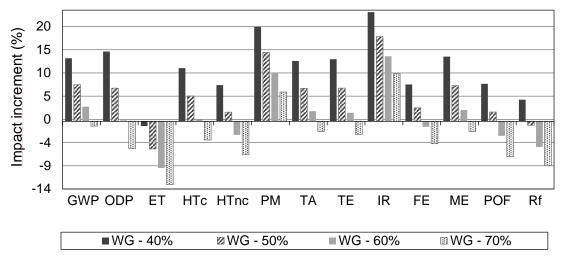


Figure 5-10. Potential impact differences in all categories due to possible changes in the acetylation yield of the conventional approach. Impact increment was calculated taking as baseline scenario the green approach. WG means weight gain.

4. Conclusions

Cellulose was successfully extracted and cellulose acetate was synthesized from corncob by applying a green approach, constituted by a combined LHW-dilute NaOH pretreatment and iodinecatalyzed acetylation free of solvent. The outstanding physicochemical properties of the synthesized cellulose acetate endorse its use as a value-added product, which was proved by the preparation of a thin film with remarkable mechanical properties. The effective fractionation and cellulose extraction capacity of the pretreatment was confirmed via chemical composition analysis and backed up by the high DS (2.68) and yield (60%) obtained after the acetylation. Along with the high yield of acetylation, the less use of chemical reagents associated with the green approach were key aspects within its environmental benefits. In comparison with a current conventional approach, the environmental sustainability of the proposed green route was confirmed through life cycle assessment. Overall LCA results revealed that the proposed green approach exhibited lower environmental impacts than the conventional approach for all categories, except for ET. Moreover, in both approaches the acetylation process accounted for the main share of impacts, which was in part due to the length of the process and the consequent amount of electricity required. In general, the most obvious way to improve even more the proposed processing design relies on the use of integrated renewable energy sources and recycling of solvents, namely DCM. Although there are few studies regarding these addressed subject for comparison purpose, the data used fulfilled the requirements of the LCA and the obtained outcome is reliable.

Acknowledgements

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Chapter 6. A novel eco-friendly synthesis processing to produce cellulose acetate/ TiO₂/MgO bionanocomposite films

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Abstract

This study aimed at developing a novel approach to produce a bionanocomposite film by applying green raw materials and eco-friendly processing techniques. Cellulose acetate emerged from agroindustrial residue was used as matrix and metal oxide nanofillers were produced by the pulsed laser ablation in liquid (PLAL) of bare titanium and magnesium targets. Green pretreatment and derivatization techniques were performed to extract and acetylate cellulose, respectively. Subsequently, the synthesized cellulose acetate was directed dissolved in the nanoparticles' colloidal suspension, and the nanocomposite film was produced via solvent casting method. Physicochemical characterization techniques were undertaken to investigate the effectivity of the nanoparticles synthesis, biomass pretreatment, acetylation, as well as the biofilm properties. Viscoelastic properties of the film were also assessed via dynamic mechanical analysis. A solid enrich-cellulose fraction could be obtained after the applied pretreatment, which reflected in the successful high-yield acetylation synthesis. Metal oxides nanoparticles with relatively large sizes could be easily and rapidly synthesized through PLAL. Formation of the metal oxidized phases, as well as their embedment into the film, were clearly identified by UV-vis spectrometry, energy dispersive X-ray spectroscopy and X-ray diffraction. In comparison to the neat cellulose acetate film, the introduction of nanoparticles increased the porous structures of the nanocomposite, which in turn decreased its stiffness, by making it more flexible. Besides, the glass transition temperature (T_e) of the composite film increased significantly, possibly due to the restrictive effects of the TiO₂/MgO nanoparticles on the segmental mobility of the cellulose acetate chains. By increasing the T_s, the operating range of the material was also extended, which is an especially important property within industrial processing requirements. Thus, the present work shows a facile and effective eco-friendly approach to produce bio-based nanocomposite films with potential active properties.

Keywords: bio-based polymer; nanocomposite; ecofriendly techniques; lignocellulosic biomass; agroindustrial residues; cellulose acetate; pulsed laser ablation.

6.1. Introduction

For more than 60 years the global production of plastics has shown an upward trend, reaching an astonishing value of 348 million tonnes in 2017 [1]. From this total, the polyolefins accounted for more than 50%, which were mostly aimed at the production of single used packaging. Besides

being petroleum-based dependents, these synthetic plastics are nonbiodegradable and can threats the environment through accumulation in landfills, oceans and other biocompartments. In the past few decades, advances in chemical processing and biotechnology have allowed the production of commercially viable bio-based and biodegradable polymers, such as polylactic acid, polyhidroxyalconoatos (PHA), starch blends and bio-based PET and polyolefins [2, 3]. However, although the production capacity of these bio-based alternatives has shown a positive growth rate over the last years, they are still expensive in comparison to the synthetic polymers (mainly because of the low crude oil price) and may compete with food production as they emerge from first-generation feedstock.

Alternatively, increasing interest has been devoted to developing packaging made of biodegradable polymers from second-generation renewable resources, namely lignocellulosic residues. Lignocellulosic biomass is the most economical and abundantly renewable source of biopolymers on earth. Annually, huge amounts of lignocellulose are available as non-edible agroindustrial residues in the form of stalk, stover, branches, leaves, straw, bagasse, husk and pells, which are mainly composed by cellulose (50%), hemicellulose (20-40%) and lignin (10-40%) [4, 5]. The valorization of these residues, besides not generating competition with food production, can still prevent their insertion into the conventional solid waste disposal paths and may promote the circular economy.

Highlighted as one of the word's largest producers of agricultural commodities, Brazil also generates fair amounts of agroindustrial residues. For instance, the country is the third-largest producer of corn and, consequently, it is estimated that impressive amounts of residues in the form of corncob (18700 ktons), husk (12470 ktons) and stover (111109 ktons) are also generated annually [5]. Corncob is the central part of the ear of maize, chemically constituted mainly by cellulose and hemicellulose [6]. Cellulose, the most abundant organic compound on earth, is a high molecular weight homopolymer generated from repeating D-glucopyranose ring units linked by β -1,4-glycosidic bonds (cellobiose unit). Many studies have already reported the potential application of cellulose extracted from agroindustrial residues for the development of biodegradable food packaging, e.g. Silvério et al. [7] – corncob, Reddy et al. [8] - Borassus fruit fibers, Vanitjinda et al. [9]– sugarcane bagasse, Zhang et al. [10] – corn husk and Otoni et al. [11] – carrot processing waste. Notwithstanding, because of the intra and intermolecular interactions of the hydroxyl groups present in the cellobiose unit, the cellulose molecules crystallize in a horizontal plane and in parallel chains, forming microfiber packages, which leads to some properties that

limits its application, namely lack of antimicrobial properties, high hydrophilicity, poor solubility in common solvents and low dimensional stability and thermoplasticity. To overcome these drawbacks and widen its manufacturing potential into value-added materials, the cellulose is usually processed into derivatives. Among the existent derivatizing techniques, the esterification of cellulose into cellulose acetate is the most commonly applied approach. In addition to its physical-chemical properties, cellulose acetate is biodegradable, and therefore it is an excellent choice for film applications as food packaging [12].

The development of cost-effective and sustainable methods to produce cellulose-based food packaging from residues is still challenging and the most innovative products available in the market are still based on high-quality cellulose from wood pulp or cotton. Furthermore, in addition to performing its essential function (i.e., to ensure the food safety and quality, as well as prevent environmental degradation) throughout the food distribution and storage chain, the development of new bio-based materials for packaging still have to meet market and consumers requirements, which currently are turned towards materials with eco-friendly and active properties [13–15].

Recently, nanotechnology has received enormous attention from researches to develop active bio-based hybrid materials with antimicrobial properties. Among the most focused nanoparticles applied for antibacterial purposes are the metals end metal oxides, such as Ag, TiO₂, CuO, ZnO and MgO [14, 16, 17]. Nevertheless, as an emerging application and research field, there are concerns related to the migration of nanoparticles to food and their toxicity [14]. In this regard, the inorganic metal oxides TiO₂ and MgO are of particular interest because they are safe for human beings, in addition to be stable under harsh processing conditions [18, 19]. On top of that, the use of TiO₂ and MgO as additive in food is allowed and regulated by food safety authorities, such as Food and Drug Administration (USA) and the European Food Safety Authority.

The use of magnesium oxide and hydroxide (Mg(OH)₂) nanoparticles have attracted a great deal of interest because of their properties, including nontoxicity, antibacterial activity and thermal stability [19, 20]. TiO₂ is among the most explored nanomaterials in active food packaging, and it is also considered a promising alternative since it can enhance photo-degradability and exhibit antimicrobial and ethylene photodegradation activity when exposed to UV light [14, 21–23]. Among the possible multifunctionalities imparted by these nanomaterials, their complementary antimicrobial effect is of particular interest since antimicrobial packaging is one of the most pursued properties of active packaging technologies [24]. While the individual antimicrobial effect of TiO₂ nanoparticles as a component of polymeric packaging are well documented [21, 23, 25, 26], the

assessment of MgO nanoparticles [16] and their combined application [27] are limited. Synthesis methods in the scope of the wet chemical routes (such as solvothermal, precipitation and sol-gel) are the main approaches used to obtaining metal oxide nanoparticles. They can involve different precursors (e.g. Ti(OCH(CH₃)2)₄), Ti(OBU)₄, thiobenzoate complex with Ti, MgCl₂, MgSO₄ and alkoxides metals), solvent medium (e.g. Lauryl alcohol + hexane, toluene, ethanol, benzyl alcohol, NaOH, sorbitol and urea) and surfactants (e.g. PEG, SDS, oleic acid and ammonia) [20, 28-32]. Although wet chemical approaches are generally scalable and low cost, the use of some chemical agents, as well as the possibility of longer reaction times and high temperature and pressure processing, can lead to negative environment effects and biological toxicity [33, 34]. Besides, chemical synthesis routes are usually complicated by impurities from precursors and incomplete conversion into uniform particles [33]. A promising, well-established and environmentally friendly physical synthesis route to produce metal oxide nanoparticles, which may efficiently overcome some of the problems arising from chemical routes, is the pulsed laser ablation in liquid. In this method, a target material immersed in a liquid medium is irradiated by an ultra-short laser pulse, which causes the removal of material from target and formation of nanoparticles through nucleation and growth [35, 36]. The production of high-purity metal oxide nanoparticles with well-defined properties have been addressed by several reports [33, 37-41].

To date, there are few reports on the use of second-generation renewable sources along with environmentally friendly processing routes to produce potential active food packaging. Therefore, the present study aimed to develop cellulose acetate/TiO₂/MgO bio-based composite film by means of a novel route, comprised of simple and environmentally friendly processing steps (from cellulose extraction to film preparation/formation). The processing steps encompass: (i) the performance of a combine liquid hot water-dilute sodium hydroxide pretreatment (LHW-dilute NaOH) to extract cellulose from corncob; (II) a solvent-free cellulose esterification to produce cellulose acetate and; (III) bionanocomposite film production through the direct dissolution of the synthesized cellulose acetate in a TiO2/MgO colloidal solution prepared by PLAL, followed solvent casting.

Nanoparticles characterization was accomplished by scanning electron microscopy (STEM), energy dispersive X-ray spectroscopy (EDS) and ultraviolet-visible spectroscopy (UV-vis), while essential properties of the developed TiO₂/MgO biofilm were investigated by Fourier transform infrared spectroscopy, X-Ray diffraction, SEM, EDS, thermal characterization and dynamic mechanical analysis.

6.2. Materials and methods

6.2.1. Materials

Corncob was obtained from local agricultural cooperatives from the North of Portugal. The corncob residue was firstly ground and sieved to particles sizes from 0.25 to 0.5 mm and stored in sealed plastic bags at room temperature. Cellulose acetate (Mn₂50.000 by GPC) and Sodium hydroxide pellets were acquired from Sigma-Aldrich. Acetic anhydride (>99%) and sodium thiosulfate (98.50%) were purchased from Acros Organics. Iodine (99.80%), acetic acid glacial (99.88%), sulfuric acid (>95%), ethanol (99.80%), methanol (99.99%), dichloromethane (99.88%) and dimethyl sulfoxide (99.99%) were purchase from Fisher Chemical. All the reactants were used without further purification.

6.2.2. Isolation of cellulose from corncob

Cellulose extraction from dried milled corncob was carried out by applying a combined green pretreatment previously reported by our research group [42]. Firstly, LHW treatment with a feedstock loading rate of 10% (residue weight / distilled water volume) was pretreated at 190 °C, 0.79 MPa, for 30 min (after the heat-up time) under constant mechanical stirring. After the required time, the reactor was cooled and the insoluble solid was collected and thoroughly washed with deionized water and dried at 60 °C. The dried liquid hot water-treated corncob (LHW - CC) was then soaked in 2 wt% NaOH at 90 °C for 1.5 h at a 1/30 solid/liquid ratio (w/w) under magnetic stirring. The alkali-treated LHW-CC (NaOH-LHW-CC) was filtered and washed with deionized water until colorless. The resulting material was dried at 60 °C and stored in a sealed plastic bag.

6.2.3. Synthesis of cellulose acetate

Cellulose acetate was synthesized based on the method described by Das et al. [43] with some modifications. Basically, a mixture of 0.4 g of cellulose (extracted from corncob), 20 mL of acetic anhydride and 0.6 g of iodine was heated to 80 °C and left reacting for 5 hours. After that, the solution was allowed to cool at room temperature and treated with 10 mL saturated solution of sodium thiosulfate with stirring. Then, 60 mL of ethanol was added to the mixture and stirred for 60 min. The product was filtered and thoroughly washed with 75% (v/v) ethanol and distilled water to remove the unreacted acetic acid and byproducts. The solid material obtained was dried at 60

°C in an oven and further dissolved in dichloromethane and filtered. After evaporating the filtrate, the cellulose acetate was collected and dried at 60 °C in a vacuum oven overnight.

6.2.4. Production of Nanoparticles/colloidal solution

The synthesis of TiO2 nanoparticles was performed using a 1064 nm high power Nd:YAG laser (OEM Plus, Italy) with an output power of 6W, spot size of 3 mm, pulse width ~35 ns and operated at the repetition rate of 20 kHz. The target metal plate was fixed on the bottom of a glass vessel filled with 35 ml liquid and a level of liquid/air interface above the targeted substrate of about 1.5 mm in height. Irradiation was carried out with a power of 0.3 mJ/pulse through focusing a lens with focal length of 160mm. During the ablation, the liquid was stirred to keep the ablated particles out of the beam path. Besides, the scanning of the target surface was carried out by means of a XY translation stage to ensure that ablation occurred on an unaffected area.

The experiments were performed in pure dichloromethane as ambient liquid solution regarding its suitability to dissolve the synthesized CA. Firstly, the titanium plate was irradiated by a laser beam with fluence of 4244.0 J/cm² during 1.5 h. After the required time has elapsed, the vessel containing the titanium plate and the NPs solution was taken out, and another vessel containing a magnesium plate was placed. The ablation processing was performed with unchanged conditions. It is worth mentioning that both metal targets contained a natural oxide surface layer. After the required processing time, the ablation process was interrupted, the TiO₂ and MgO colloidal solutions were stored for further analysis and dissolution processing.

6.2.5. Cellulose acetate/TiO₂/MgO nanocomposite film preparation

After the ablation process is over, both colloidal solutions were mixed and the synthesized cellulose acetate was added into the solution containing DCM/metal oxides nanoparticles. This solution was kept under stirring for 4h at 35 °C. A schematic representation of the proposed process is depicted in Figure 6.1. Succeeding complete dissolution, the mixture was sonicated for 10 min, in order to remove air bubbles, poured on a Teflon dish with 4 cm of diameter and placed in the vacuum oven overnight to allow the solvent to evaporate. The bionanocomposite films produced were stored in sealed plastic bags for further analysis.

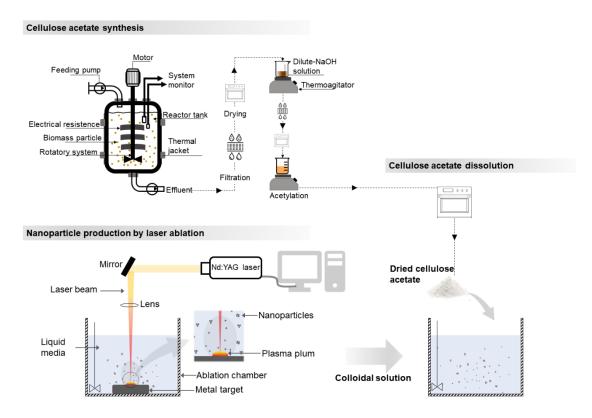


Figure 6-1. Schematic representation of the eco-friendly approach to produce bionanocomposite films.

6.2.6. Material characterization

6.2.6.1. Nanoparticles characterization

TiO₂ and MgO colloidal suspensions were characterized by optical absorbance spectroscopy and scanning electron with transmission electron detection (STEM). Optical absorption spectra of the mixed colloidal suspensions were recorded in the range from 225 to 800 nm of wavelength by using an UV–vis absorption spectrophotometer Model 2501 PC, Shimadzu. Observation of colloidal particles was performed by STEM (FEI Nova 200 Nano SEM) operated at 20 Kv. For STEM characterization, samples were prepared by adding droplets of the colloidal solutions on carbon-coated copper grids.

6.2.6.2. Chemical composition of untreated and treated corncob

The chemical composition of untreated and treated corncob was determined by deconvolution of the derivative curves obtained from the thermogravimetry analysis [44–48]. Briefly, the decomposition processes of the three pseudo-components presented in the lignocellulosic biomass (cellulose, hemicellulose and lignin) were modeled by a Gaussian distribution, and the percentage of each component was assumed to correspond to the integrated area above each single

deconvoluted reaction curve (the modeled equation, as well as graphs and parameters are available in the Appendix A).

6.2.6.3. Fourier transform infrared spectroscopy (FTIR)

FTIR spectra of raw corncob, pretreated corncob and cellulose acetate samples were recorded on a Jasco FT/IR-4100 spectrometer, in the wavelength range of 400 – 4000 cm⁻¹ and at a resolution of 4 cm⁻¹, using the attenuated total reflection technique.

6.2.6.4. X-ray diffraction analysis

The X-ray diffractograms patterns of all samples were obtained at room temperature using a diffractometer Bruker D8 Discover, operated with Cu-K α radiation (wavelength of 0.154 nm) at a power of 40 mA and accelerating voltage of 40 kV. Diffraction intensities were scanned in the range of $2\theta = 5 - 50^{\circ}$ with a scanning rate of 0.04°/second.

6.2.6.5. ¹HNMR spectra

The ¹HNMR spectroscopy was performed in a Bruker Avance III, 400 MHz, and deutered dimethyl sulfoxide was used as solvent. The spectra were internally referenced to the residual proton tetramethylsilane (TMS, δ = 0.00 ppm) and the chemical shifts (δ) are reported as part per million (ppm). Based on Biswas et al. [49], the degree of substitution of acetyl groups was calculated by dividing 1/3 of the three methyl proton absorbance peak area of acetyl group, in the range of 1.5-2.2, by 1/7 of the seven anhydrocellulose absorbance peak area, in the range of 3.5-5.2 ppm.

6.2.6.6. Morphological analysis of the films

The surface and cross-section morphologies of untreated and treated samples were examined by scanning electron microscopy (SEM) using a FEI Nova 200 equipped with energy-dispersive X-ray (EDS) analyzer. For SEM analysis, samples were previously sputter coated with gold.

6.2.6.7. Thermogravimetry analysis

Thermogravimetry analyses of untreated corncob, extracted cellulose and CA samples were performed on TA Instruments SDT 2960 model. The sample weight was in the range of 3-5 mg and heated from 30 to 700 °C at a rate of 10°C/min, under purified argon to prevent thermo-oxidative degradation.

6.2.7. Dynamic mechanical analysis of the bionanocomposite film

Dynamic mechanical analysis was performed on a PerkinElmer DMA 800 using a dual-cantilever bending mode at a constant frequency. The viscoelastic properties were characterized based on the storage modulus (E'), the loss modulus (E") and the damping capacity (tan δ), the later calculated as the ratio of the loss to the storage modulus. CA Films with 20/5.7/0.04 mm of dimensions, approximately, were analyzed under temperature ranging from 40 – 160°C, heating rate of 5 °C/min, preload 1N, 10 μ m of displacement amplitude and 1 Hz of frequency.

6.3. Results

6.3.1. Cellulose extraction from corncob

In substitution of the conventional cellulose extraction technique typically applied for this purpose (namely alkali-bleaching treatment), a less polluting combined pretreatment reported by Araújo et al. [44], performed as a two-stage LHW/dilute-NaOH process, was applied. The removal of non-cellulosic constituents was visually stressed by physical changes after each stage of the pretreatment, which converted the corncob into a fine, light gray and fluffy powder. The LHW treatment effectively solubilized around 90% of the initial hemicellulose content. In turn, the dilute-alkali treatment selectively removed a great share of lignin (leading to a lignin reduction by 50%). By partially disrupting the lignin-hemicellulose complex, the dilute-NaOH treatment could also remove hemicellulose, but at a lower proportion. Consequently, in comparison to the chemical composition of raw corncob (cellulose – 36.35%, hemicellulose – 47.32% and lignin – 16.32), the pretreated sample had a higher cellulose content (84.73%) while the hemicellulose (4.58%) and lignin (10.68%) content were lower.

6.3.2. Cellulose acetate synthesis

The extracted cellulose was acetylated by a free-solvent iodine-catalyzed method, reported by Das et al. [43] as an eco-friendly technique. Under the experimental conditions applied, an acetylation yield (calculated as dry weight percent gain) of 60% was obtained. Similar yields have already been reported in response to the acetylation of other cellulose-enrich lignocellulosic biomasses [43, 50]. Through ¹H NMR spectrum, the degree of substitution of the three hydroxyl groups in the glucose monomer was calculated as 2.68.

6.3.3. Characterization of nanoparticles

The morphology and size distribution of MgO/TiO₂ nanoparticles were assessed by STEM. Figure 6.2a and 6.2b revealed that, under the same ablation parameters, MgO and TiO₂ particles were produced with different morphology and size. After the Mg ablation, the solution became turbid with a milky appearance and significant particle agglomeration occurred even during the ablation process. The observed MgO particles have irregular shapes (with some resembling platelet-like and spherical shapes) and formed clusters of agglomerates. Because of their morphology, the size distribution was calculated taking into account the largest length of the particles, which gave rise to a mix of nanoparticles and microparticles with sizes ranging from 59 to 313 nm. The largest length was taken into account since large nanoparticles size can impair certain applications and properties of the nanoparticles. The related histogram (Figure 6.3d) shows a unimodal, skewed right pattern, with 50% of particles not exceeding 175 nm and an average size around 155 nm. PLAL synthesized MgO particles with similar shapes were also obtained by Somanathan et al. [51] and Liang et al. [52]. The gray shades seen on images suggest that the organic solvent medium was decomposed or altered due to laser heating and this contributed to the poor visibility of particles. Furthermore, based on the difference of weight, the ablated Mg mass was calculated as 1.8 mg, leading to a solution with concentration of 0.15 mg/ml.

Besides the use of oxidized target materials, the formation of oxidized nanoparticles also takes place during the ablation process. During PLAL, the laser pulse delivery energy to the target through multiphoton absorption and direct photoionization, causing the detachment of high ionized species and the formation of a plasma-plume. As this plasma-plume expands and quench, the ablated species interact with the solution and oxidized phases of the bulk material are formed. In addition, energy is released to the surrounding liquid and a cavitation bubble rises as effect. Subsequently, this cavitation expands and collapses, leading to nanomaterials nucleation and growth and finally the release of nanoparticles to the liquid medium [35]. It is worth mentioning that all this process occurs in a fraction of seconds, of the order of 10-4 s [35]. Although the solvent used is not the most suitable medium for the formation of oxidized phases, very short liquid depth (measured from the target surface) was used to provide natural aeration with the upper air atmosphere. As both target materials (Ti and Mg) are very reactive, oxidation is expected to occur under the experimental setup. The formation of oxidized phases will be further confirmed through X-ray diffraction analysis.

The large particles produced after laser ablation are probably originated from the ejection of melted drops. Despite material detachment occur in a region almost coincident with the laser

spot, the energy density at the border of the laser beam is lower than that of the central. As a response, the heating temperature in this surrounding area only melts the target, and large nanoparticles in the form of melted drops can be blown away by the shockwave generated during plasma plume formation and cavitation bubble collapsing [53].

The formation of titanium dioxide, as well as the production of large particles evidenced in Figure 6.2a, follow the same synthesis mechanisms applied to the MgO. Based on the difference of weight, the ablated mass of TiO₂ was calculated as 2.8 mg, leading to a solution with concentration of 0.224 mg/ml. Through PLAL, TiO₂ is formed as well-defined spherical nanoparticles and diameters ranging from 15 to 330 nm. The related histogram (Figure 6.2c) shows a skewed right and unimodal pattern, with 50% of nanoparticles not exceeding 33 nm and an average size around 71.8 nm. Similar histogram pattern distribution and NPs size have recently been reported by Zimbone et al. [38], Zuñiga-Ibarra et al. [54], Singh et al. [37] and Aziz et al. [55]. In contrast to the MgO NPs, during ablation the solution color changes to a metallic gray and the produced TiO₂ NPs did not significantly cluster.

The presence of both nanoparticles (MgO and TiO₂) in the mixed colloidal solution can be detected via the UV-vis spectroscopy analysis presented in Figure 6.2e. The UV-vis spectrum exhibited a characteristic absorption band with a single plasmon absorption peak located in the low UV-region at 242 nm. The maximum absorption band verified, as well as the absorption edge around 400 nm, are typical of both nanoparticles [39, 54–61]. The presence of MgO and TiO₂ in the scope of the UV-vis spectroscopy can also be verified through the analysis of the band gap parameter. The band gap can be defined as the distance between the valence band (VB) of electrons and the conduction band (CB), and represents the minimum energy that is required to excite an electron up to a state in the conduction band so that it becomes free [62]. This parameter is crucial in many applications of TiO₂ nanoparticles as it largely influences their photocatalytic activity [54, 63]. The optical band gap (E_s) can be obtained via the Tauc method [64], which relates optical absorption strength with the photon energy and the band gap. Through this method, the optical band gap can be calculated by plotting $\alpha h v^{\alpha n}$ versus h v (Tauc plots), where h is the Planck's constant, ν is the frequency of light, α is the absorption coefficient and the exponent n is related to the electronic of the band gap (here denoted as indirect allowed transitions). The linear trend given by the Tauc plots is modeled as the tangent of the curve and extrapolated to the point where $\alpha h V^{(1/n)}$ is zero, which gives the E_e. From the inset graph in Figure 6.2e, the band gap of the mixed colloidal solution was estimated as 3.24 eV, which is higher than that of bare TiO₂ (close to

3.1 eV) and lower than that of bare MgO, that requires excitation energy in the range of 8-9 eV [65, 66]. In Ye et al. [61] the introduction of MgO as additive in TiO₂ films also caused the enlargement of the band gap. Therefore, the band gap value confirms that both nanoparticles were successfully synthesized and are present in the mixed colloidal suspension.

Although the enlarged band gap may impair the properties of TiO2 under the visible light spectrum, when combined with TiO_2 the MgO NPs can act as electron trap and barrier for recombination, enhancing its photocatalytic activity [66]. When TiO₂ is exposed to UV radiation, particles absorb light leading to band gap extinction and consequent charge separation of electrons and holes into the CB and VB, respectively. However, a large proportion of charge carriers (e- / h+ pairs) recombine quickly at the surface and interior of particles, dissipating the absorbed energy. This issue stands out one of the main drawbacks related to the photocatalytic property of TiO₂, since the electrons and holes, that work as electrons donors and acceptors, respectively, in reduction and oxidation reactions, are responsible for the generation of reactive oxygen species [62, 63]. In the scope of active packaging applications, these reactive radicals are recognized to effectively kill some microorganisms and enhance biodegradation of biodegradable polymers [67, 68]. Therefore, the improvement of the photocatalytic properties of TiO₂ NPs is imperative depend on the application, namely in active packaging. In this regard, MgO may play an important role when applied in combination with TiO₂ NPs since its microcrystalline structure contains a high density of defects with anion vacancies and cations vacancies sites. The anion vacancies are capable of trapping the excited electrons produced by photo-irradiation of TiO2, while the holes remain on the TiO₂, as evidenced in Bandara et al. [66]. Consequently, photogenerated negative and positive charges are widely separated and recombination can be suppressed [61, 66].

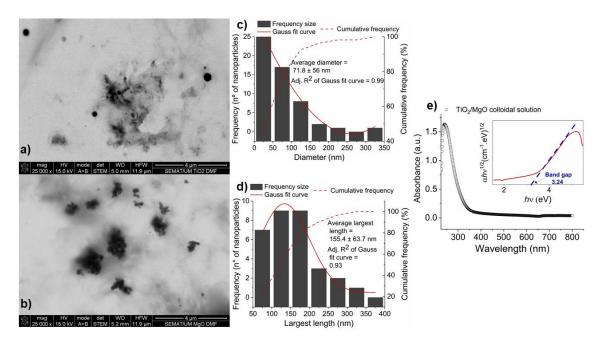


Figure 6-2. Nanoparticles characterization. (a) SEM image of TiO_2 NPs, (b) SEM image of MgO, (c) NPs distribution histogram of TiO_2 , (d) NPs distribution histogram of MgO and (e) UV-vis spectra of colloidal TiO_2 /MgO ablated nanoparticles in DCM (the inset graph shows the Tauc plot for $\alpha h \nu^{\text{U}/\text{O}}$ versus $h \nu$ and the respective band gap).

6.3.4. Fourier-transformed infrared spectroscopy

FTIR spectra were performed to verify the effectivity of the pretreatment and chemical modification applied, as well as possible chemical interactions between the CA matrix and TiO₂/MgO nanoparticles fillers (Figure 6.3). Absorbed peaks in extracted cellulose profile around 2890 cm⁻¹ (C-H stretching in cellulose-rich material), 1431 cm⁻¹ (CH₂ banding vibration in cellulose), 1368 cm⁻¹ (C-H asymmetric deformation in cellulose), 1315 cm⁻¹ (CH₂ wagging in cellulose), 1161 (asymmetric C-O-C stretching), 1102 cm⁻¹ (C-OH skeletal vibration), 1028 cm⁻¹ (C-O-C pyranose ring skeletal vibration) and 897 cm⁻¹ (C-H deformation), typical of cellulose-rich materials [8, 69–72], confirm the successful performance of the pretreatment applied to extract cellulose from corncob.

Regarding the acetylation procedure, the rising of new absorption peaks in the acetylated samples spectra, namely at 1734 cm⁻¹, 1368 cm⁻¹ and 1217 cm⁻¹ related to C=O stretching of ester group, C-H in O-(C=O)-CH₃ and C-O stretching of acetyl groups, respectively, confirm that the chemical modification was undoubtedly accomplished [43, 73]. Additional evidence of the acetylation is provided by the reduced intensity of peaks assigned to -OH stretching and through comparisons with the CA commercial spectra.

When it comes to the CA/MgO/TiO₂ nanocomposite film spectra, the absence of significant new peaks indicates that the nanoparticle fillers are only physically mixed into the CA matrix. No chemical interaction between the polymeric matrix and metal oxide nanoparticles has also been

verified by Swaroop et al. [16] and Ahmed et al. [74]. Notwithstanding, the rise of a broad peak from 3750 to 3100 cm⁻¹ is possibly associated with the OH stretching vibrations of MgO surface-adsorbed water molecules [75]. Peaks in the region of 2920-2850 can be attributed to asymmetrical stretching vibrations of hydrocarbon methyl groups from residual dichloromethane [76]. Although the FTIR analysis is not conclusive about the successful synthesis and embedment of MgO/TiO₂ nanoparticles into CA films, these issues will be, indeed, confirmed and explored in detail in the following X-ray diffractogram and EDS analysis.

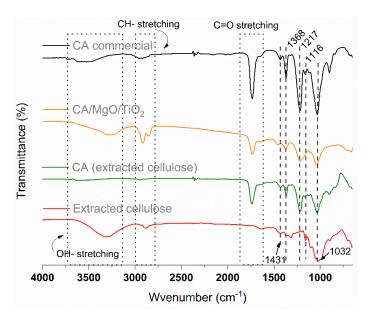


Figure 6-3. FTIR spectra of the extracted cellulose (LHW-dilute NaOH pretreatment), commercial CA, synthesized CA and CA/MgO/TiO₂ bionanocomposite film.

6.3.5. X-ray diffraction analysis

The diffractograms of the synthesized CA and the nanocomposite films are presented in Figure 6.4b. No sharp peaks are observed in the synthesized CA film profile, indicating that the produced film has reduced crystallinity. The appearance of a broad peak ranging from $2\theta = 13^{\circ}$ to $2\theta = 17^{\circ}$ can be assigned to the introduction of acetyl groups in the cellulosic fibers and the consequent enlargement of spaces between layers in the crystal facets [77, 78]. This feature indicates that occurred a change from crystalline to an amorphous structure during the esterification. The rise of another distinct peak around $2\theta = 8^{\circ}$ is also typical of cellulose acetylation and refers to a disorder in the crystalline structure of the cellulose. The disorder is associated with an increase in the interfibrillar distance due to the insertion of acetate groups, as well as with the breakdown of microfibrillar structures, as depicted in Figure 6.4a.

The prepared nanocomposite film also demonstrates the same peaks corresponding to amorphous CA, in addition to the rise of new peaks around $2\theta = 37.5^{\circ}$ and 43° , typical of MgO NPs [16, 79]. Additional peaks at $2\theta = 19.3^{\circ}$ and $2\theta = 34.2^{\circ}$ and 47.7° may also disclose the coexistence of Mg(OH)₂ and pure Mg nanoparticles, respectively [39, 41]. The defined peak at $2\theta = 23^{\circ}$ and 27.5° indicates that rutile was the most prominent phase of TiO₂ produced, which was also evidenced by other authors who conducted TiO₂ NPs synthesis through PLAL [37, 54]. However, typical peaks related to anatase TiO₂ ($2\theta = 25^{\circ}$, 31.3° and 47.7°) have been also clearly detected [53, 79]. In general, the XRD patterns related to TiO₂ and MgO nanoparticles show sharp peaks, suggesting that they consist of an array of crystalline grains, but low intensities, which is probably because the prepared materials are not presented in high concentrations. These findings support the successful synthesis of the metal oxides by PLAL, as well as the effective preparation of nanocomposite films through direct dissolution of the synthesized CA in the TiO₂/MgO/DCM colloidal solution, followed by solvent casting.

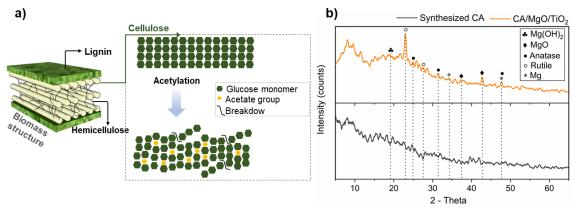


Figure 6-4. (a) Schematic representation of disorder caused in the cellulose structure due to acetate groups insertion and (b) XRD diffractograms of the synthesized CA and CA/MgO/TiO₂ films.

6.3.6. Scanning electron microscopy of the films

The surface morphology and cross-section of the synthesized CA and CA/MgO/TiO₂ films are presented, respectively, in Figures 6.5 (a and b) and Figure 6.5 (d and e). In general, surface investigation of the films revealed the occurrence of some roughness patterns and holes. The rough surface and the few porous structures verified in the cross-section relative to the synthesized CA may be attributed to the solvent evaporation processes, the casting surface or some undissolved remaining material. Notwithstanding, the quite smooth, flat and homogeneous surface of the produced synthesized CA film, with no fiber-like structure typical of the start cellulosic material,

denotes the successful acetylation procedure. This also highlights the effectivity of the combined pretreatment applied, which, by removing a great share of recalcitrant components, allowed the penetration of the acetylation reagents into the cellulose fibers.

By introducing the TiO₂ and MgO nanoparticles into the CA film, the porous size and porosity increased. This was expected to happen since the CA/acetone ratio in the casting solution of both films was kept the same. Besides, the agglomeration of metal oxide nanoparticles during film formation can give rise to microporous structures [80]. At high filler loading, MgO and TiO₂ tend to agglomerate in order to minimize their surface energy, and this effect has been reported to cause micro-cracks inside polymeric matrices [80, 81]. From surface and cross-section SEM images, it is evident that no physical defects are present on the CA/MgO/TiO₂ film, suggesting that the concentration of nanoparticles used was not able to cause structural defects in the film.

Although the nanoparticles are not visible in Figures 6.5d and 6.5e, possibly due to limitation of the SEM equipment or to their perfect embedment/immobilization inside the film, their presence (detected as Ti and Mg elements in Figure 6.6f) was confirmed by EDS analysis. Additional EDS analysis (not shown) confirmed uniform dispersion of nanoparticles throughout the film. Elemental composition of the synthesized CA film is also illustrated in Figure 6.5c, and confirms that the high content of carbon (C) and oxygen (O) comes from the cellulose acetate, and the presence residual elements (Na, Cl and S) are originated from previous treatments.

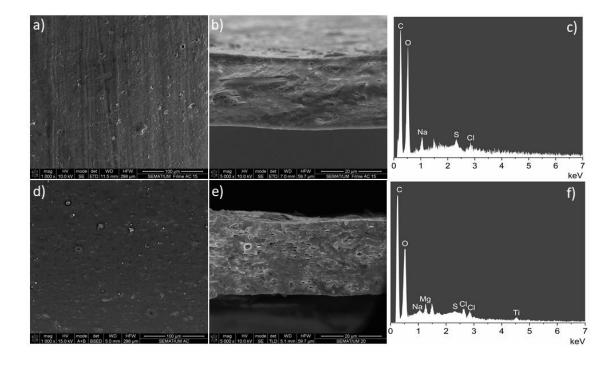


Figure 6-5. SEM micrographs and EDS analysis of the produced films: (a), (b), (c) refer to synthesized CA surface, cross-section and EDS analysis, respectively, and (d), (e), (f) refer to CA/TiO2/MgO surface, cross section and EDS, respectively.

6.3.7. Thermogravimetry analysis of films

Thermal stability of synthesized CA and CA/TiO₂/MgO films was determined using thermogravimetry analysis technique as shown in Figure 6.6. Degradation reactions observed in the synthesized CA curve occurred in a single step and in a narrow temperature range ($T_{onset} = 320$; T_{end} = 366). Regarding the CA/TiO₂/MgO film, two main weight loss zones are clearly identified in the DTG thermal profile. The distinguished initial weight loss of about 8.6 % in the temperature range of 140 – 202 °C can be related to an isolated thermal degradation of polymer chains due to the formation of a new crystalline phase favored by the introduction of TiO2, as reported by Valentim et al. [82] and Mohr et al. [68]. The latter authors concluded that TiO2 NPs act as nucleation agents for new crystalline phases on PLA with lower melting temperatures. Mohr et al. [68] also suggested that TiO₂ can increase the amorphous phase due to molecular disorganization leading to reduced visible radiation transmission. Overall, in comparison to the neat CA, the nanocomposite film started to degrade early. Indeed, the photoactivity/catalytic effect related to metal oxides, including TiO₂ and MgO, can make the polymer matrix degrade more easily and therefore decrease its degradation temperature [83, 84]. Other authors have also reported reduction on Tonet after incorporating TiO2, MgO and other metal oxides NPs into polymeric matrices [16, 81, 85]. For instance, in Restrepo et al. [84] the introduction of zinc oxide nanoparticles (5 % at weight basis) into polylactic acid (PLA) caused a reduction up to 117 °C on Tonset. Despite this, the temperature of maximum rate of weight loss (T_{max} - 350 °C) and the temperature at which the degradation processes of the CA/TiO₂/MgO film ends (T_{end}) were maintained the same as the neat synthesized CA. After incorporating TiO₂ nanoparticles into a PLA matrix, Sangiorgi et al. [83] verified this same thermal pattern. A similar result was also verified by Das & Gebru [86] when TiO₂ NPs were embedded into CA to prepare hybrid membranes, and by Mohr et al. [68]. At 600 °C the weight loss of the CA/TiO₂/MgO (89.23%) film was slightly higher than that of the neat CA (80.14 %). In response, a higher residual mass, partly related to the NPs contribution, was also recorded.

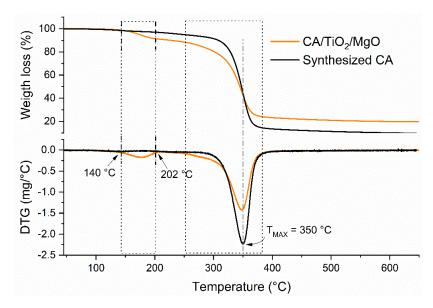


Figure 6-6. TG (upper graph) and DTG (lower graph) curves of the synthesized CA and CA/TiO_z/MgO films.

6.3.8. Dynamic mechanical analysis

The viscoelastic behavior of both films was characterized by the storage modulus E' (a measure of the material's ability to recover from deformation), the loss modulus E" (a measure of the material's ability to loses energy as heat) and the Tan δ or damping (an indicator of how efficiently the material loses energy to molecular rearrangement and internal friction), as illustrated in Figure 6.7a and 6.7b. The storage modulus profiles of both films show a pattern typical of amorphous polymers, which is related to the introduction of acetyl groups onto the AGU unity of cellulose fibers and the consequent unpackaging of chains. By comparing the viscoelastic properties of the films, it is clear that, after the embedment of nanoparticles, a reduction in the E' of about three times took place (Figure 6.7b). Based on the SEM images, this was expected since the neat CA film showed to has a denser structure, which translated into high stiffness when it comes to mechanical properties. In contrast, the enlarged porous structures brought by the embedment of TiO₂/MgO NPs into the CA matrix reduced stiffness and made the nanocomposite more flexible, which is supported by the higher Tan δ values. Another possible cause for this issue lies with the agglomeration of particles during film formation. Agglomeration leads to domains that are rich in the nanofillers and poor in the polymer, which ultimately can act as stress concentrating centers in a way that, if the inter-particle attractions are weak, they may play a large role in the storage and loss of applied mechanical energy [80, 87]. On the order hand, aggregation has been reported as an improving factor of photocatalytic activity for colloidal suspensions with mixed-phase anataserutile TiO₂ nanoparticles, since the migration of electrons across a phase junction was beneficial

for charge separation [38]. The decrease in elastic modulus of polymeric matrices provided by the introduction of metal oxide nanoparticles has already been reported by other authors [81, 85, 88]. Mohr et al. [68] associated the elastic modulus reduction of a PLA/TiO₂ nanocomposite with the formation of new crystalline forms that act as defects, handicapping the mechanical performance of the film. In general, from the literature it is possible to conclude that mechanical properties may change in function of the nanoparticle's concentration, type and dimension, as well as surround environmental conditions [16, 80, 81, 88].

Furthermore, in both films the storage modulus decreases when temperature increases due to increased segmental mobility of chains. However, a distinguished small plateau in the range of 60-75 °C is verified in the E' profile of the nanocomposite film (Figure 6.7b). This can be attributed to the reinforcement effect provided by the NPs, or to the formation of a small neck, observed exclusively in the concerned sample (inset image highlighted in Figure 6.7b). Within this neck, the chain axes become aligned parallel to the elongation direction, which leads to localized strengthening. In comparison to the neat CA film, the nanocomposite film also exhibited a less significant drop in E' over the entire temperature range, which can be related to mechanical restrictions introduced by the NPs. Besides, the nanocomposite had a broader transition instead of a steep drop-off in the E', indicating the presence and influence of embedded nanoparticles.

The broadening of the loss modulus curve of the nanocomposite film may also reflect a compositional gradient or a molecular mobility gradient [89]. Distinctly, a sharp peak is observed in the E'' of the neat CA film curve preceding its drop around 136 °C. A broad secondary peak in E'' of the neat CA film is also verified at about 64 °C. In response there is a decrease in E' from 40 °C to 100 °C, probably originated from the natural mobility of chains in response to temperature rising and relaxation transitions triggered by local mobility of chains plasticized with water [90].

By applying the peak of Tan δ curve method to obtain the glass transition temperature (T_s), it is noticeable that the incorporation nanoparticles increased significantly the T_s of the nanocomposite film ($T_s = 176$ °C). This trend is in line with the E' and E'', which once more indicates that, near the nanoparticles, the cellulose acetate chains have their mobility affected. The T_s value of the pure synthesized CA film was found to be similar to other cellulose acetate films reported in literature [91–94]. The glass transition temperature is an important parameter to be defined since it determines the operating range of the material, which was extended in the nanocomposite film.

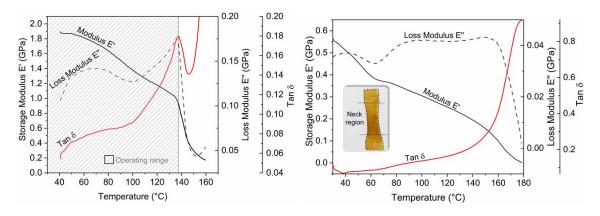


Figure 6-7. Dynamic mechanical analysis of the (a) synthesized CA and (b) CA/TiO₂/MgO film.

6.4. Conclusions

This study was focused on presenting an eco-friendly processing route to produce bio-based nanocomposites with potential to be applied as active packaging. By applying a green two-stage combined pretreatment, non-cellulosic components of corncob could be removed, which greatly favored the synthesis of cellulose acetate with remarkable yield. Parallel to that, colloidal suspensions of TiO2 and MgO were effortlessly produced through PLA in DCM. The mixed colloidal suspensions could be promptly used to dissolve the synthesized CA, which was further turned into a film via the solvent casting method. The FTIR analysis did not present any response regarding the metal oxides, suggesting that there was no chemical interaction between the CA matrix and nanofillers. The formation of crystalline oxidized phases of the metals, as well as their presence in the mixed colloidal suspension and embedment into the CA matrix, could be confirmed through UV-vis spectroscopy, XRD and EDS, attesting the suitability of the proposed processing. The thermal stability of the CA matrix was not significantly affected after the introduction of nanoparticles, but its onset temperature was lowered as expected. Conversely, the stiffness of the CA matrix was reduced due to the increase of porous structures, which leads to flexible nanocomposites. By restricting chain mobility of the cellulose acetate, the metal oxide nanoparticles could also extend the operating range of the bionanocomposite. Overall, in addition to the eco-friendly aspect, the versatility and scalability of the applied techniques, as well as the low cost of raw-materials used, also underpin the technical feasibility of the presented synthesis processing. Changes in cellulose acetate and nanoparticle properties can be easily accomplished by tailoring their synthesis methodologies, and their effects on nanocomposite film properties can be studied in the future. Furthermore, the active properties of the produced nanocomposite (namely biodegradability and antimicrobial properties) can be assessed and validated in the forthcoming activities.

Acknowledgements

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Chapter 7. Conclusion and Future Work

This chapter presents the general conclusions of this PhD project, the suggestions for future works and the further contributions of this thesis.

7.1 General conclusions

The current thesis has looked into the application of eco-friendly techniques to turn agroindustrial residues into cellulose-based bioplastics. Following the selection of a suitable agroindustrial residue, the outstanding extraction of cellulose by means a combined green pretreatment was the pivot achievement for the successful valorization of the biomass. The processing of cellulose to produce value-added biomaterials/nanocomposites was carried out by performing innovative and environmentally sustainable synthesis routes, including ionic liquid dissolution/regeneration, solvent-free acetylation and laser ablation in liquids.

Chapter III presented a diagnosis study assisted by the implementation of a linear model in addressing the identification of suitable agroindustrial residues in Brazil. The study revealed the huge availability of residues and its expected growth in the next years. Apart from the prompt availability, these residues represent an alternative low-cost source of raw-material as there is still no market demand for them. The linear suitability model indicates plenty of agroindustrial residues with the potential to be applied in the manufacturing of cellulose-based products, although some issues have to be tackled to ensure their maximum valorization, manly regarding the observed spatio-temporal variability. The use of versatile processing techniques, that can be tailored to meet possible changes on processing design, including the treatment of residues with different physicochemical characteristics, may partially overcome this drawback.

Supported by the above-mentioned study, corncob was selected as a potential source of cellulose. In **Chapter IV** different green pretreatments technologies, namely ionic liquid, liquid hot water (LHW), dilute-NaOH, as well as their combined application, were performed and assessed regarding their potential effects on the physicochemical properties of the corncob. The results revealed that, depending on the pretreatment technology, different physicochemical changes may occur, whether at the chemical composition or crystalline structure. The treatment with ionic liquid (BmimCl) did not significantly change the chemical composition of the corncob, but in turn modified its crystalline structure to an amorphous phase. LHW and dilute-NaOH were highly efficient to extract hemicellulose and lignin, respectively, while maintained unchanged the crystalline structure. Among the combined pretreatments performed, LHW-dilute NaOH achieved remarkable delignification efficiency under mild conditions. By selectively removing non-cellulosic components, the recalcitrance of the biomass was significantly decreased in response of disruption of the interand intramolecular bonds. The obtained cellulose-enrich sample was suitable to produce a bioplastic film with notable mechanical properties.

For the concerned application, the combined LHW-dilute NaOH pretreatment has proved to be the most promising approach, as a high cellulose fraction was obtained and no changes on the crystallinity of the raw biomass have been set. Besides, it allowed the selective fractionation of hemicellulose and lignin in different stages. This is a crucial requirement for the economic viability of biomass valorization into cellulose-based products. Therefore, this pretreatment design was applied as the first processing step of subsequent treatments undertaken in this work. In **Chapter V**, the extracted cellulose was synthesized into cellulose acetate – a polymeric product with a wide range of applications and recognized biodegradable property. In line with the bias of the current thesis, a solvent-free eco-friendly acetylation technique was performed. The high yield obtained after acetylation, along with the less use of chemicals, endorsed the effectivity of the applied technique, which in turn had a positive contribution in reducing environmental impacts.

The synthesized CA was also applied to produce a bionanocomposite in **Chapter VI**. Nanocomposites based on bio-based matrices reinforced by metal oxide nanofillers are an emerged type of material in the field of active packaging. However, the development of sustainable processing technologies, whether to produce the matrices or the nanofillers, has become imperative. Therefore, in the concerned chapter, TiO₂ and MgO were directly synthesized via pulsed laser ablation in a solvent media suitable to dissolve the cellulose acetate. The proposed technique has shown to be versatile, easy-to-manage, clean and suitable to produce and embed the nanoparticles into the bio-based matrix. Considering that the bionanocomposite produced features good physicochemical properties and possible active antimicrobial/biodegradable activity, the proposed synthesis route appears to be a promising approach for developing novel active packaging components.

Overall, this thesis shows that agroindustrial residues may be a promising source of raw material to produce value-added polymeric materials. In addition to prevent environmental impacts related to unsuitable disposal of biomass and avoid the consumption of fossil resources, the emergence of these new bio-based products can also create new market niches and foster the circular economy. On top of that, besides to be highly effective for the concerned application, the proposed eco-friendly techniques can meet the main objectives of green chemistry. Notwithstanding, when applied as an integrated processing, some issues must be taken into account in order to ensure techno-economic feasibility, notably with regards to the valorization of the non-cellulosic components selectively extracted in the pretreatment step. It is also worth

mentioning that some of the techniques used have received increasing acceptance for large scale applications, namely LHW and laser technology, but this is still a challenging prospect.

7.2 Suggestions for future works

Based on the developed work and results obtained, some suggestions of complementary and prospective investigations may include:

- The processing of other agroindustrial residues, preferentially with higher cellulose contents, and assessment of the proposed combined LHW-dilute NaOH pretreatment with regard to the fractionation of a mix of residues;
- The valorization of the other lignocellulose components, namely hemicellulose and lignin, extracted during the pretreatment step;
- Development of new approaches for the synthesis of metal oxides by means of laser ablation technology;
- Assessment of the antimicrobial and biodegradability activity, as well as barrier properties,
 of the produced bionanocomposite in order to prove the active properties provided by the
 embedment of TiO₂/MgO nanoparticles;
- Study of migration of nanoparticles from the produced bionanocomposite.
- Modeling of other life cycle assessment scenarios, taking into account the valorization of the non-cellulosic components, the laser ablation step, as well as the assumption of possible end-of-life technologies.

Appendix A

This appendix presents the additional information regarding the methods and data used to determine the chemical compositions of untreated and pretreated corncob.

Supplementary information - Compositional analysis method and related data of untreated and pretreated samples

It is important to keep in mind that, in order to respect the typical thermal behavior of the concerned substances (cellulose, hemicellulose and lignin), when necessary some parameters were kept fixed. Therefore, the selection of the best fit modeling curves was based mainly on the Adj. R-square, the F value and the regular residual graph.

Adj. R-square: The Adj. R-square is a modified version of R-square, which is adjusted for the number of predictors in the fitted line. Thus, it can be used to compare the fitted lines with different numbers of predictors. If the number of predictors is greater than 1, Adj.-square is always smaller than R-square.

F value: is a ratio of two mean squares, which can be computed by dividing the mean square of fitted model by the mean square of error. It is a test statistic for a test of whether the fitted model differs significantly from the model y=constant, which is a flat line with slope being equal to zero. It can be inferred that the more this ratio deviates from 1, the stronger the evidence for the fitted model differing significantly from the model y=constant.

Regular residual curve: was used to assess the modelled fit curve quality.

Corncob compositional analysis – deconvolution parameters and statistic:

Table 1. Deconvolution parameters. Where **y0** is the initialization parameter, **xc** curve center, **w** curve width and **A** curve area.

| Parameters | | Value | Standard error |
|----------------------|--------|----------|----------------|
| Hemicellulose peak 1 | y0 | -0.01832 | 3.42E-04 |
| Hemicellulose peak 1 | XC | 268.3016 | 0.12606 |
| Hemicellulose peak 1 | W | 32.98282 | 0.26402 |
| Hemicellulose peak 1 | Α | -9.58108 | 0 |
| Hemicellulose peak 1 | sigma | 16.49141 | 0.13201 |
| Hemicellulose peak 1 | FWHM | 38.83431 | 0.31086 |
| Hemicellulose peak 1 | Height | -0.23178 | 0.00186 |
| Hemicellulose peak 2 | yO | -0.01832 | 3.42E-04 |
| Hemicellulose peak 2 | XC | 301.7857 | 0 |
| Hemicellulose peak 2 | W | 32.31568 | 0.11413 |
| Hemicellulose peak 2 | Α | -24.3289 | 0 |
| Hemicellulose peak 2 | sigma | 16.15784 | 0.05706 |
| Hemicellulose peak 2 | FWHM | 38.0488 | 0.13437 |
| Hemicellulose peak 2 | Height | -0.60069 | 0.00212 |
| Cellulose | y0 | -0.01832 | 3.42E-04 |
| Cellulose | XC | 340.5019 | 0 |
| Cellulose | W | 26.84495 | 0.07827 |
| Cellulose | Α | -26.0462 | 0.10211 |
| | | | |

| Cellulose | sigma | 13.42247 | 0.03913 | |
|-----------|--------|----------|----------|--|
| Cellulose | FWHM | 31.60751 | 0.09215 | |
| Cellulose | Height | -0.77414 | 0.00213 | |
| Lignin | yO | -0.01832 | 3.42E-04 | |
| Lignin | XC | 340.5019 | 0 | |
| Lignin | W | 173.7918 | 3.49218 | |
| Lignin | А | -11.6974 | 0 | |
| Lignin | sigma | 86.89589 | 1.74609 | |
| Lignin | FWHM | 204.6242 | 4.11172 | |
| Lignin | Height | -0.0537 | 0.00108 | |

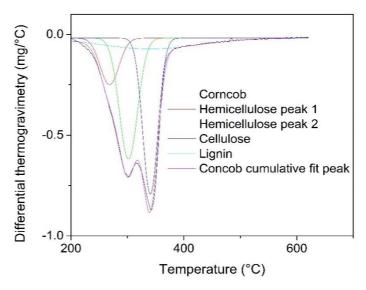


Figure 1. Corncob deconvoluted differential thermogravimetric curve

Table 2. Statistcs and ANOVA

| Statistics | |
|-------------------------|----------|
| Number of Points | 1806 |
| Degrees of Freedom | 1799 |
| Reduced Chi-Sqr | 1.57E-04 |
| Residual Sum of Squares | 0.28315 |
| Adj. R-Square | 0.99759 |

| ANOVA | DF | Sum of Squares | Mean Square | F Value |
|--------------------------|------|----------------|-------------|----------|
| Regression | 7 | 177.81273 | 25.40182 | 161392.5 |
| Residual | 1799 | 0.28315 | 1.57E-04 | |
| Uncorrected Total | 1806 | 178.09588 | | |
| Corrected Total | 1805 | 117.73283 | | |

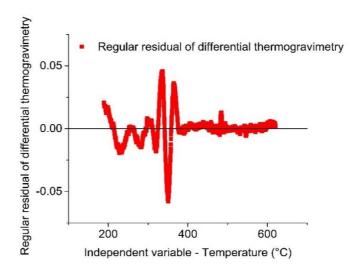


Figure 2. Regular residual from differential thermogravimetric of corncob.

Adjusted model equation - corncob cumulative fit peak (Equation 1).

$$y = y0 + \{ \frac{A}{(w \times \sqrt{\pi/2})} \times e^{\left[-2 \times \frac{(x - xc)^2}{w^2}\right]} \}$$
 (1)

• **LHW-treated corncob compositional analysis** - deconvolution parameters and statistic

Table 3. Deconvolution parameters. Where y0 is the initialization parameter, xc curve center, w curve width and A curve area.

| Parameters | | Value | Standard Error |
|---------------|----|----------|----------------|
| Hemicellulose | у0 | -0.0311 | 9.50E-04 |
| Hemicellulose | XC | 300.7745 | 0 |
| Hemicellulose | Α | -2.93221 | 0 |
| Hemicellulose | W | 60.83908 | 0 |
| Cellulose | y0 | -0.0311 | 9.50E-04 |
| Cellulose | XC | 350.9776 | 0 |
| Cellulose | Α | -58.098 | 0.39646 |
| Cellulose | W | 32.48151 | 0.1636 |
| Lignin | y0 | -0.0311 | 9.50E-04 |
| Lignin | XC | 350.7821 | 0 |
| Lignin | Α | -13.1306 | 0.61305 |
| Lignin | W | 100.7608 | 0 |

Appendix A

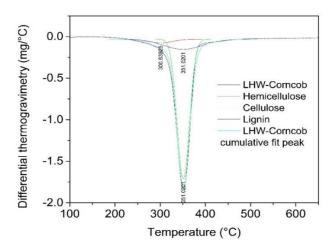


Figure 3. LHW-treated corncob deconvoluted differential thermogravimetric curve

Table 4. Statistcs and ANOVA

| Statistics | | |
|-------------------------|---------|--|
| Number of Points | 2869 | |
| Degrees of Freedom | 2865 | |
| Reduced Chi-Sqr | 0.00192 | |
| Residual Sum of Squares | 5.51418 | |
| Adj. R-Square | 0.98325 | |

| ANOVA | DF | Sum of Squares | Mean Square | F Value |
|-------------------|------|-------------------|-------------|----------|
| Regression | 4 | 379.6543 | 94.91357 | 49314.18 |
| Residual | 2865 | 5.51418 | 0.00192 | |
| Uncorrected Total | 2869 | 385.16848 | | |
| Corrected Total | 2868 | 329.52425 | | |

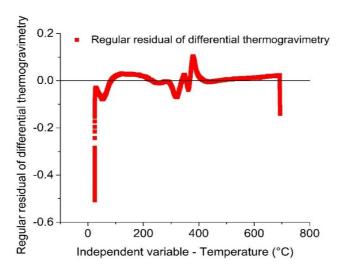


Figure 4. Regular residual from differential thermogravimetric of LHW-treated corncob.

Adjusted model equation – LHW-treated corncob cumulative fit peak (Equation 2)

$$y=y0+\left[\frac{A}{w\times\sqrt{\pi/4\times\ln(2)}}\right]\times e^{\left\{-4\times\ln(2)\times\left[\frac{(x\cdot xc)^2}{w^2}\right]\right\}}$$
 (2)

• **LHW-IL-treated corncob compositional analysis** - deconvolution parameters and statistic

Table 5. deconvolution parameters. Where y0 is the initialization parameter, xc curve center, w curve width and A curve area.

| Parameters | | Value | Standard Error |
|---------------|--------|----------|----------------|
| Hemicellulose | y0 | -0.02383 | 8.73E-04 |
| Hemicellulose | XC | 265 | 0 |
| Hemicellulose | W | 21 | 0 |
| Hemicellulose | Α | -2.875 | 0 |
| Hemicellulose | sigma | 10.5 | 0 |
| Hemicellulose | FWHM | 24.72561 | 0 |
| Hemicellulose | Height | -0.10923 | 0 |
| Cellulose 1 | y0 | -0.02383 | 8.73E-04 |
| Cellulose 1 | XC | 309.2415 | 0.14612 |
| Cellulose 1 | W | 48 | 0 |
| Cellulose 1 | Α | -53 | 0 |
| Cellulose 1 | sigma | 24 | 0 |
| Cellulose 1 | FWHM | 56.51568 | 0 |
| Cellulose 1 | Height | -0.881 | 0 |
| Cellulose 2 | y0 | -0.02383 | 8.73E-04 |
| Cellulose 2 | XC | 355 | 0 |
| Cellulose 2 | W | 17.5 | 0 |
| Cellulose 2 | Α | -4.859 | 0 |
| Cellulose 2 | sigma | 8.75 | 0 |
| Cellulose 2 | FWHM | 20.60468 | 0 |
| Cellulose 2 | Height | -0.22154 | 0 |
| Lignin | y0 | -0.02383 | 8.73E-04 |
| Lignin | XC | 288.4015 | 4.19703 |
| Lignin | W | 155.255 | 0 |
| Lignin | Α | -10.73 | 0 |
| Lignin | sigma | 77.6275 | 0 |
| Lignin | FWHM | 182.7988 | 0 |
| Lignin | Height | -0.05514 | 0 |

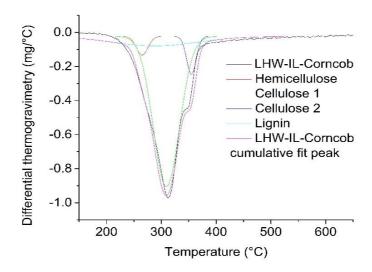


Figure 5. LHW-IL-treated corncob deconvoluted differential thermogravimetric curve.

Table 6. Statistcs and ANOVA

| Statistcs | |
|-------------------------|---------|
| Number of Points | 2867 |
| Degrees of Freedom | 2864 |
| Reduced Chi-Sqr | 0.00218 |
| Residual Sum of Squares | 6.25735 |
| Adj. R-Square | 0.95485 |

| ANOVA | DF | Sum of Squares | Mean Square | F Value | |
|--------------------------|------|----------------|-------------|----------|--|
| Regression | 3 | 179.43908 | 59.81303 | 27376.51 | |
| Residual | 2864 | 6.25735 | 0.00218 | | |
| Uncorrected Total | 2867 | 185.69643 | | | |
| Corrected Total | 2866 | 138.68612 | | | |

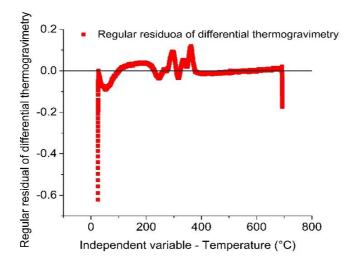


Figure 6. Regular residual from differential thermogravimetric of LHW-IL-treated corncob.

Adjusted model equation – LHW-IL-treated corncob cumulative fit peak (Equation 3)

$$y = y0 + \{\frac{A}{(w \times \sqrt{\pi/2})} \times e^{\left[-2 \times \frac{(x - xc)^2}{w^2}\right]}\}$$
 (3)

• **LHW-NaOH treated corncob compositional analysis** - deconvolution parameters and statistic

Table 7. deconvolution parameters. Where y0 is the initialization parameter, xc curve center, w curve width and A curve area.

| P | arameters | Value | Standard Error |
|---------------|-----------|----------|----------------|
| Hemicellulose | y0 | -0.03872 | 0.0014 |
| Hemicellulose | XC | 328 | 0 |
| Hemicellulose | W | 15 | 0 |
| Hemicellulose | Α | -3 | 0 |
| Hemicellulose | sigma | 7.5 | 0 |
| Hemicellulose | FWHM | 17.66115 | 0 |
| Hemicellulose | Height | -0.15958 | 0 |
| Cellulose | y0 | -0.03872 | 0.0014 |
| Cellulose | XC | 358.2 | 0 |
| Cellulose | W | 25 | 0 |
| Cellulose | Α | -55.5 | 0 |
| Cellulose | sigma | 12.5 | 0 |
| Cellulose | FWHM | 29.43525 | 0 |
| Cellulose | Height | -1.7713 | 0 |
| Lignin | y0 | -0.03872 | 0.0014 |
| Lignin | XC | 318 | 0 |
| Lignin | W | 38 | 0 |
| Lignin | Α | -7 | 0 |
| Lignin | sigma | 19 | 0 |
| Lignin | FWHM | 44.74158 | 0 |
| Lignin | Height | -0.14698 | 0 |

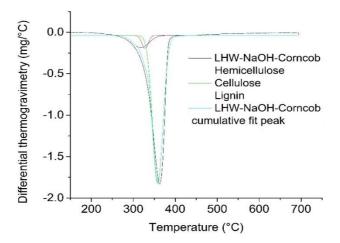


Figure 7. LHW-NaOH-treated corncob deconvoluted differential thermogravimetric curve.

Table 8. Statistcs and ANOVA

| Statistcs | |
|-------------------------|----------|
| Number of Points | 2862 |
| Degrees of Freedom | 2861 |
| Reduced Chi-Sqr | 0.00558 |
| Residual Sum of Squares | 15.96532 |
| Adj. R-Square | 0.95209 |

| ANOVA | DF | Sum of Squares | Mean Square | F Value |
|--------------------------|------|----------------|-------------|----------|
| Regression | 1 | 369.61839 | 369.61839 | 66235.95 |
| Residual | 2861 | 15.96532 | 0.00558 | |
| Uncorrected Total | 2862 | 385.58371 | | |
| Corrected Total | 2861 | 333.22655 | | |

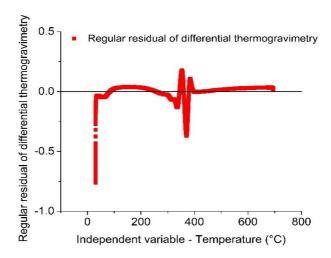


Figure 8. Regular residual from differential thermogravimetric of LHW-NaOH-treated corncob.

Adjusted model equation LHW-NaOH-treated corncob cumulative fit peak (Equation 4)

$$y = y0 + \{\frac{A}{(w \times \sqrt{\pi/2})} \times e^{\left[-2 \times \frac{(x - xc)^2}{w^2}\right]}\}$$
 (4)

• **LHW-NaOH-IL treated corncob compositional analysis** - deconvolution parameters and statistic

Table 9. Deconvolution parameters. Where y0 is the initialization parameter, xc curve center, w curve width and A curve area.

| | Parameters | Value | Standard Error |
|---------------|------------|----------|----------------|
| Hemicellulose | y0 | -0.0142 | 8.02E-04 |
| Hemicellulose | XC | 258.2081 | 0 |
| Hemicellulose | Α | -2.72845 | 0 |
| Hemicellulose | W | 40.12654 | 0 |

| Cellulose peak 1 | y0 | -0.0142 | 8.02E-04 | |
|------------------|----|----------|----------|--|
| Cellulose peak 1 | XC | 296.5643 | 0 | |
| Cellulose peak 1 | Α | -65.1565 | 0 | |
| Cellulose peak 1 | W | 43.12656 | 0 | |
| Cellulose peak 2 | y0 | -0.0142 | 8.02E-04 | |
| Cellulose peak 2 | XC | 353.1305 | 0 | |
| Cellulose peak 2 | Α | -1.34377 | 0 | |
| Cellulose peak 2 | W | 18.12654 | 0 | |
| Lignin | y0 | -0.0142 | 8.02E-04 | |
| Lignin | XC | 330.6556 | 0 | |
| Lignin | Α | -8.00355 | 0 | |
| Lignin | W | 150.1266 | 0 | |

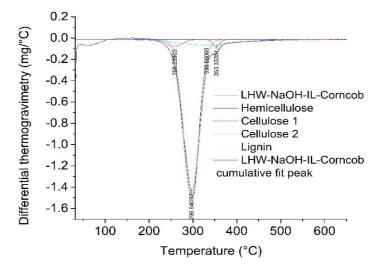


Figure 9. LHW-NaOH-IL-treated corncob deconvoluted differential thermogravimetric curve.

Table 10. Statistcs and ANOVA

| Statistcs | |
|-------------------------|---------|
| Number of Points | 2860 |
| Degrees of Freedom | 2859 |
| Reduced Chi-Sqr | 0.00184 |
| Residual Sum of Squares | 5.25825 |
| Adj. R-Square | 0.97925 |

| ANOVA | DF | Sum of Squares | Mean Square | F Value |
|--------------------------|------|----------------|-------------|----------|
| Regression | 1 | 295.10028 | 295.10028 | 160451.2 |
| Residual | 2859 | 5.25825 | 0.00184 | |
| Uncorrected Total | 2860 | 300.35853 | | |
| Corrected Total | 2859 | 253.39564 | | |

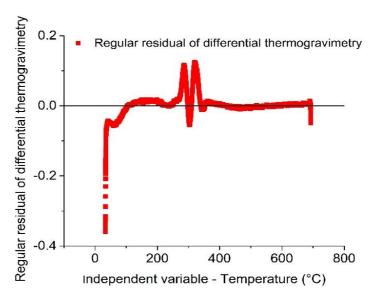


Figure 10. Regular residual from differential thermogravimetric of LHW-NaOH-IL-treated corncob.

Adjusted model equation LHW-NaOH-IL-treated corncob cumulative fit peak (Equation 5).

$$y=y0+\left[\frac{A}{w\times\sqrt{\pi/4\times\ln(2)}}\right]\times e^{\left\{-4\times\ln(2)\times\left[\frac{(x\cdot xc)^2}{w^2}\right]\right\}}$$
(5)

 NaOH-treated corncob compositional analysis - deconvolution parameters and statistic.

Table 11. Deconvolution parameters. Where y0 is the initialization parameter, xc curve center, w curve width and A curve area.

| Para | meters | Value | Standard Error |
|----------------------|--------|----------|----------------|
| Hemicellulose peak 1 | y0 | -0.02426 | 0.00151 |
| Hemicellulose peak 1 | XC | 284 | 0 |
| Hemicellulose peak 1 | W | 27 | 0 |
| Hemicellulose peak 1 | Α | -3 | 0 |
| Hemicellulose peak 1 | sigma | 13.5 | 0 |
| Hemicellulose peak 1 | FWHM | 31.79007 | 0 |
| Hemicellulose peak 1 | Height | -0.08865 | 0 |
| Hemicellulose peak 2 | y0 | -0.02426 | 0.00151 |
| Hemicellulose peak 2 | XC | 314.5 | 0 |
| Hemicellulose peak 2 | W | 28 | 0 |
| Hemicellulose peak 2 | Α | -17 | 0 |
| Hemicellulose peak 2 | sigma | 14 | 0 |
| Hemicellulose peak 2 | FWHM | 32.96748 | 0 |
| Hemicellulose peak 2 | Height | -0.48443 | 0 |
| Cellulose | у0 | -0.02426 | 0.00151 |
| Cellulose | XC | 348 | 0 |
| Cellulose | W | 28 | 0 |
| Cellulose | Α | -46 | 0 |
| Cellulose | sigma | 14 | 0 |
| Cellulose | FWHM | 32.96748 | 0 |

| Cellulose | Height | -1.31081 | 0 | |
|-----------|--------|----------|---------|--|
| Lignin | yO | -0.02426 | 0.00151 | |
| Lignin | XC | 385 | 0 | |
| Lignin | W | 150 | 0 | |
| Lignin | Α | -5.74 | 0 | |
| Lignin | sigma | 75 | 0 | |
| Lignin | FWHM | 176.6115 | 0 | |
| Lignin | Height | -0.03053 | 0 | |

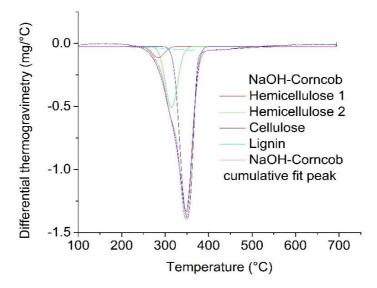


Figure 11. NaOH-treated corncob deconvoluted differential thermogravimetric curve.

Table 12. Statistcs and ANOVA

| Statistcs | |
|-------------------------|----------|
| Number of Points | 2859 |
| Degrees of Freedom | 2858 |
| Reduced Chi-Sqr | 0.00648 |
| Residual Sum of Squares | 18.51896 |
| Adj. R-Square | 0.9261 |

| ANOVA | DF | Sum of Squares | Mean Square | F Value |
|--------------------------|------|----------------|-------------|----------|
| Regression | 1 | 280.42404 | 280.42404 | 43277.37 |
| Residual | 2858 | 18.51896 | 0.00648 | |
| Uncorrected Total | 2859 | 298.943 | | |
| Corrected Total | 2858 | 250.58362 | | |

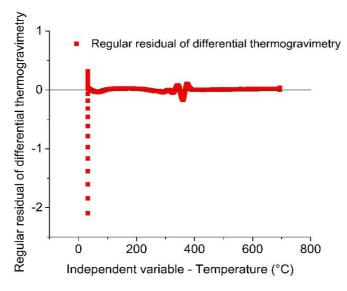


Figure 12. Regular residual from differential thermogravimetric of LHW-NaOH-IL-treated corncob.

Adjusted model equation – NaOH-treated corncob cumulative fit peak (Equation 6)

$$y = y0 + \left\{ \frac{A}{(w \times \sqrt{\pi/2})} \times e^{\left[-2 \times \frac{(x - xc)^2}{w^2}\right]} \right\}$$
 (6)

• **IL-treated corncob compositional analysis** - deconvolution parameters and statistic.

Table 13. Deconvolution parameters. Where y0 is the initialization parameter, xc curve center, w curve width and A curve area.

| Para | meters | Value | Standard Error |
|----------------------|--------|----------|----------------|
| Hemicellulose peak 1 | y0 | -0.0168 | 4.90E-04 |
| Hemicellulose peak 1 | XC | 237.8476 | 0 |
| Hemicellulose peak 1 | W | 16.8874 | 0 |
| Hemicellulose peak 1 | Α | -0.25829 | 0 |
| Hemicellulose peak 1 | sigma | 8.4437 | 0 |
| Hemicellulose peak 1 | FWHM | 19.8834 | 0 |
| Hemicellulose peak 1 | Height | -0.0122 | 0 |
| Hemicellulose peak 2 | y0 | -0.0168 | 4.90E-04 |
| Hemicellulose peak 2 | XC | 281 | 0 |
| Hemicellulose peak 2 | W | 43.16918 | 0.19918 |
| Hemicellulose peak 2 | Α | -33.1181 | 0.13856 |
| Hemicellulose peak 2 | sigma | 21.58459 | 0.09959 |
| Hemicellulose peak 2 | FWHM | 50.82783 | 0.23452 |
| Hemicellulose peak 2 | Height | -0.61211 | 0.00231 |
| Lignin peak 1 | y0 | -0.0168 | 4.90E-04 |
| Lignin peak 1 | XC | 322 | 0 |
| Lignin peak 1 | W | 27 | 0 |
| Lignin peak 1 | Α | -7 | 0 |
| Lignin peak 1 | sigma | 13.5 | 0 |
| Lignin peak 1 | FWHM | 31.79007 | 0 |
| Lignin peak 1 | Height | -0.20686 | 0 |
| Cellulose | y0 | -0.0168 | 4.90E-04 |

| Cellulose xc 353.9608 0.05454 Cellulose w 27.87115 0.08778 Cellulose A -30 0 Cellulose sigma 13.93557 0.04389 Cellulose FWHM 32.81577 0.10335 Cellulose Height -0.85883 0.0027 Lignin peak 2 y0 -0.0168 4.90E-04 Lignin peak 2 xc 400 0 Lignin peak 2 w 110 0 Lignin peak 2 sigma 55 0 Lignin peak 2 FWHM 129.5151 0 Lignin peak 2 FWHM 129.5151 0 | | | | | |
|--|---------------|--------|----------|----------|--|
| Cellulose A -30 0 Cellulose sigma 13.93557 0.04389 Cellulose FWHM 32.81577 0.10335 Cellulose Height -0.85883 0.0027 Lignin peak 2 y0 -0.0168 4.90E-04 Lignin peak 2 xc 400 0 Lignin peak 2 w 110 0 Lignin peak 2 A -6 0 Lignin peak 2 sigma 55 0 Lignin peak 2 FWHM 129.5151 0 | Cellulose | XC | 353.9608 | 0.05454 | |
| Cellulose sigma 13.93557 0.04389 Cellulose FWHM 32.81577 0.10335 Cellulose Height -0.85883 0.0027 Lignin peak 2 y0 -0.0168 4.90E-04 Lignin peak 2 xc 400 0 Lignin peak 2 w 110 0 Lignin peak 2 A -6 0 Lignin peak 2 sigma 55 0 Lignin peak 2 FWHM 129.5151 0 | Cellulose | W | 27.87115 | 0.08778 | |
| Cellulose FWHM 32.81577 0.10335 Cellulose Height -0.85883 0.0027 Lignin peak 2 y0 -0.0168 4.90E-04 Lignin peak 2 xc 400 0 Lignin peak 2 w 110 0 Lignin peak 2 A -6 0 Lignin peak 2 sigma 55 0 Lignin peak 2 FWHM 129.5151 0 | Cellulose | Α | -30 | 0 | |
| Cellulose Height -0.85883 0.0027 Lignin peak 2 y0 -0.0168 4.90E-04 Lignin peak 2 xc 400 0 Lignin peak 2 w 110 0 Lignin peak 2 A -6 0 Lignin peak 2 sigma 55 0 Lignin peak 2 FWHM 129.5151 0 | Cellulose | sigma | 13.93557 | 0.04389 | |
| Lignin peak 2 y0 -0.0168 4.90E-04 Lignin peak 2 xc 400 0 Lignin peak 2 w 110 0 Lignin peak 2 A -6 0 Lignin peak 2 sigma 55 0 Lignin peak 2 FWHM 129.5151 0 | Cellulose | FWHM | 32.81577 | 0.10335 | |
| Lignin peak 2 xc 400 0 Lignin peak 2 w 110 0 Lignin peak 2 A -6 0 Lignin peak 2 sigma 55 0 Lignin peak 2 FWHM 129.5151 0 | Cellulose | Height | -0.85883 | 0.0027 | |
| Lignin peak 2 w 110 0 Lignin peak 2 A -6 0 Lignin peak 2 sigma 55 0 Lignin peak 2 FWHM 129.5151 0 | Lignin peak 2 | y0 | -0.0168 | 4.90E-04 | |
| Lignin peak 2 A -6 0 Lignin peak 2 sigma 55 0 Lignin peak 2 FWHM 129.5151 0 | Lignin peak 2 | XC | 400 | 0 | |
| Lignin peak 2 sigma 55 0 Lignin peak 2 FWHM 129.5151 0 | Lignin peak 2 | W | 110 | 0 | |
| Lignin peak 2 FWHM 129.5151 0 | Lignin peak 2 | Α | -6 | 0 | |
| | Lignin peak 2 | sigma | 55 | 0 | |
| Lignin pook 2 Hoight 0.04352 0 | Lignin peak 2 | FWHM | 129.5151 | 0 | |
| Ligitii peak 2 Height -0.04552 0 | Lignin peak 2 | Height | -0.04352 | 0 | |

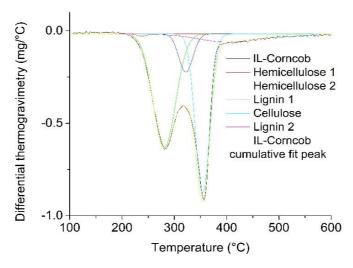


Figure 13. IL-treated corncob deconvoluted differential thermogravimetric curve.

Table 14. Statistcs and ANOVA.

| Statistcs | |
|-------------------------|----------|
| Number of Points | 2817 |
| Degrees of Freedom | 2812 |
| Reduced Chi-Sqr | 5.57E-04 |
| Residual Sum of Squares | 1.56576 |
| Adj. R-Square | 0.98757 |

| ANOVA | DF | Sum of Squares | Mean Square | F Value |
|-------------------|------|----------------|-------------|----------|
| Regression | 5 | 172.9983 | 34.59966 | 62138.49 |
| Residual | 2812 | 1.56576 | 5.57E-04 | |
| Uncorrected Total | 2817 | 174.56407 | | |
| Corrected Total | 2816 | 126.19144 | | |

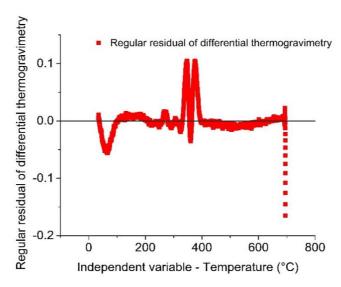


Figure 14. Regular residual from differential thermogravimetric of IL-treated corncob.

Adjusted model equation – IL-treated corncob cumulative fit peak (Equation 7).

$$y = y0 + \left\{ \frac{A}{(w \times \sqrt{\pi/2})} \times e^{\left[-2 \times \frac{(x - xc)^2}{w^2}\right]} \right\}$$
 (7)

• **NaOH-IL-treated corncob compositional analysis** - deconvolution parameters and statistic.

Table 15. Deconvolution parameters. Where y0 is the initialization parameter, xc curve center, w curve width and A curve area.

| Parameters | | Value | Standard Error |
|----------------------|--------|----------|----------------|
| Hemicellulose peak 1 | y0 | -0.04048 | 0.00333 |
| Hemicellulose peak 1 | XC | 239.9392 | 3.76466 |
| Hemicellulose peak 1 | W | 25 | 0 |
| Hemicellulose peak 1 | Α | -3 | 0 |
| Hemicellulose peak 1 | sigma | 12.5 | 0 |
| Hemicellulose peak 1 | FWHM | 29.43525 | 0 |
| Hemicellulose peak 1 | Height | -0.09575 | 0 |
| Hemicellulose peak 2 | y0 | -0.04048 | 0.00333 |
| Hemicellulose peak 2 | XC | 277.8687 | 5.77703 |
| Hemicellulose peak 2 | W | 28 | 0 |
| Hemicellulose peak 2 | Α | -13 | 0 |
| Hemicellulose peak 2 | sigma | 14 | 0 |
| Hemicellulose peak 2 | FWHM | 32.96748 | 0 |
| Hemicellulose peak 2 | Height | -0.37045 | 0 |
| Cellulose | y0 | -0.04048 | 0.00333 |
| Cellulose | XC | 277.8688 | 2.15564 |
| Cellulose | W | 32 | 0 |
| Cellulose | Α | -43 | 0 |
| Cellulose | sigma | 16 | 0 |
| Cellulose | FWHM | 37.67712 | 0 |
| Cellulose | Height | -1.07216 | 0 |

| Lignin | y0 | -0.04048 | 0.00333 | |
|--------|--------|----------|---------|--|
| Lignin | XC | 300 | 0 | |
| Lignin | W | 120 | 0 | |
| Lignin | А | -5 | 0 | |
| Lignin | sigma | 60 | 0 | |
| Lignin | FWHM | 141.2892 | 0 | |
| Lignin | Height | -0.03325 | 0 | |

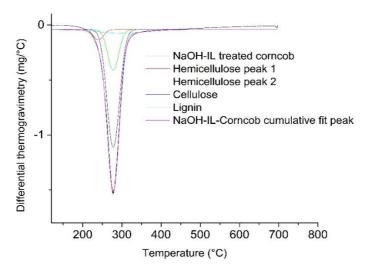


Figure 15. NaOH-IL-treated corncob deconvoluted differential thermogravimetric curve.

 Table 16.
 Statistcs and ANOVA.

| Statistcs | |
|-------------------------|----------|
| Number of Points | 2859 |
| Degrees of Freedom | 2855 |
| Reduced Chi-Sqr | 0.03176 |
| Residual Sum of Squares | 90.66297 |
| Adj. R-Square | 0.71065 |

| ANOVA | DF | Sum of Squares | Mean Square | F Value |
|--------------------------|------|----------------|-------------|-----------|
| Regression | 4 | 274.96236 | 68.74059 | 2164.6588 |
| Residual | 2855 | 90.66297 | 0.03176 | |
| Uncorrected Total | 2859 | 365.62533 | | |
| Corrected Total | 2858 | 313.66171 | | |

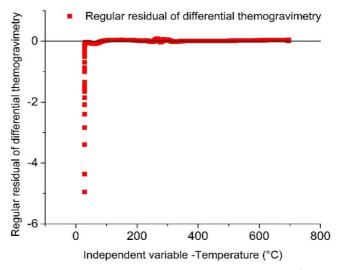


Figure 16. Regular residual from differential thermogravimetric of NaOH-IL-treated corncob.

Adjusted model equation – IL-treated corncob cumulative fit peak (Equation 8).

$$y = y0 + \{\frac{A}{(w \times \sqrt{\pi/2})} \times e^{\left[-2 \times \frac{(x - xc)^2}{w^2}\right]}\}$$
 (8)

Appendix B

This appendix presents additional information regarding the life cycle inventory used to implement the life cycle assessment model.

Supplementary information - Life cycle Inventory and additional Information.

Table 1. Processes used in the life cycle assessment obtained from Ecoinvent 3.3.

| Process | Process name in the Ecoinvent 3.3 | | | | | |
|---|---|--|--|--|--|--|
| Electricity production and | Electricity voltage transformation from high to medium/electricity | | | | | |
| distribution | medium voltage/ Cutoff, U -BR | | | | | |
| Production of NaOH | Sodium hydroxide, without water, in 50% solution state, Cutoff, U -RER | | | | | |
| | Chlorite dioxide, Cutoff, U - RER | | | | | |
| Production of NaClO ₂ ^a | Hydrogen peroxide, without water, in 50% solution state, Cutoff, U - RER | | | | | |
| | Sodium hydroxide, without water, in 50% solution state, Cutoff, U -RER | | | | | |
| Production of H ₂ SO ₄ | Sulfuric acid, Cutoff, U - RER | | | | | |
| Production of CH₃COOH | Acetic acid, without water, in 98% solution state, Cutoff, U -RER | | | | | |
| Production of Ac ₂ O | Acetic anhydride, Cutoff, U -RER | | | | | |
| Production of CH₃OH | Methanol, Cutoff, U -RER | | | | | |
| Production of C₂H₅OH | Ethanol, without water, in 99.7% solution state, from ethylene, Cutoff, U | | | | | |
| | -RER | | | | | |
| Production of iodine | Iodine, Cutoff, U -RER | | | | | |
| Production of sodium | Sodium sulfite, Cutoff, U -RER | | | | | |
| thiosulfate ^b | Sulfur, Cutoff, U -GLO | | | | | |
| Production of deionized | Water, deionized, from tap water, at user, Cutoff, U -RoW | | | | | |
| water | | | | | | |
| Production of CH ₂ Cl ₂ | Dichloromethane, Cutoff, U -RER | | | | | |

^aSodium chlorite production process does not exist in the Ecoinvent database. Thus, it was modeled in OpenLCA according to the production process proposed by lan et al., (2007).

Table 2. Life cycle inventory for producing 10 g of cellulose acetate by means the green acetylation approach.

| Green acetylation of o | cellulose | Materials inventory | | | | |
|------------------------|--------------------------|--------------------------|----------|------|--|--|
| | First stage of t | he treatment | | | | |
| Input | Output | Amount Unit | | | | |
| Dried treated corncob | | Electricity | 13608 | kJ | | |
| lodine | | Dried pretreated corncob | 6.25 | g | | |
| Ethanol | Acetylated solution | lodine | 9.375 | g | | |
| Sodium thiosulfate | | Acetic anhydride | 312.5 | ml | | |
| Acetic anhydride | | Sodium thiosulfate | 156.25 | ml | | |
| Energy | | Ethanol | 937.5 | ml | | |
| | Second stage of | the treatment | | | | |
| Input | Output | | Amount | Unit | | |
| Acetylated solution | Dried cellulose acetate | Electricity | 95775 | kJ | | |
| Deionized water | Filtered/washed solution | Acetylated solution | 1279.91 | g | | |
| Dichloromethane | dichloromethane | Ethanol | 1171.875 | ml | | |
| Electricity | | Deionized water | 390.695 | ml | | |
| | | Dichloromethane | 937.5 | ml | | |

Additional information used to build the inventory:

Sodium thiosulfate production process does not exist in Ecoinvent database. Therefore, it was modeled in OpenLCA according to production process proposed by Ferreira (2017).

- The process was modeled taking into account the functional unit of 10g of cellulose acetate and a yield of 160%.
- Formulation for energy consume (E): E = 1000*kw*t(s). In which t refers to time of processing and kw the potency of the equipment.
- The recovery of 90 ml of dichloromethane took 10 min.
- The green acetylation methodology was based on Das et al. (2015).

Table 3. Additional information to build the inventory

| Sodium thiosulfate solubility | 1.667 g/cm3 |
|-------------------------------|-------------|
| Acetic anhydride density | 1.08 g/cm3 |
| Ethanol density | 0.789 g/cm3 |
| Dichloromethane density | 1.33 g/cm3 |

Table 4. Life cycle inventory for producing 10 g of cellulose acetate by means the standard acetylation approach

| асетупации арргиасті. | | | | | |
|----------------------------|-------------------------|---------------------|----------|------|--|
| Standard acetylation of ce | ellulose | Materials inventory | | | |
| | First stage of t | the treatment | | | |
| Input | Output | | Unit | | |
| Dried treated corncob | | Electricity | 120420 | kJ | |
| Acetic acid | | Dried pretreated | 7.143 | g | |
| | | corncob | | | |
| Acetic anhydride | Acetylated solution | Acetic acid | 225 | g | |
| Sulfuric acid | | Acetic anhydride | 100 | g | |
| Methanol | | Sulfuric acid | 0.63 | g | |
| Electricity | | Methanol | 285.72 | ml | |
| | Second stage of | the treatment | | | |
| Input | Output | | Amount | Unit | |
| Acetylated solution | Dried cellulose acetate | Electricity | 971145.5 | kJ | |
| Deionized water | | | 559.07 | g | |
| | solution | | | | |
| Methanol | dichloromethane | Methanol | 1785.75 | ml | |
| Dichloromethane | | Dichloromethane | 1071.45 | ml | |
| Electricity | | | | | |

Additional information used to build the inventory:

- The process was modeled taking into account the functional unit of 10g of cellulose acetate and a yield of 140%;
- Formulation for energy consume (E): E = 1000*kw*t(s). In which t refers to time of processing and kw the potency of the equipment;
- The green acetylation methodology was based on Jo et al. (2016), Coletti et al (2013), Bello et al. (2016), Mostafa et al. (2018) and Cao et al. (2018).

Table 5. Impact assessment associated with LCA scenarios.

| Impact category | Model | | AC star | ndrd convt. | AC stndre | d Green process | Unit |
|---|------------------|-----------|----------|-------------|-----------|--------------------|----------------------------|
| GWP | IPCC 2013 | IPCC 2013 | | 1 | 1.89E+01 | 1.76E+01 | kg CO2- Eq |
| land use | - CML 2001 | | 2.15E-02 |) | 2.01E-02 | 2.03E-02 | m2a |
| competition ODP | CML 2001 | | 9.68E-05 | 5 | 9.67E-05 | 8.46E-05 | kg CFC- 11-Eq |
| stratospheric ozone depletion ODP 20a | CML 2001 | | 1.07E-06 | 5 | 9.80E-07 | 8.44E-07 | kg CFC- 11-Eq |
| HTc | USEtox | | 6.72E-07 | 7 | 6.42E-07 | 6.05E-07 | CTU |
| HTnc | USEtox | | 2.44E-06 | õ | 2.32E-06 | 2.27E-06 | CTU |
| PM | Selected LC | l results | 1.11E-02 |) | 1.04E-02 | 9.28E-03 | kg |
| IR | ILCD 1.0.8 | 2016 | 1.65E+0 | 0 | 1.53E+00 | 1.34E+00 | kg U235- |
| FE | ReCiPe (H) | | 4.42E-03 | 3 | 4.14E-03 | 4.11E-03 | Eq kg P-Eq |
| ME | ReCiPe (H) | | 1.82E-02 | 2 | 1.75E-02 | 1.61E-02 | kg N-Eq |
| POF | ReCiPe (H) | | 5.99E-02 | 2 | 5.77E-02 | 5.56E-02 | kg |
| environmental impact | TRACI - | | 4.95E+0 | 0 | 4.73E+00 | 4.40E+00 | NMVOC moles of H+-Eq |
| acidification TA | ILCD 1.0.8 | 2016 | 1.14E-01 | [| 1.09E-01 | 1.01E-01 | mol H+- Eq |
| TE | ILCD 1.0.8 | 2016 | 1.93E-01 | L | 1.86E-01 | 1.71E-01 | mol N-Eq |
| ET | USEtox | | 3.20E+0 | 1 | 3.02E+01 | 3.24E+01 | CTU |
| Rf | Cummulative | 2.62E+02 | 2.50E+02 | 2.52E+02 | MJ-Eq | | |
| | exergy demand | | | | | | |
| fossil | ReCiPe (H) | 6.38E+00 | 6.09E+00 | 6.10E+00 | kg oil- | | |
| depletion - | | | | | Eq | | |
| FDP | | | | | | | |
| resources - | ILCD 1.0.8 | 1.34E-04 | 1.27E-04 | 1.78E-04 | kg Sb- | | |
| mineral, | 2016 | | | | Eq | | |
| fossils and | | | | | | | |
| renewables | | | | | | | |

Table 6. Environmental impacts after modification of the green approach parameters for the

sensitivity analysis study.

| Impact category | green | yield 140% | Yield 150% | yield 160% | yield 170% | Unit |
|--|----------|---------------|---------------|---------------|---------------|------------------|
| climate change - GWP 100a | 1.76E+01 | 1.99E+01 | 1.89E+01 | 1.81E+01 | 1.74E+01 | kg CO2- Eq |
| stratospheric ozone depletion - ODP steady state | 8.46E-05 | 9.68E-05 | 9.04E-05 | 8.47E-05 | 7.98E-05 | kg CFC- 11-Eq |
| ecotoxicity - total | 3.24E+01 | 3.20E+01 | 3.05E+01 | 2.92E+01 | 2.81E+01 | CTU |
| human toxicity - carcinogenic | 6.05E-07 | 6.72E-07 | 6.37E-07 | 6.08E-07 | 5.82E-07 | CTU |
| human toxicity - non- carcinogenic | 2.27E-06 | 2.44E-06 | 2.31E-06 | 2.20E-06 | 2.11E-06 | CTU |
| air - particulates, < 2.5 um | 9.28E-03 | 1.11E-02 | 1.06E-02 | 1.02E-02 | 9.85E-03 | kg |
| ecosystem quality - freshwater and terrestrial acidification | 1.01E-01 | 1.14E-01 | 1.08E-01 | 1.04E-01 | 9.93E-02 | mol H+- Eq |
| ecosystem quality - terrestrial eutrophication | 1.71E-01 | 1.93E-01 | 1.83E-01 | 1.74E-01 | 1.67E-01 | mol N-Eq |
| human health - ionising radiation | 1.34E+00 | 1.65E+00 | 1.58E+00 | 1.52E+00 | 1.48E+00 | kg U235- Eq |
| freshwater eutrophication - FEP | 4.11E-03 | 4.42E-03 | 4.23E-03 | 4.06E-03 | 3.92E-03 | kg P-Eq |
| marine eutrophication - MEP | 1.61E-02 | 1.82E-02 | 1.72E-02 | 1.64E-02 | 1.57E-02 | kg N-Eq |
| photochemical oxidant formation - POFP | 5.56E-02 | 5.99E-02 | 5.66E-02 | 5.39E-02 | 5.15E-02 | kg NMVOC |
| fossil - non-renewable energy resources, fossil | 2.56E+02 | 2.68E+02 | 2.54E+02 | 2.43E+02 | 2.33E+02 | MJ-Eq |

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